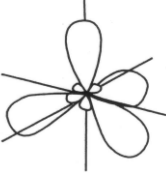
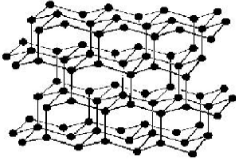

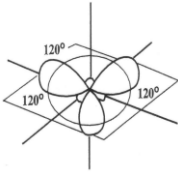
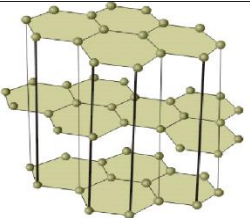
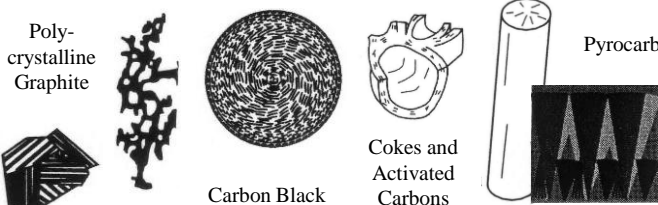

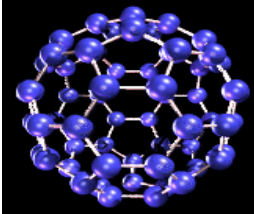
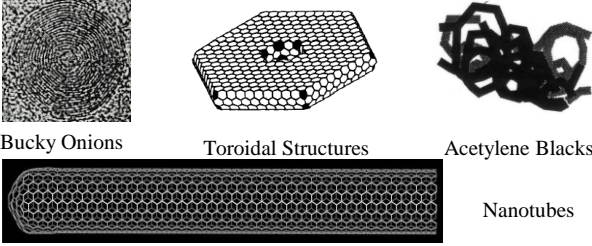
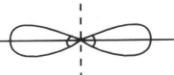
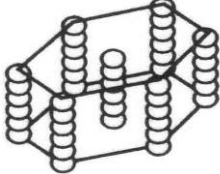
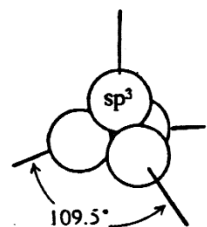


# Li-ion電池負極(I)

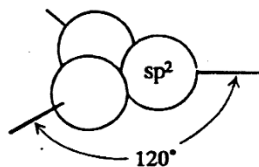
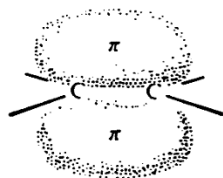
1. Li-ion電池および負極
2. 黒鉛系負極

Bonding Hybridization	Allotropes	Derived and Defective Forms
 <p data-bbox="417 382 475 414"><b>SP<sup>3</sup></b></p>	 <p data-bbox="618 385 795 414"><b>Cubic diamond</b></p>	 <p data-bbox="1066 376 1306 405"><b>Diamond-like Carbon</b></p>
 <p data-bbox="417 665 475 696"><b>SP<sup>2</sup></b></p>	 <p data-bbox="598 674 830 702"><b>Hexagonal graphite</b></p>	 <p data-bbox="927 474 1020 545">Polycrystalline Graphite</p> <p data-bbox="1128 634 1251 656">Carbon Black</p> <p data-bbox="1290 588 1383 656">Cokes and Activated Carbons</p> <p data-bbox="1506 488 1619 511">Pyrocarbons</p> <p data-bbox="1418 668 1541 691">Carbon Fibers</p>
 <p data-bbox="351 882 583 953"><b>SP<sup>2+ε</sup> rehybridization</b></p>	 <p data-bbox="656 939 772 968"><b>Fullerene</b></p>	 <p data-bbox="911 856 1039 879">Bucky Onions</p> <p data-bbox="1108 862 1282 885">Toroidal Structures</p> <p data-bbox="1352 862 1506 885">Acetylene Blacks</p> <p data-bbox="1391 925 1487 948">Nanotubes</p>
 <p data-bbox="417 1142 475 1173"><b>SP<sup>1</sup></b></p>	 <p data-bbox="672 1182 776 1210"><b>Carbyne</b></p>	<p data-bbox="923 1016 1607 1188">Ref.) Bourrat, X. Structure in Carbons and Carbon Artifacts. In: <i>Sciences of Carbon Materials</i>. Marsh, H.; Rodriguez-Reinoso, F., Eds., Universidad de Alicante, <b>2000</b>. pp1-97.</p>

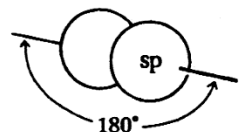
## Carbon Allotropes



sp<sup>3</sup> 混成軌道



sp<sup>2</sup> 混成軌道



sp 混成軌道

π 軌道の概形

(a)

σ 軌道を球で表した時の配置

(b)

図-1 炭素の混成軌道

表-1 各種炭素-炭素結合の結合解離エネルギーと結合距離<sup>1)</sup>

化合物	結合解離エネルギー (kcal/mol)	結合距離 (Å)
H <sub>3</sub> C-C <sub>3</sub> H	88	1.53
H <sub>2</sub> C=C <sub>2</sub> H	163	1.34
HC≡CH	198	1.21

表-2 炭素同素体の種類<sup>3)</sup>

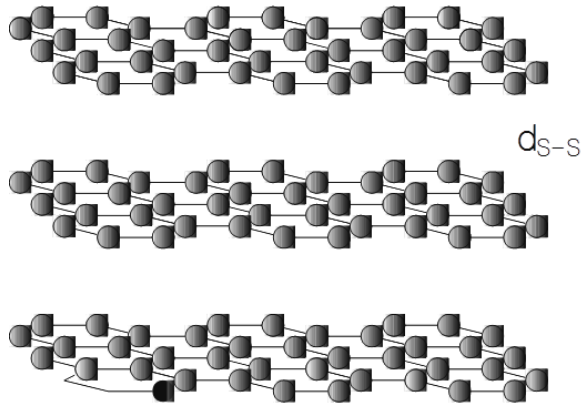
結合の種類	配位数	炭素同素体
sp	2	カルビン (ポリイン, クムレン)
sp <sup>2</sup>	3	グラファイト (六方晶, 菱面体晶) フラーレン (C <sub>60</sub> , C <sub>70</sub> , バックキイチューブなど)
sp <sup>3</sup>	4	ダイヤモンド (立方晶, 六方晶, 菱面体晶*) ダイヤモンド多形体 (6H, bc-8*など) ダイヤモンドライクカーボン (DLC), i-カーボン
イオンまたは 金属的	6	単純立方晶*, β-スズ型*
	8	体心立方晶*
	12	面心立方晶*, 六方最密充填*

\* 実!

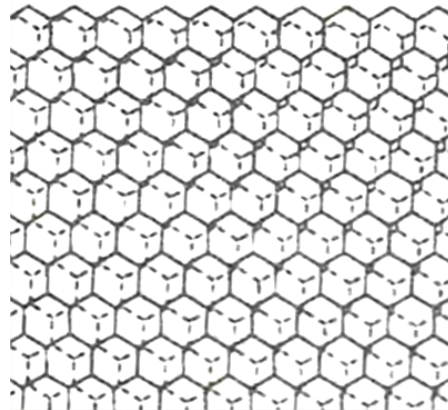
表-4 IV族sp<sup>3</sup>立方晶体の性質<sup>6)</sup>

性質	ダイヤモンド	β-SiC	Si
格子定数 (Å)	3.567	4.358	5.430
密度 (g/cm <sup>3</sup> )	3.515	3.216	2.328
熱膨張率 (×10 <sup>-6</sup> /°C)	1.1	4.7	2.6
融点 (°C)	4000	2540	1420
バンドギャップ (eV)	5.45	3.0	1.1
キャリア移動度 (cm <sup>2</sup> /(V·S))			
電子	2200	400	1500
ホール	1600	50	600
熱伝導率 (W/(cm·K))	20	5	1.5
硬度 (kg/mm <sup>2</sup> )	10000	3500	1000

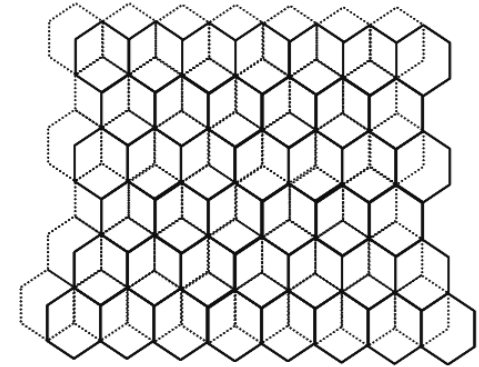
# Molecular structures of graphite



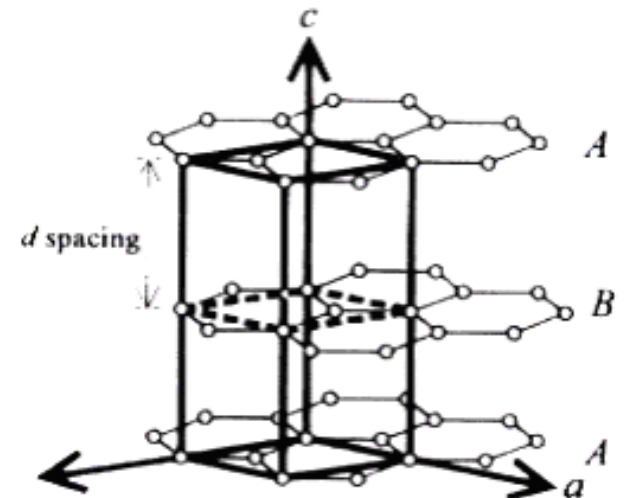
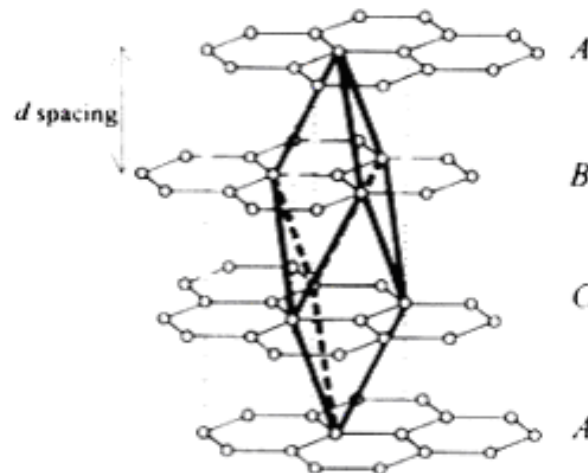
(a) Basic Structure of Graphite



(b) Turbostratic structure (low crystallinity)



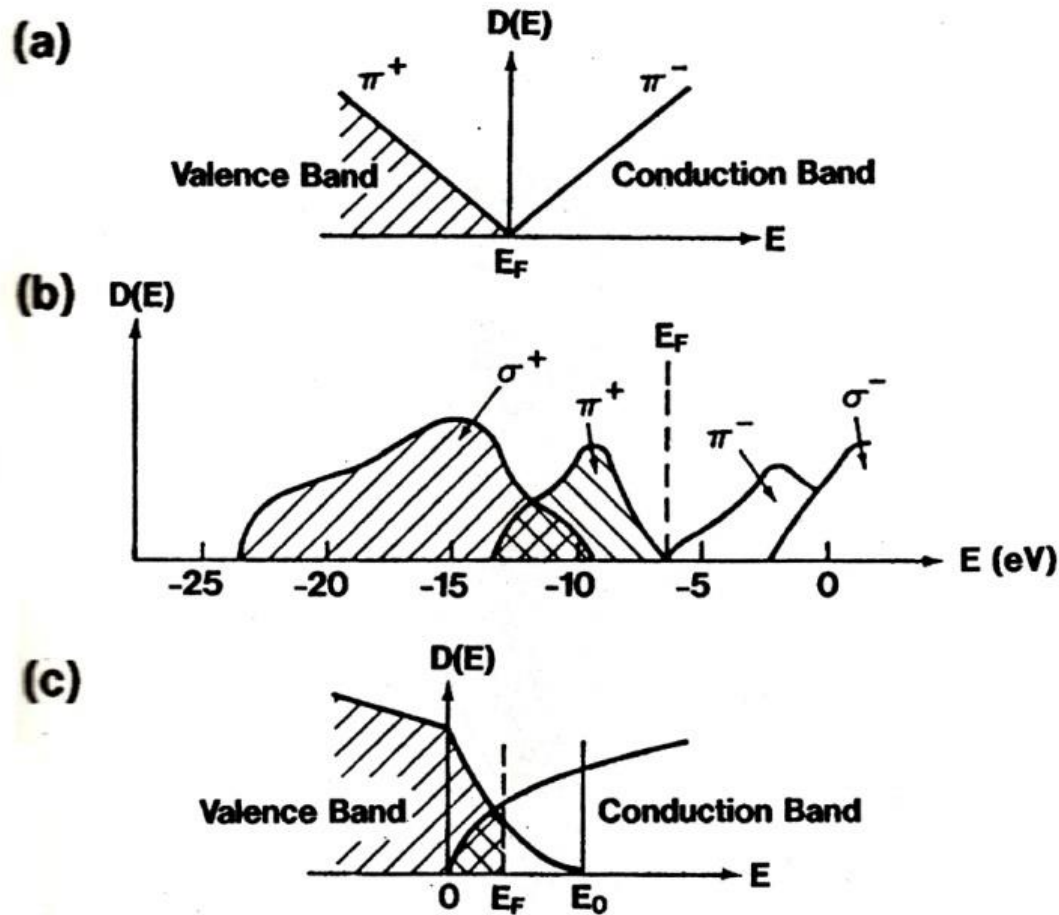
(c) Graphitic structure (high crystallinity)





# Characteristics of carbons

- **Thermal stability**
- **High thermal and electric conductivities**  
SWNT, Diamond : 4000 W/mK, K-11  
carbon fiber: 1100 W/mK
- **Small heat expansion**
- **High thermal shock properties**
- **High chemical stability**
- **Abrasion and lubricant properties**
- **High mechanical properties**



電子の状態密度  $D(E)$  のエネルギー  $E$  依存: 2次元黒鉛に対するフェルミエネルギー  $E_F$  近傍の  $\pi$  電子の電子状態分布 (a), 黒鉛の全エネルギー領域における電子状態分布密度分布 (b), および黒鉛の  $E_F$  近傍の  $\pi$  電子の状態密度分布 (c)

# Carbon is key element for Batteries !!

## ① Li-ion



[High capacity]

(+) :  $\text{LiCoO}_2$

(-) : **Carbon(Graphite)**

Conductor : **Carbon**

## ② Dry Battery



[Cheap]

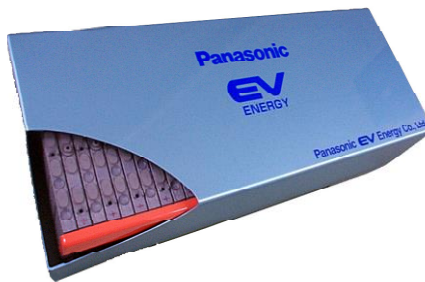
[Easy Available]

(+) :  $\text{MnO}_2$

(-) : Zn

Conductor : **Carbon**

## ③ Ni-MH



[High power]

[Total balance]

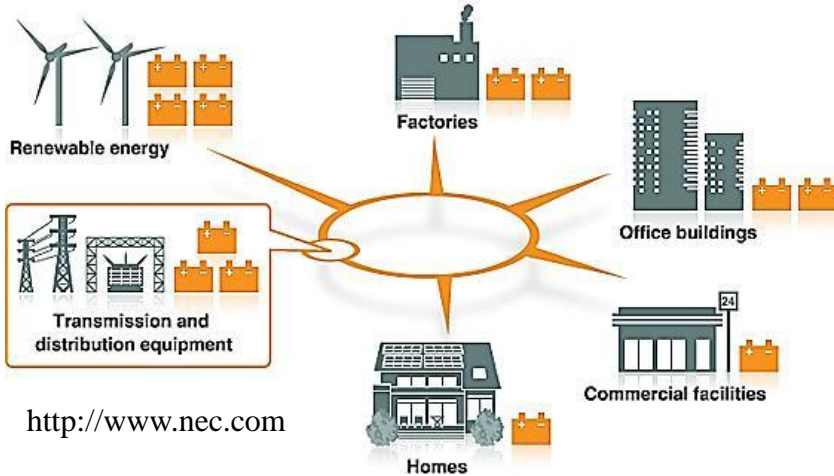
(+) :  $(\text{Ni-Co})(\text{OH})_2$

(-) :  $\text{Mm}(\text{Ni-Mn-Al-Co})_5$

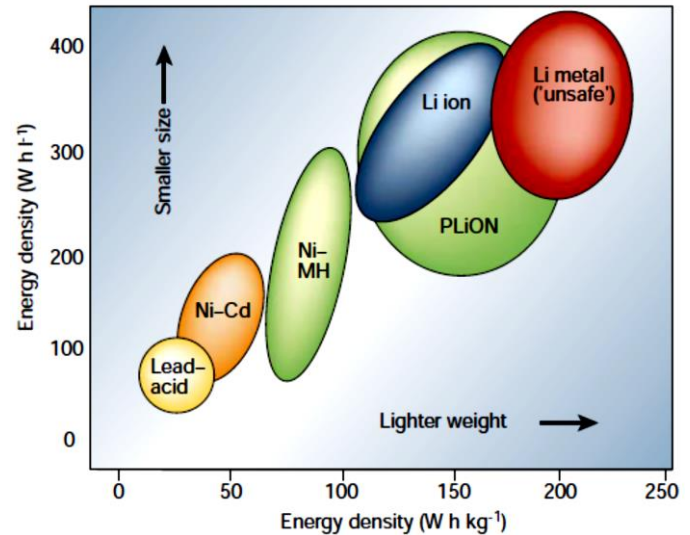
substrate: Nickel and **Carbon**

# Applications and necessity of Li-ion battery

## Energy storage system in smart grid



## Energy density of various rechargeable batteries



J.M. Tarascon, M. Armand, Nature 414 (2001) 359.

Li-ion battery is paid much attention as power sources of ESS and electric vehicles in a variety of rechargeable batteries.

## Power source of electric vehicles

### ICE



< Toyota Camry >

### Electric motor



< Toyota RAV4 EV >

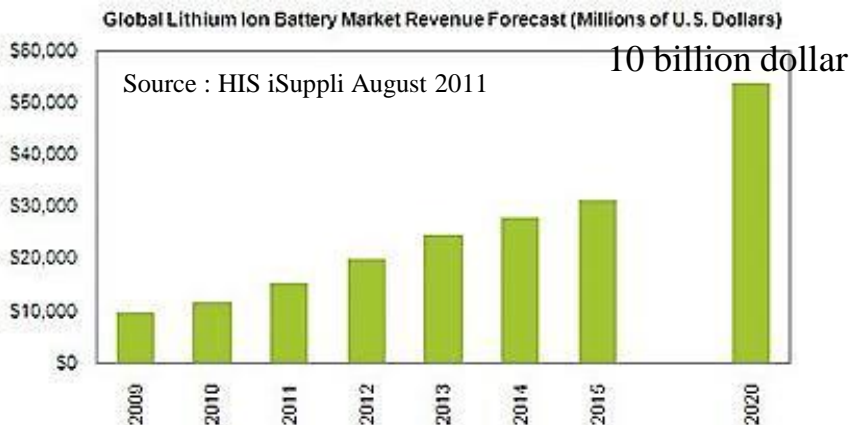
ICE : Internal combustion engine, ESS : Energy storage system

# Global market and requirements of Li-ion battery

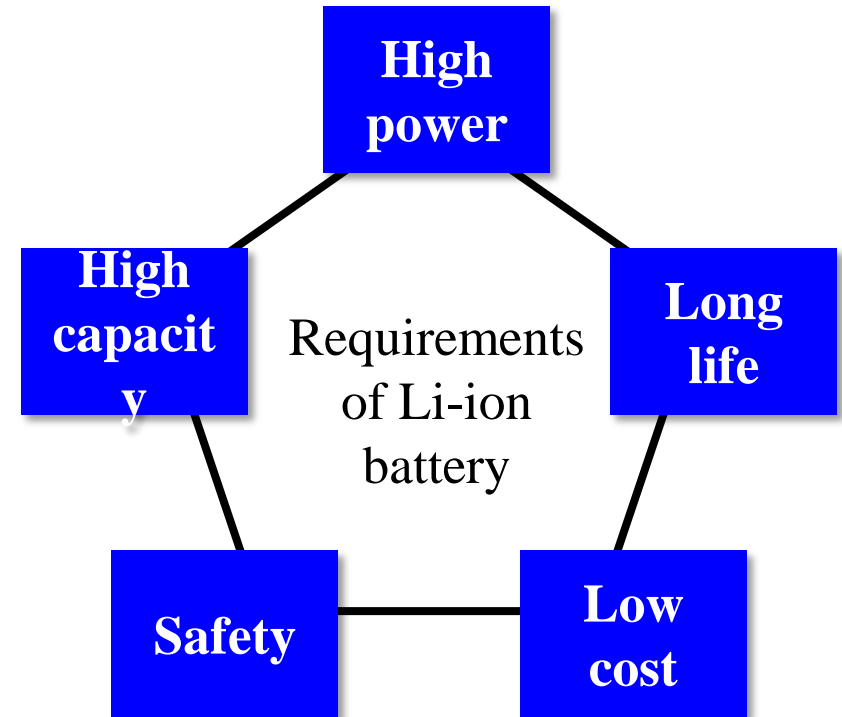
## Global market of Li-ion battery in ESS



## Global market of Li-ion battery in EV

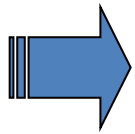


## Requirements of Li-ion battery as power sources of ESS and EV



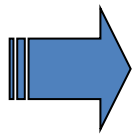
# Carbon Electrode for Li-ion Battery

- **Graphite electrode is currently established.**
  - Low cost with cheaper natural graphite
  - Limited capacity less than 372 mAh/g
  - Limited power density



Larger power density for hybrid vehicle

➔ Glassy carbon with small crystalline unit (Low Cond.)  
Thinner carbon nanofiber



Larger capacity

➔ Glassy carbon with large inner surface  
Si or Sn family (Large volumetric change at Ch/Disc)  
⇒ **Functional nano-composites**

# Roles of Carbon for Anode of Li-ion Batteries

- Anodic Electrode to Hold Reduced Li-ion Intercalation → Graphite  
Surface Electron Transfer into Sealed Void → Carbon
- Electron Conductive Material  
Anodic Carbon and Cathodes Material
- Expansion Moderation  
Holding and Release of Ion Is Accompanied with Volumetric Charge  
Larger Capacity per Volume → Larger Expansion



# 電池負極物質

- 電池の中で還元剤として機能

自身が酸化(イオン化)し、電解質に溶解することで負極は負に帯電する。活量=1の水溶液中、標準状態における標準水素電極に対するその電位は標準電極電位 $E^0$ と呼ばれ、元素ごとに表に示す異なる値を取る。

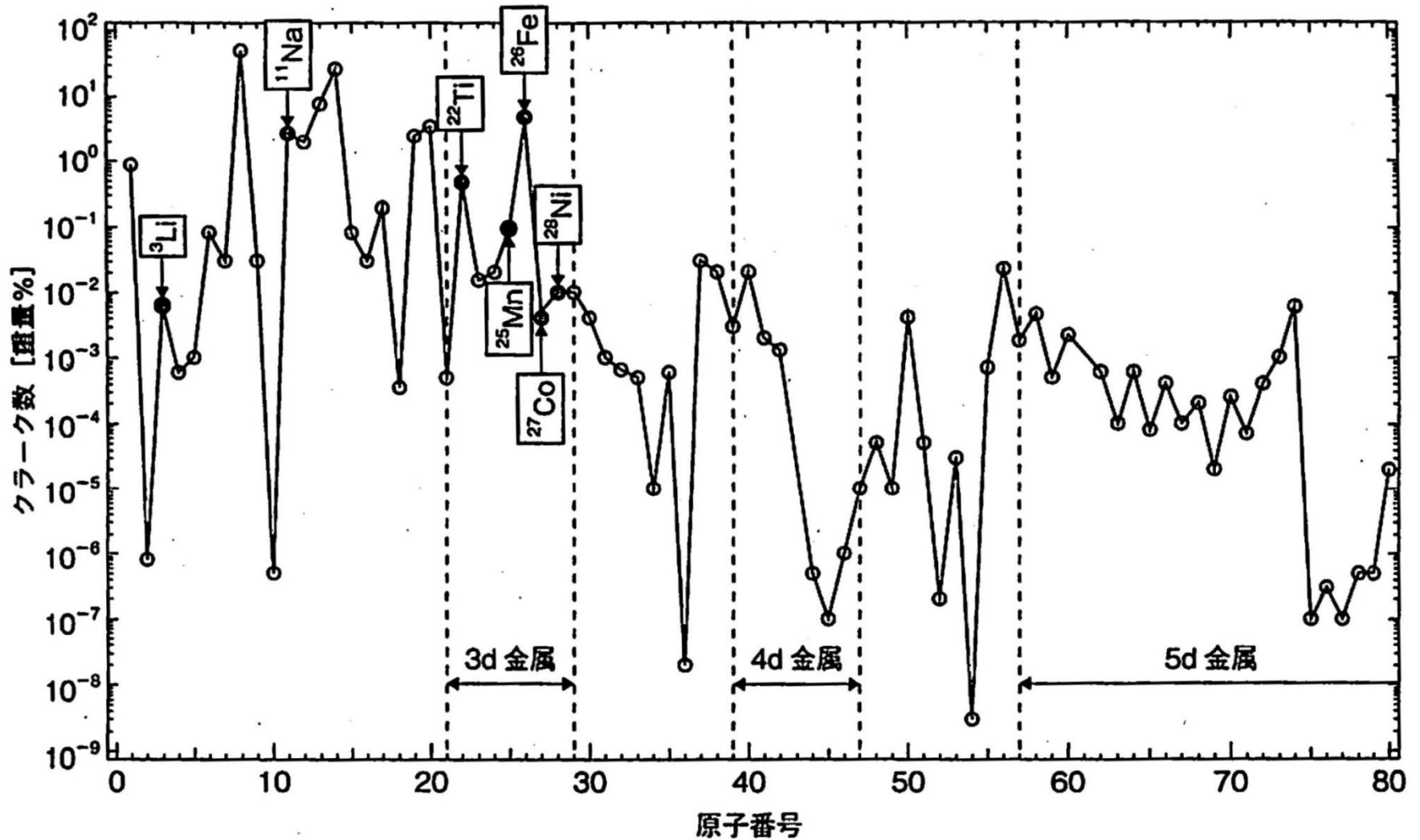
$$E^0 = -\Delta G/nF \text{ (イオン化傾向、真空中で陽イオンになりやすさ)}$$

表1 代表的な金属元素の標準電極電位と第1イオン化エネルギーの相関

電極	放電反応	ギブスエネルギー	標準電極電位	イオン化エネルギー
Li	$\text{Li} \rightarrow \text{Li}^+ + \text{e}^-$	-293.31 kJ/mol	-3.045 V	520 kJ/mol
K	$\text{K} \rightarrow \text{K}^+ + \text{e}^-$	-283.27 kJ/mol	-2.925 V	419 kJ/mol
Ca	$\text{Ca} \rightarrow \text{Ca}^{2+} + 2\text{e}^-$	-553.58 kJ/mol	-2.866 V	590 kJ/mol
Na	$\text{Na} \rightarrow \text{Na}^+ + \text{e}^-$	-261.9 kJ/mol	-2.714 V	496 kJ/mol
Mg	$\text{Mg} \rightarrow \text{Mg}^{2+} + 2\text{e}^-$	-454.8 kJ/mol	-2.363 V	738 kJ/mol
Al	$\text{Al} \rightarrow \text{Al}^{3+} + 3\text{e}^-$	-485 kJ/mol	-1.662 V	578 kJ/mol
Zn	$\text{Zn} \rightarrow \text{Zn}^{2+} + 2\text{e}^-$	-147.06 kJ/mol	-0.763 V	906 kJ/mol
Fe	$\text{Fe} \rightarrow \text{Fe}^{2+} + 2\text{e}^-$	-78.9 kJ/mol	-0.44 V	759 kJ/mol
H	$\text{H} \rightarrow \text{H}^+ + \text{e}^-$	0 kJ/mol	0 V	1,312 kJ/mol
Cu	$\text{Cu} \rightarrow \text{Cu}^{2+} + 2\text{e}^-$	65.49 kJ/mol	0.337 V	745 kJ/mol
Ag	$\text{Ag} \rightarrow \text{Ag}^+ + \text{e}^-$	77.1 kJ/mol	0.7991 V	731 kJ/mol



# 地殻中の各元素の存在比



# Li2次電池における負極材の研究動向

## 1. Li金属負極:

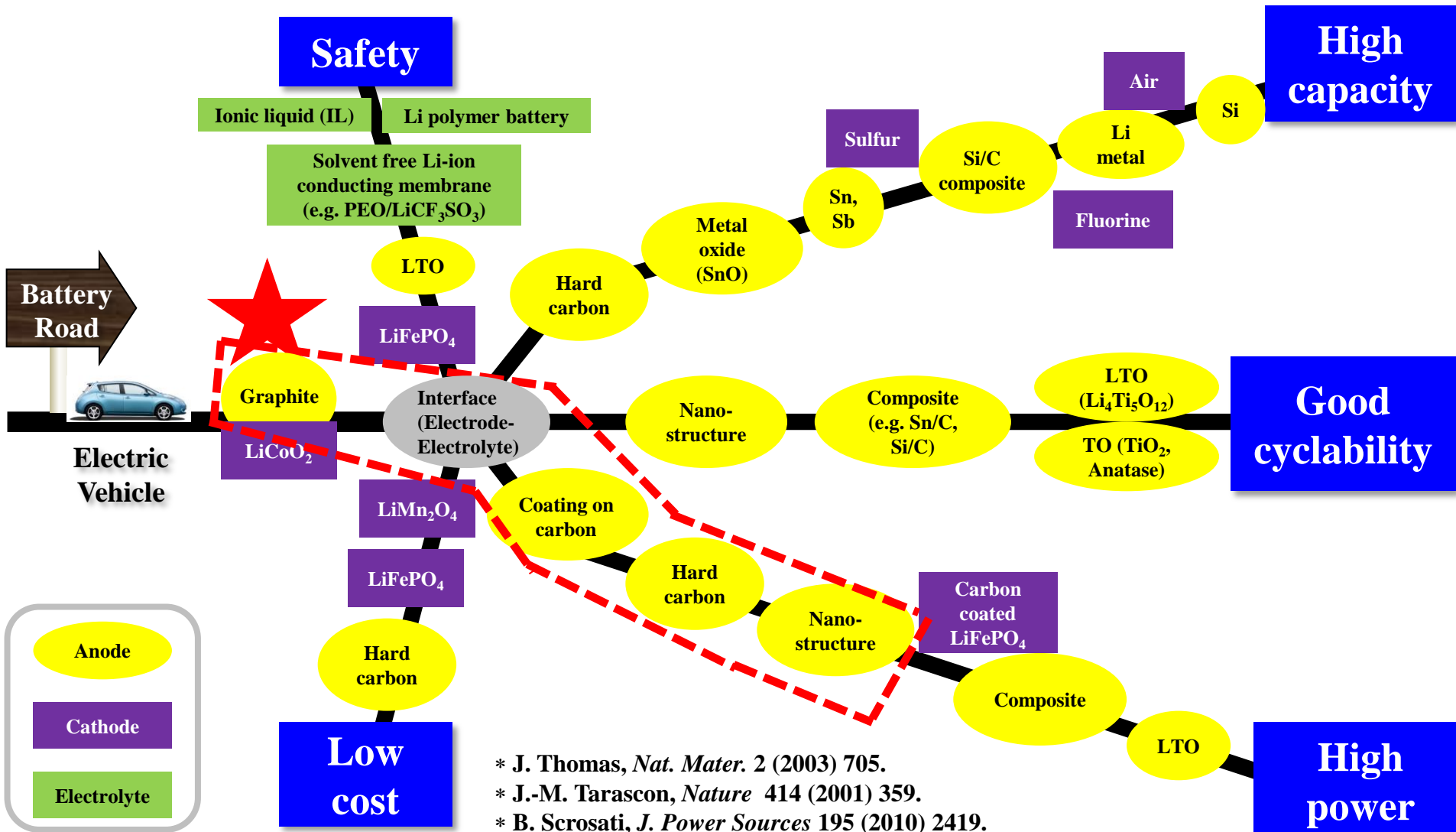
- 還元力の強さが仇となり、殆どの電解液を還元分解してしまう問題点あり。
- 還元の際、Dendrite結晶状として還元
- モリエナジー(カナダ)1989年、NTT形態で内部短絡事故

## 2. Carbon電極

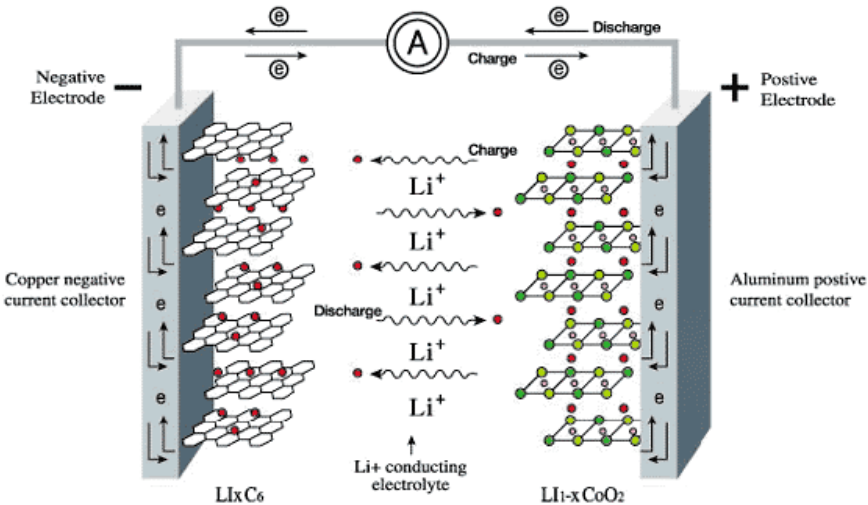
- 1991年SONYが採択、Li-ion電池化、世界初
- C6Li, 372 mAh/g

## 3. Si, Sn系、チタニア系、バナディウム系...

# Li-ion Battery Road



## Ch./Dis. Principle of Li-ion 2nd Batteries



## Anodic Materials for Li-ion 2<sup>nd</sup> Batteries

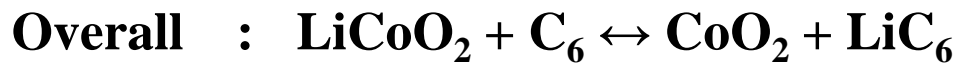
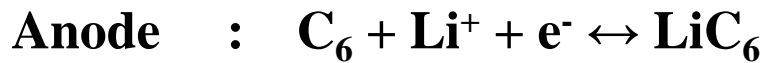
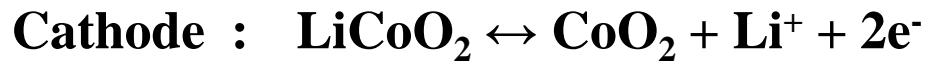
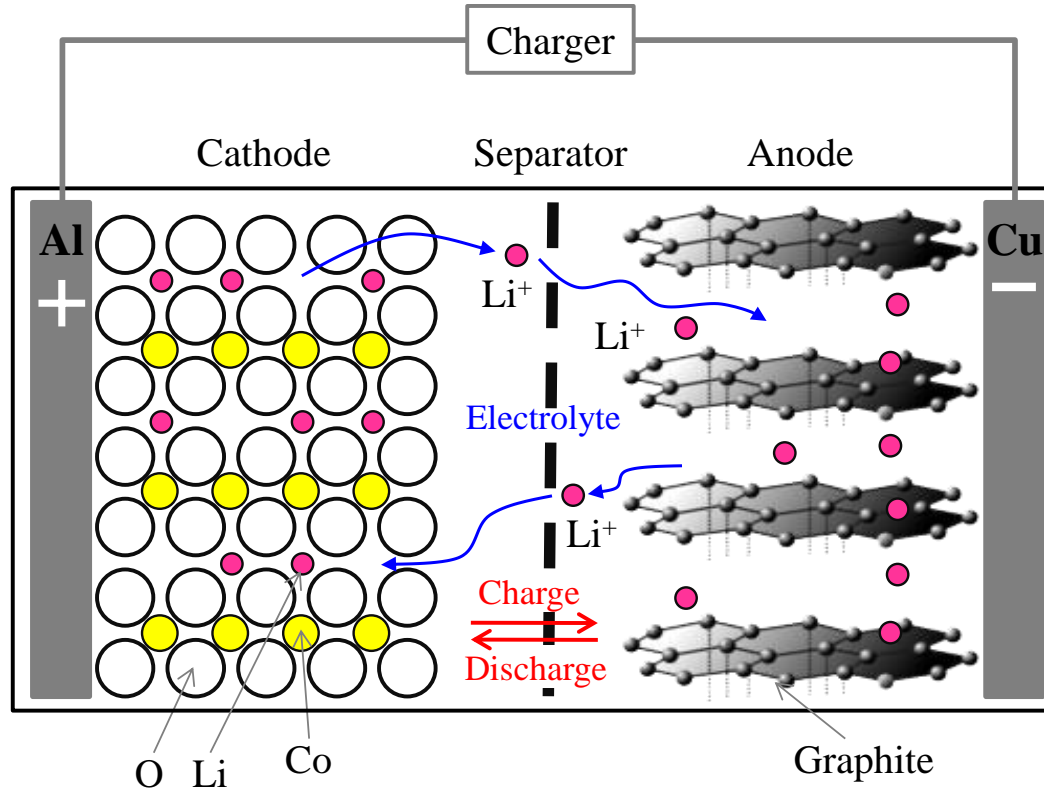
	Carbon	Si alloys	Li alloys
Theoretical Cap.(mAh/g)	372 (LiC <sub>6</sub> )	4200 (Li <sub>4.4</sub> Si)	3860
Present Stage	Commercialized	Developing	Developing
Merit	Low Cost Good Cycle Life Good Chemical Stability	High Capacity	High Capacity
De-merit	Low Rate Capability	High Volume Expansion ⇒ Bad Cycle	Strong Reaction ⇒ Bad Cycle & Thermal Stability
Materials	Graphite, Soft/Hard carbon	-	-
User	Sanyo, Matsushita, STC, A&T Battery, Shin-Kobe, GS, Moli, Mitsubishi, Sony, SDD, Hitachi Maxcel, LG Chem.	-	-

## Characteristics and materials of 2<sup>nd</sup> Batteries

[Ref. KISTI, Materials for 2<sup>nd</sup> Batteries \(2004/06\)](#)

		Ni-Cd	Ni-MH	Li-ion	Li polymer
Cathodic material		NiOOH	NiOOH	LiMO <sub>2</sub>	LiMO <sub>2</sub>
Anodic material		Cd	MH	Carbon	Carbon
Electrolyte		KOH/H <sub>2</sub> O	KOH/H <sub>2</sub> O	LiX/Organic Solution	LiX/Polymer electrolyte
Operating voltage(V)		1.2	1.2	3.6	3.6
Cycle		1000	1000	1200	1000
Self discharge rate (%/month)		-	20~25	< 10	≪ 10
Environmental pollutant		Yes	Yes	No	No
Energy density	Per weight (Wh/kg)	-	65	120	100
	Per volume (Wh/L)	160	240	280	220
Manufacturing company		Sanyo, Toshiba	Matsushita, Sanyo, Toshiba	Sony, Sanyo, Matsushita	Valence, Ultralife

# Mechanism of charge & discharge

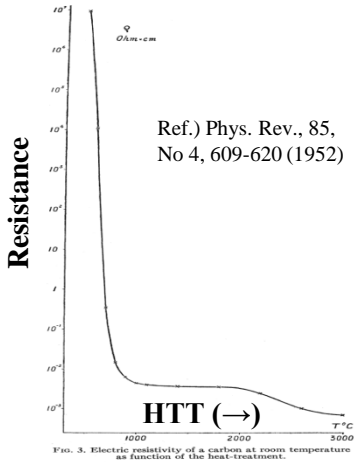


# Carbon materials of LIB

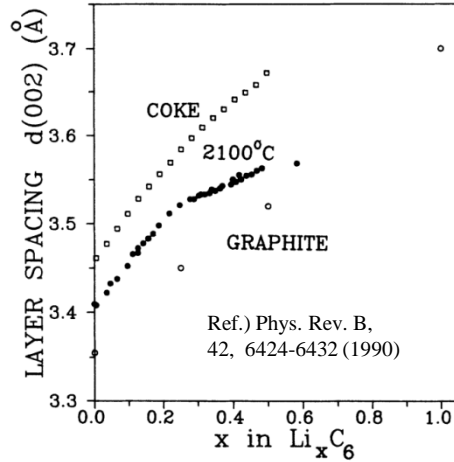
	Precursor	Advantages	Disadvantages
<b>Graphite</b> (over 2800°C)	Natural / Artificial graphite MCMB, Needle cokes VGCF	Low discharge potential ( $\approx 0.2V$ ) Long cycle life	Low discharge capacity (372 mAh/g) Poor rate performance High cost
<b>Soft Carbon</b> Graphitizable carbon (600~800°C)	MCMB Meso phase pitch Green cokes	High capacity (700~1000mAh/g) Low cost	High discharge potential ( $\approx 1.0V$ ) High irreversible capacity Poor cycle stability
<b>Hard Carbon</b> Non-graphitizable carbon (1000~1400°C)	Thermosetting polymer Glassy carbon, Coal Organic material Stabilized isotropic pitch	High capacity (400~700mAh/g) High rate performance Low discharge potential ( $\approx 0.1V$ ) Low cost	Large irreversible capacity

# Characteristics of Carbon Material

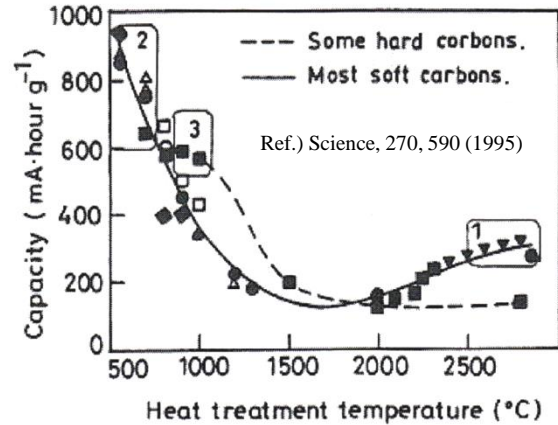
## HTT ↔ Resistance



## Li content ↔ d-spacing

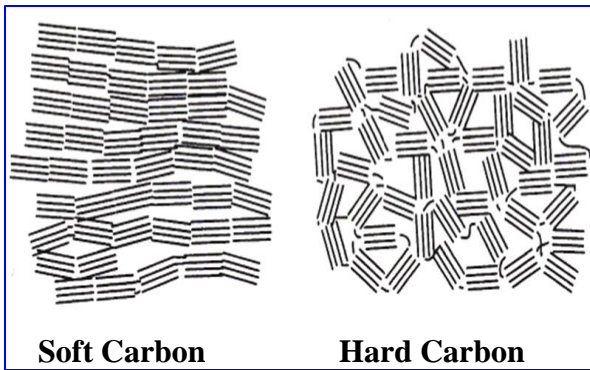


## HTT ↔ Capacity



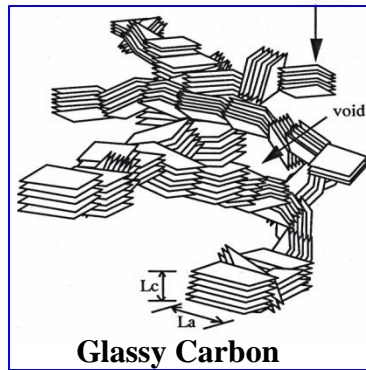
## Structural mechanism of carbon

Ref.) Proc. R. Soc. A209 (1951) 196-218



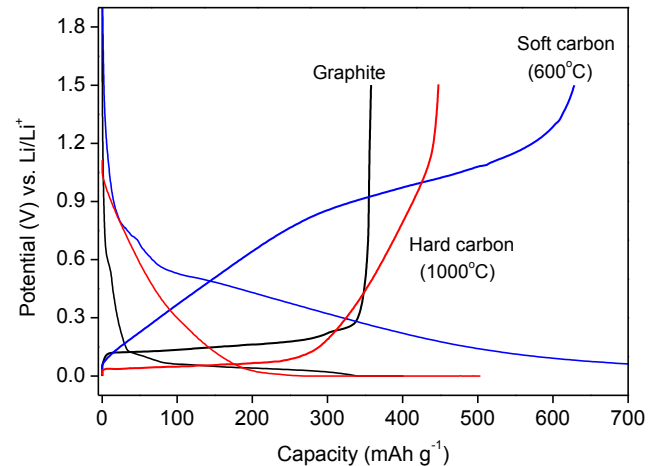
Franklin model

Ref.) Report of Kyushu Univ. 12 (1) (1998) 45-57



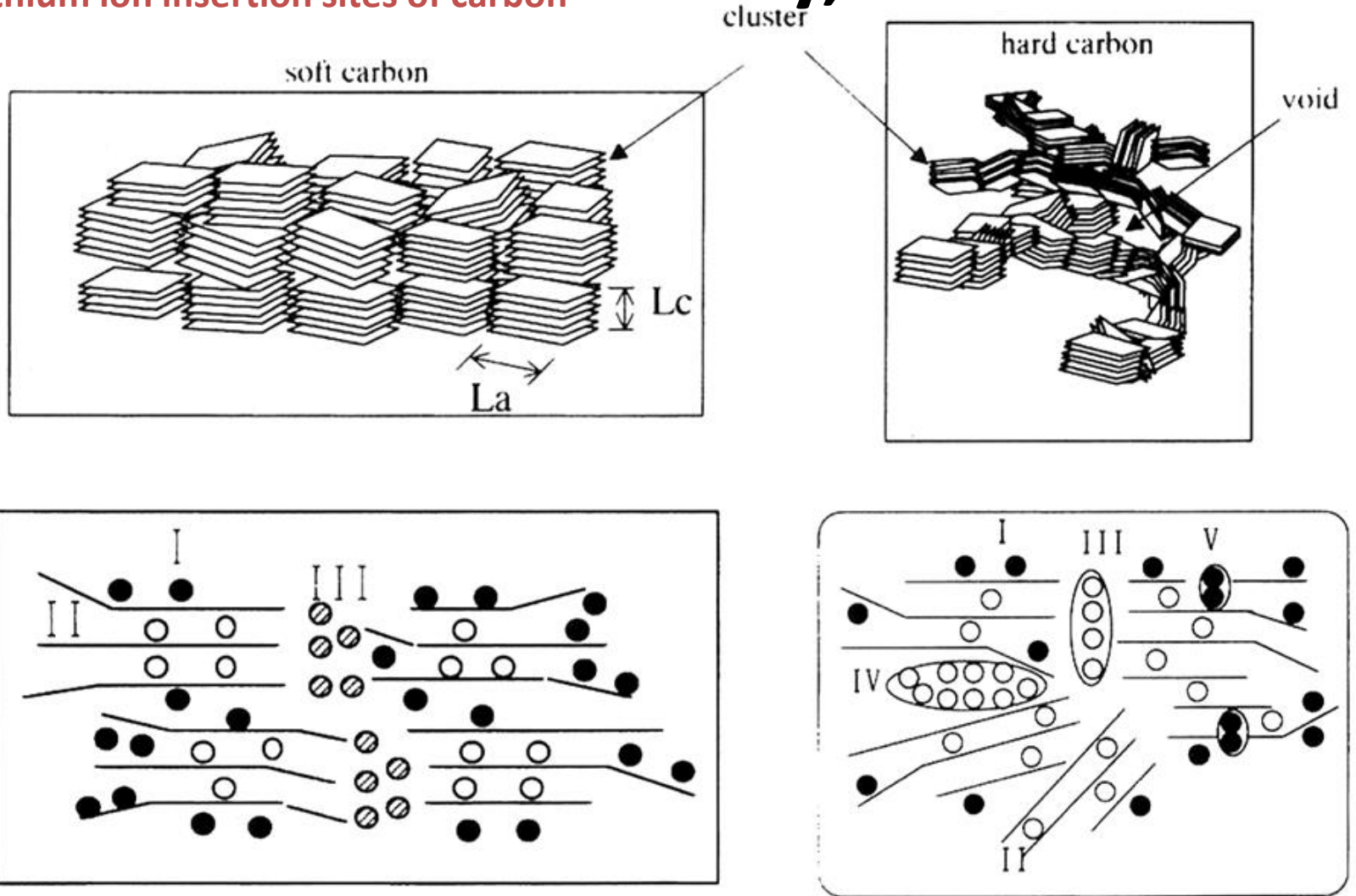
Mochida model

## Charge-Discharge Profile



# Lithium Ion Battery, Electrode

Lithium ion insertion sites of carbon



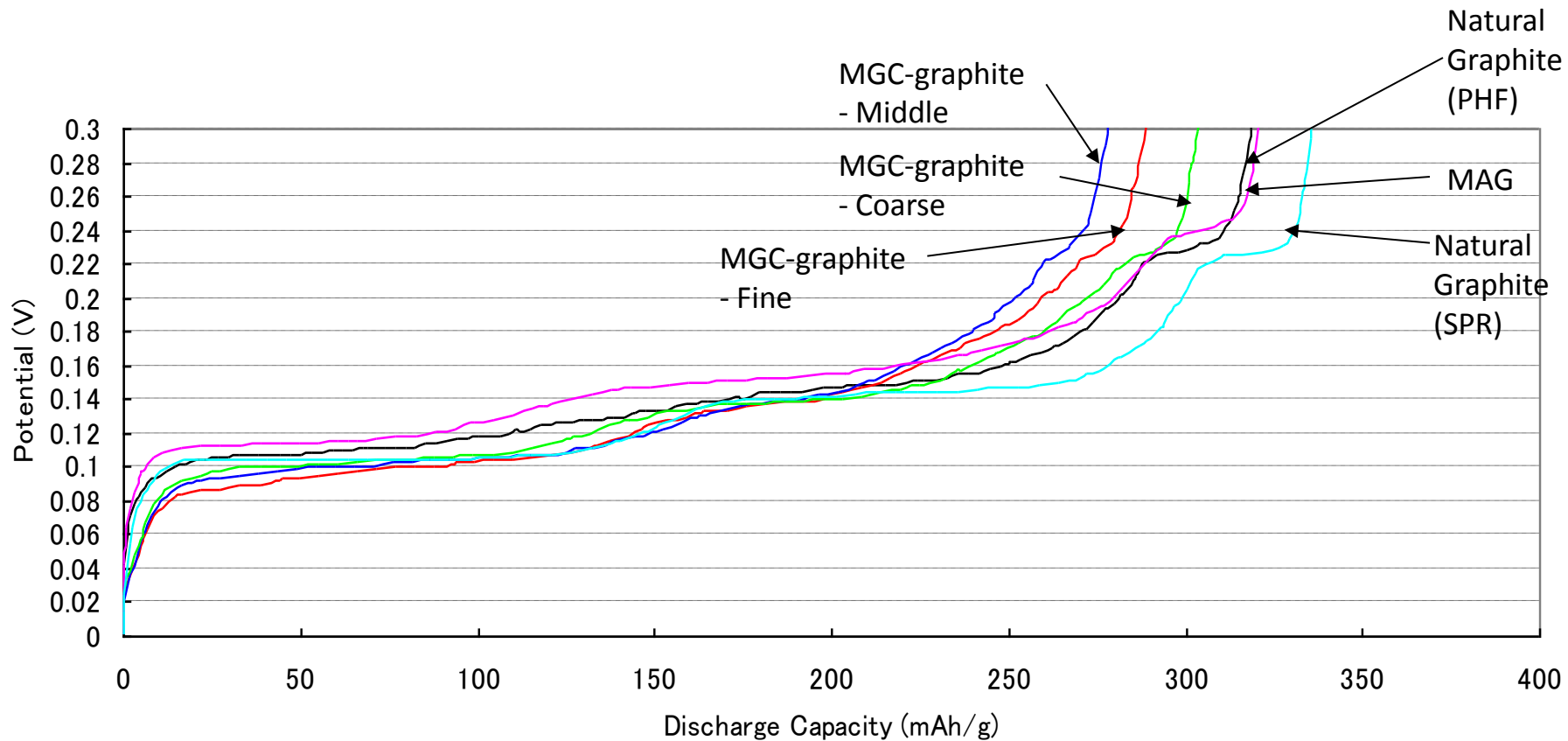


- Larger Capacity : Energy Density
- Larger Rate : Power Density
- Safety : Stable Forms of Reduced Alkaline Ion
- Lowest Unusable Ion

→ Complete Electrode Material

Carbon Is Always A Key Material.

# Typical Properties of Synthetic Graphites



# GraphiteとGraphene

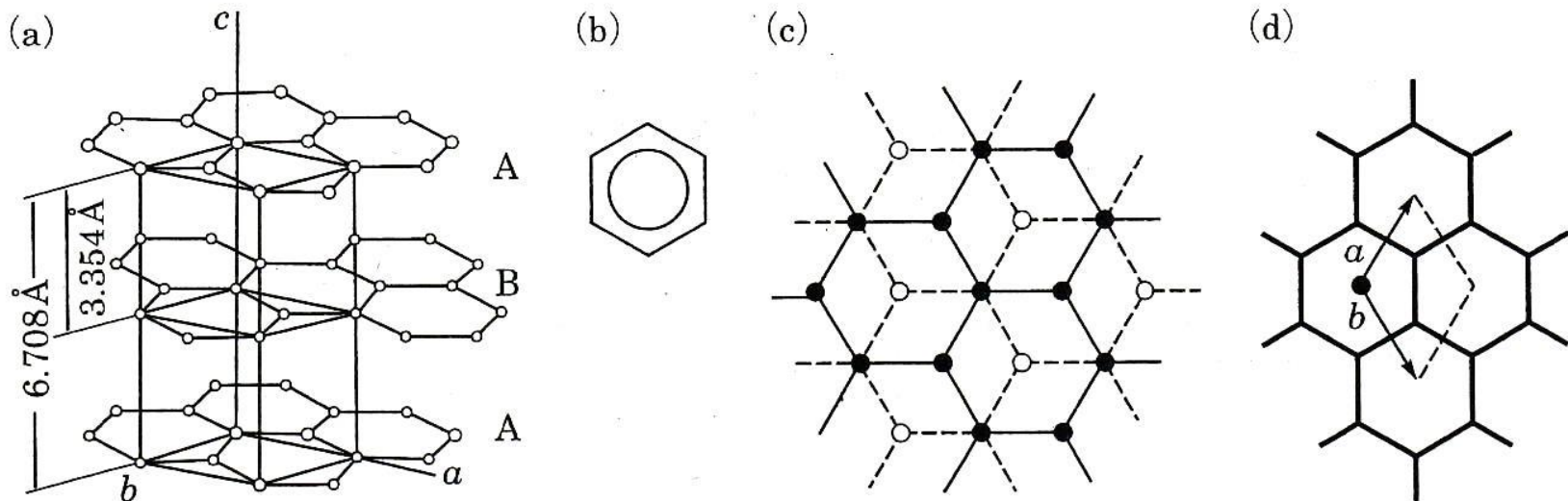
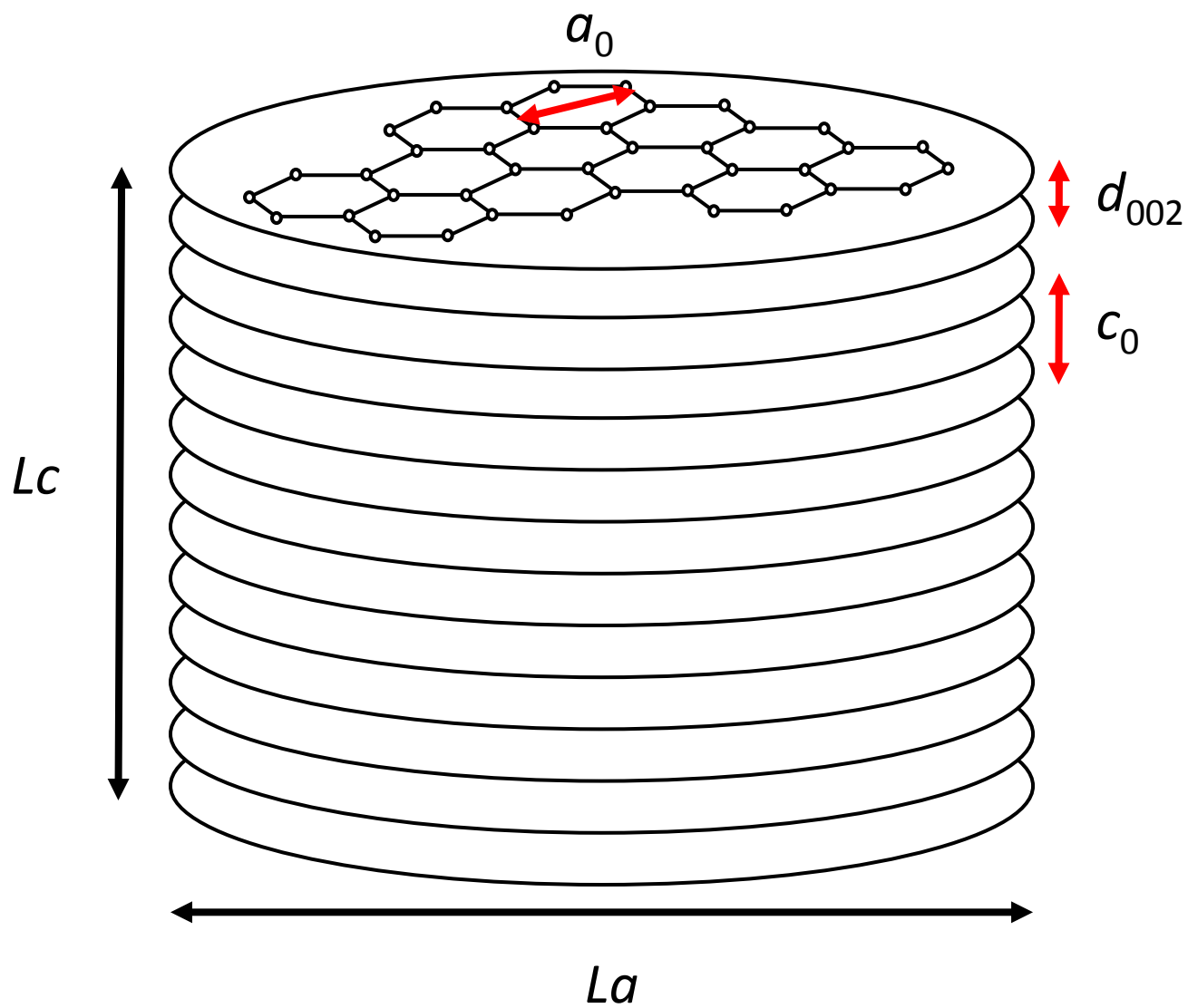


図4.2 グラファイトの構造

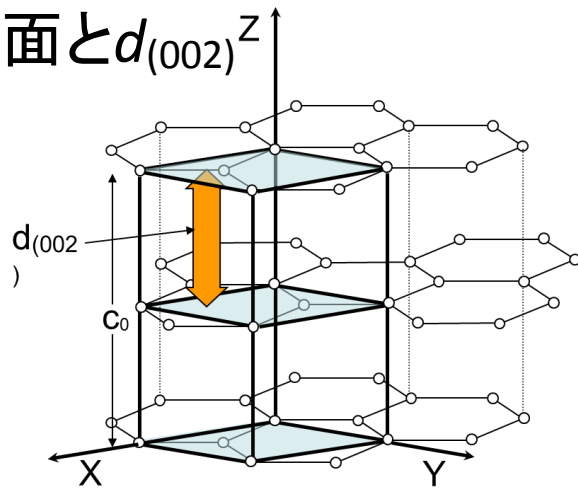
(a) グラファイトの結晶構造. ABA 様式で積層している. (b) ベンゼン. (c)  $c$  軸方向の上方から2枚面の重なりを見たもの. ●が上の面の炭素原子で, ○が下の面の炭素原子を表す. (c)  $ab$  面の構成. 単一の  $ab$  面 (グラフェン面) だけなら, 単位セルは菱形の部分で, 2個の炭素原子を含む.  $ab$  面の並進長さは,  $|a|=|b|=2.462 \text{ \AA}$ .

# 炭素の結晶構造パラメータ

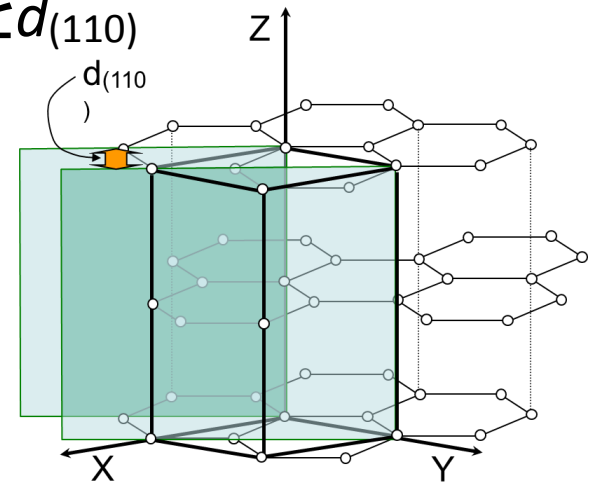


# 結晶面と面間隔の関係

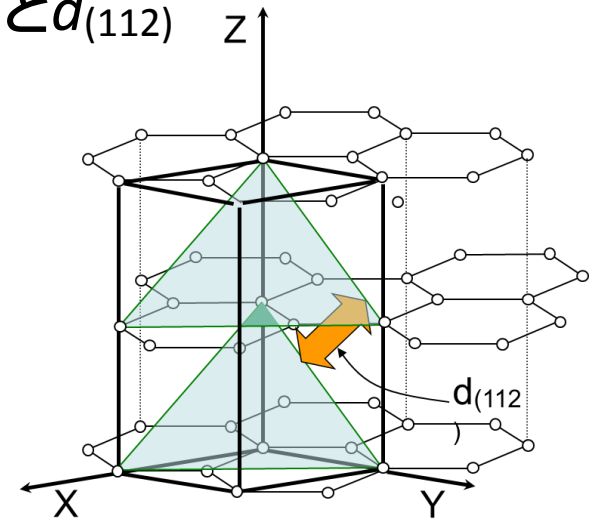
(002)面と $d_{(002)}$



(110)面と $d_{(110)}$



(112)面と $d_{(112)}$



# Graphiteの構造

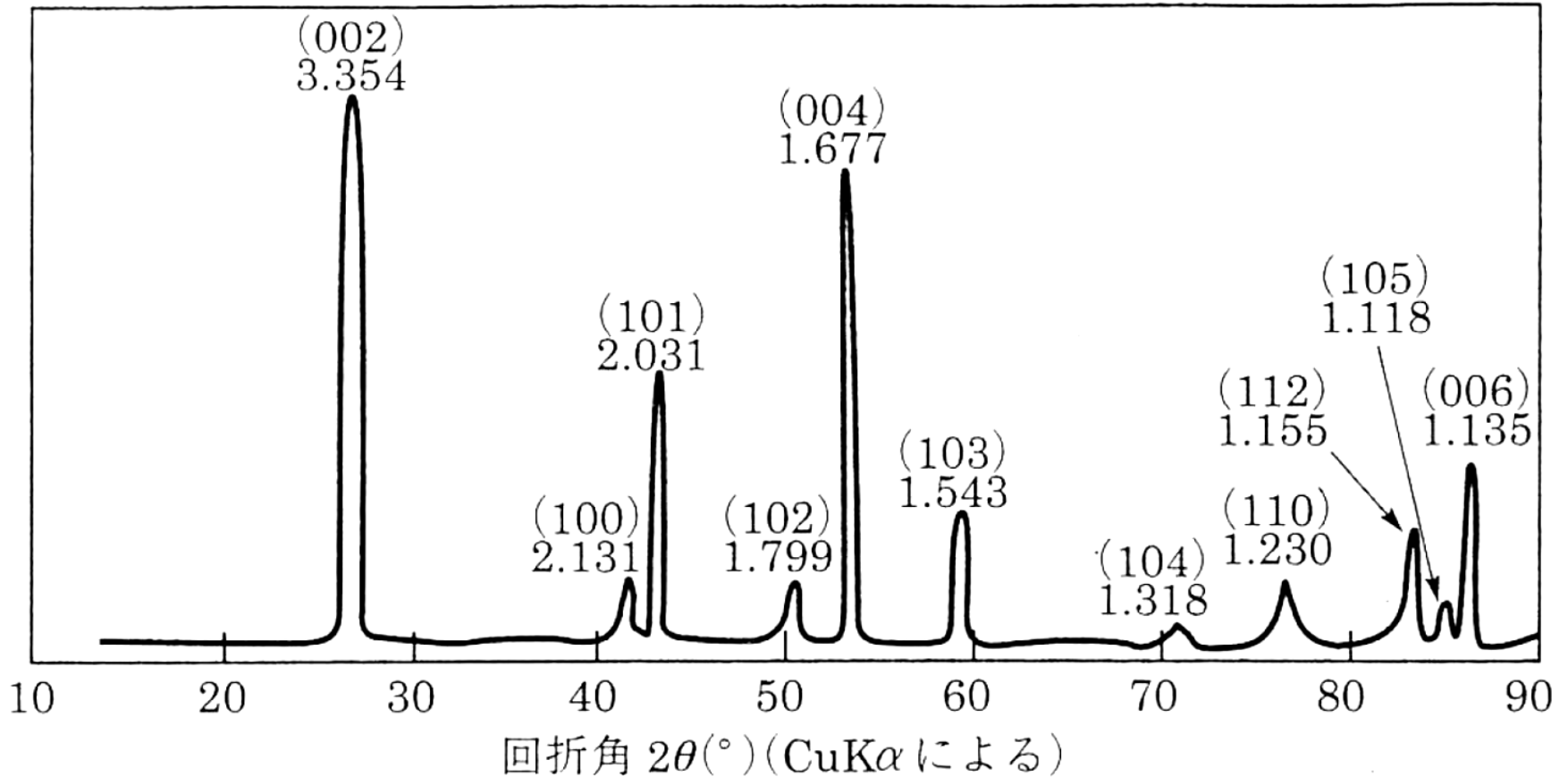
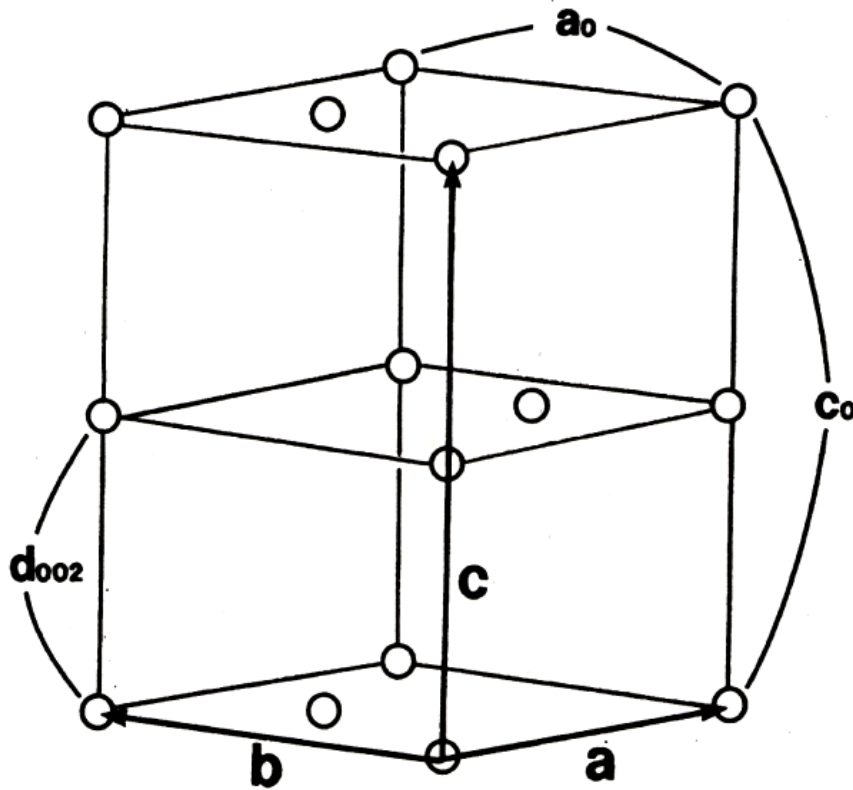


図4.4 グラファイトのX線回折パターン

# 黒鉛の電子構造

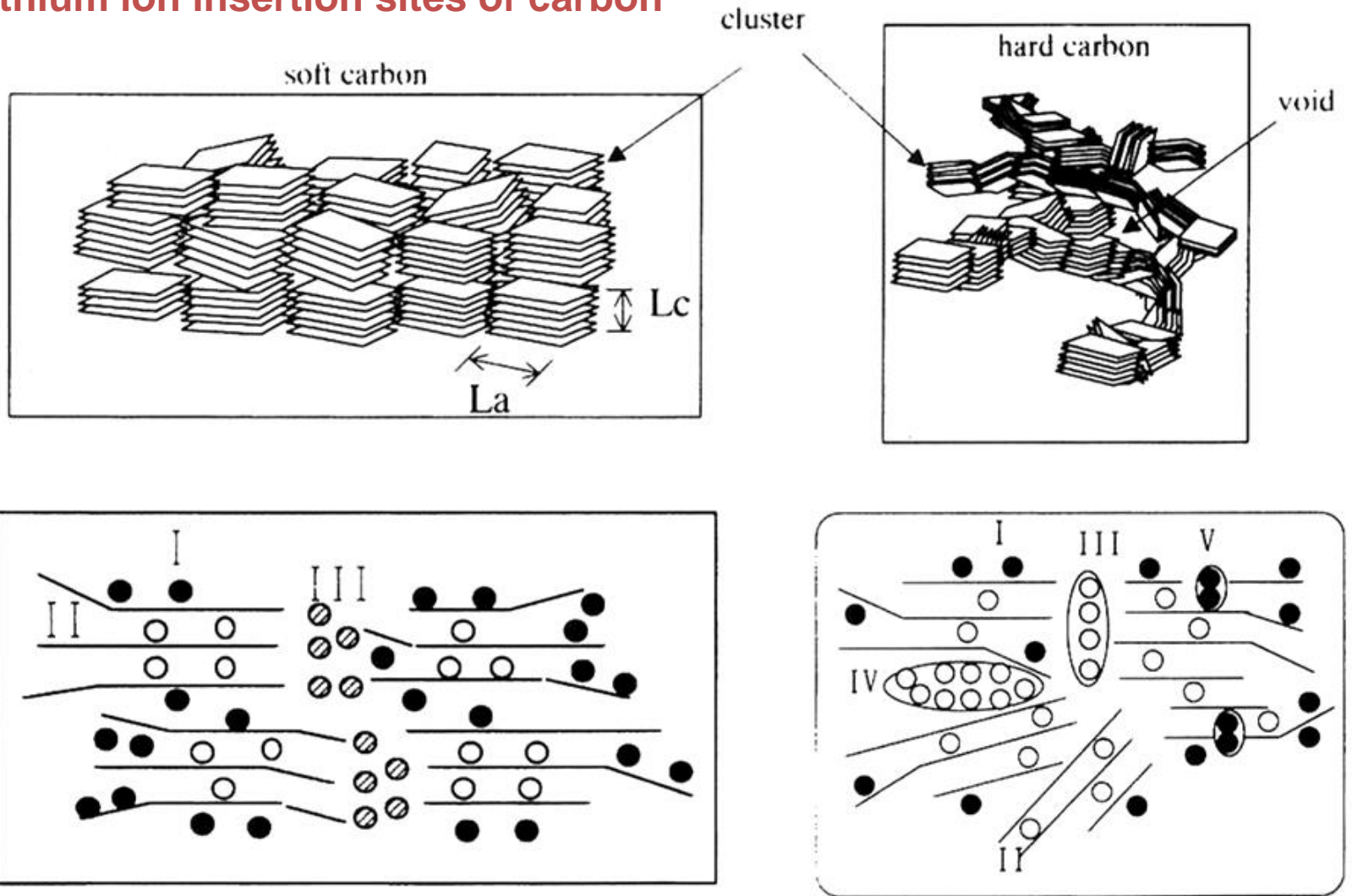


最隣接原子間距離: 0.1421nm  
 第2隣接原子間距離: 0.2461nm  
 層間距離: 0.3354nm

黒鉛結晶の単位格子と格子定数 $a_0, c_0$ および基本格子ベクトル $a, b, c$

# Lithium Ion Battery, Electrode

Lithium ion insertion sites of carbon





# Graphiteの反応性

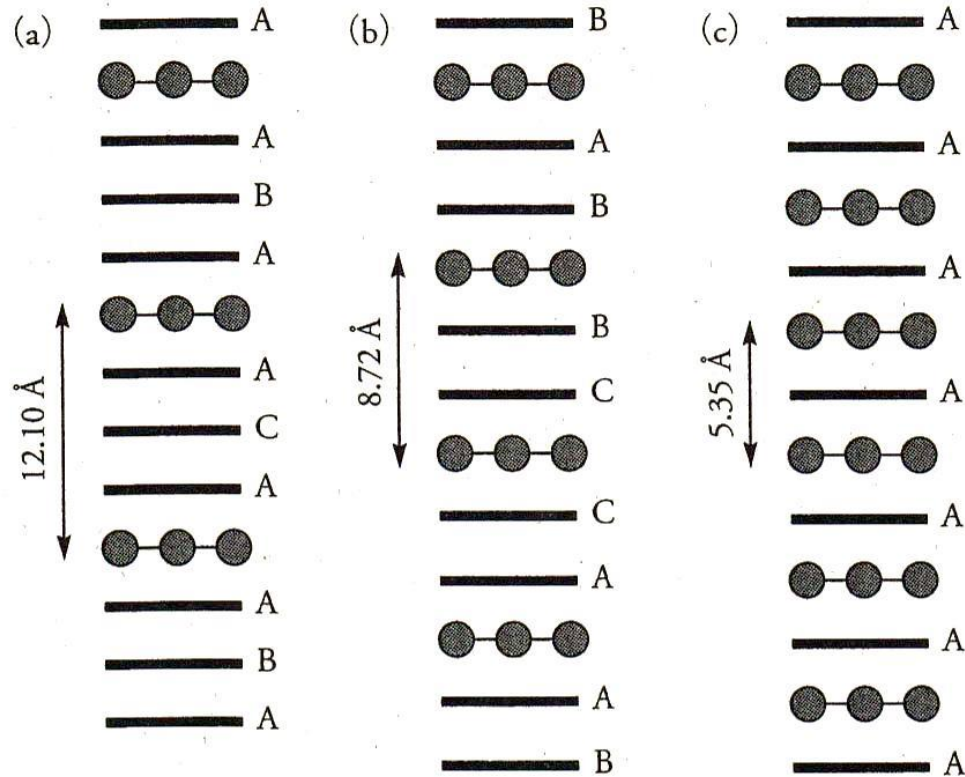
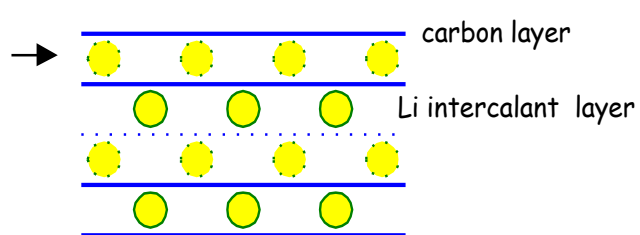


図4.6 カリウムをインターカレートとした GIC のステージ構造

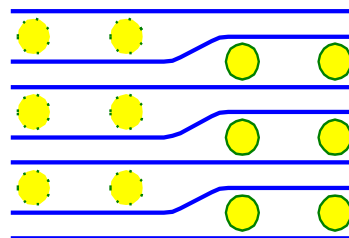
実線はグラファイトの  $ab$  面で、A, B, C はその積層様式. 灰色の丸印はカリウム原子.

(a)  $\text{KC}_{36}$  (ステージ 3), (b)  $\text{KC}_{24}$  (ステージ 2), (c)  $\text{KC}_8$  (ステージ 1).

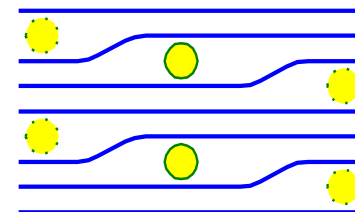
GICでは、HOバンドの頂上から電子が引き抜かれることによって正孔が注入されたり、LUバンドの底に余剰電子が与えられたりするので、フェルミ状態の密度が増加して導電性が上がる。



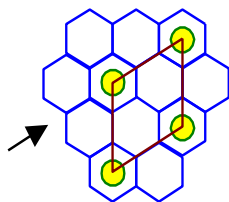
$\text{LiC}_6, \text{LiC}_9$  - stage 1  
: A type of superlattice

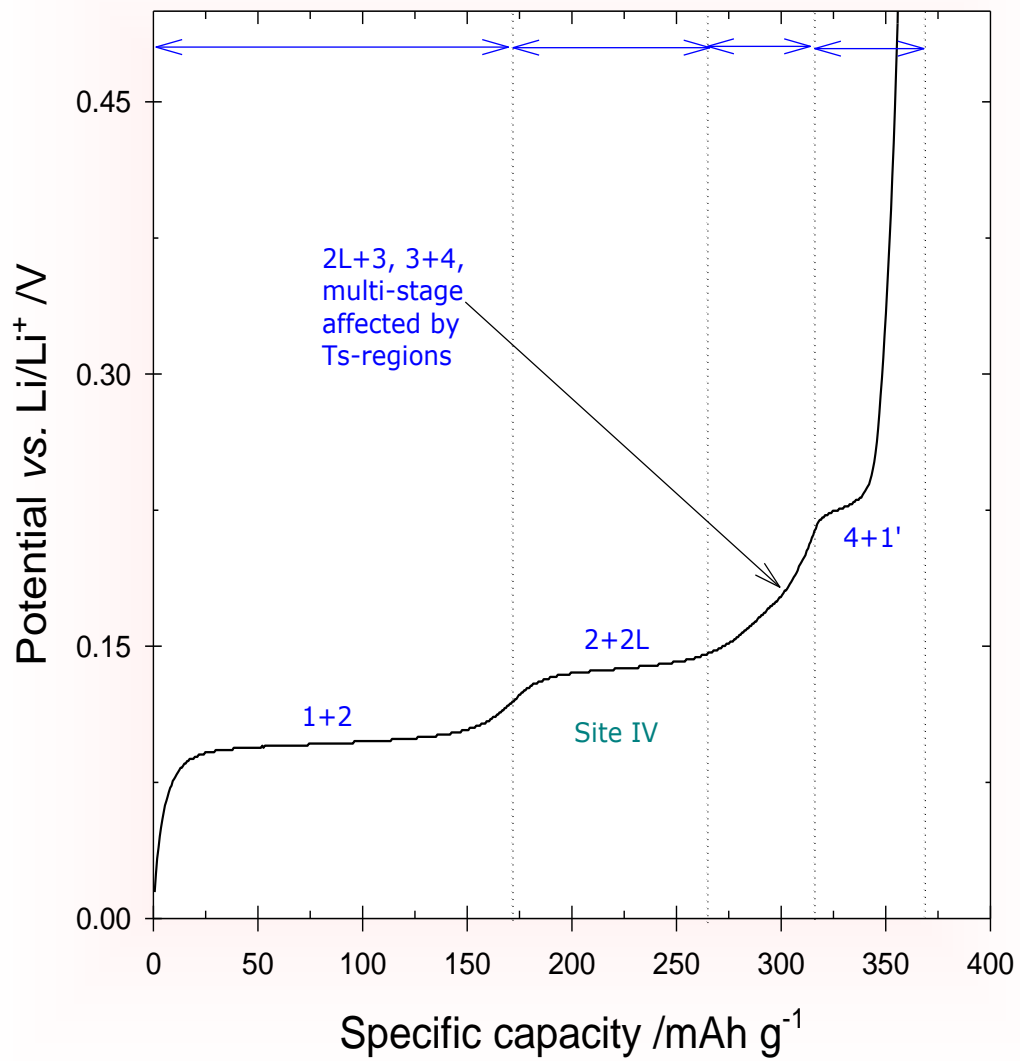


$\text{LiC}_{12}$  - stage 2



$\text{LiC}_{18}$  - stage 3





# Voltammetric behavior of graphite

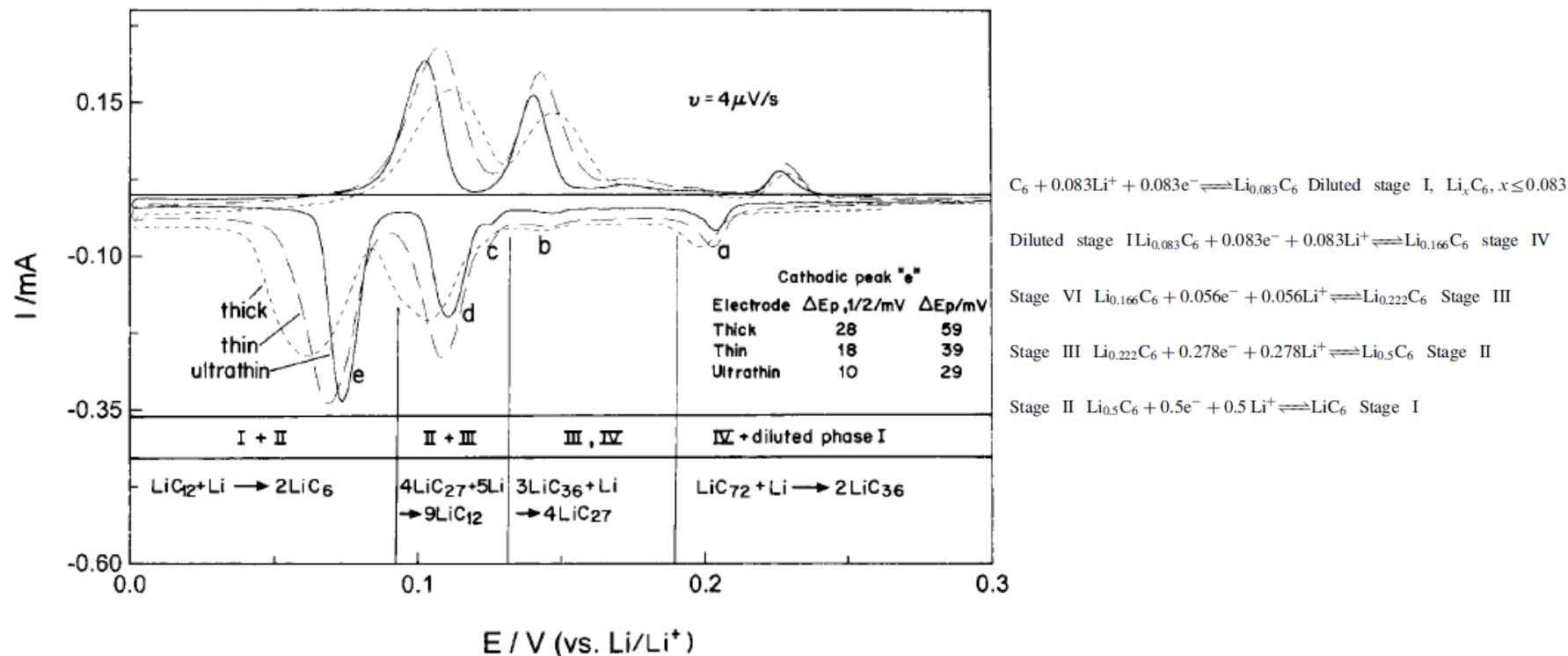
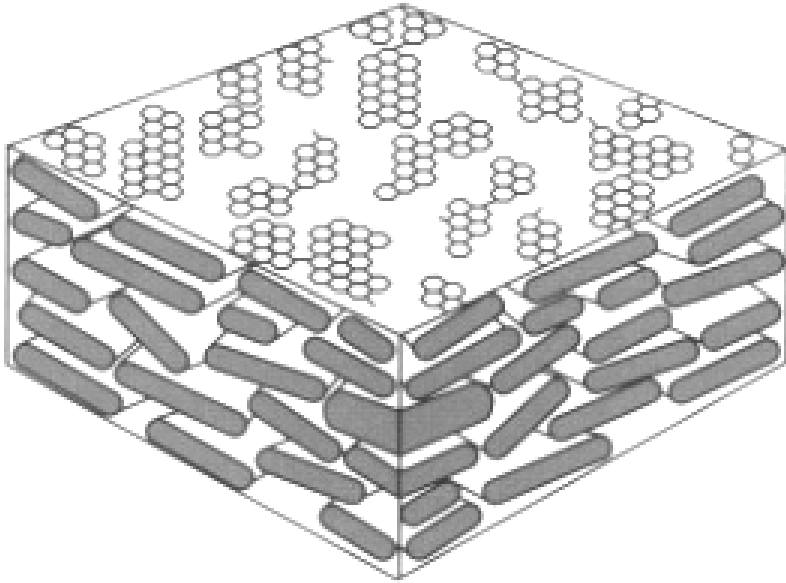


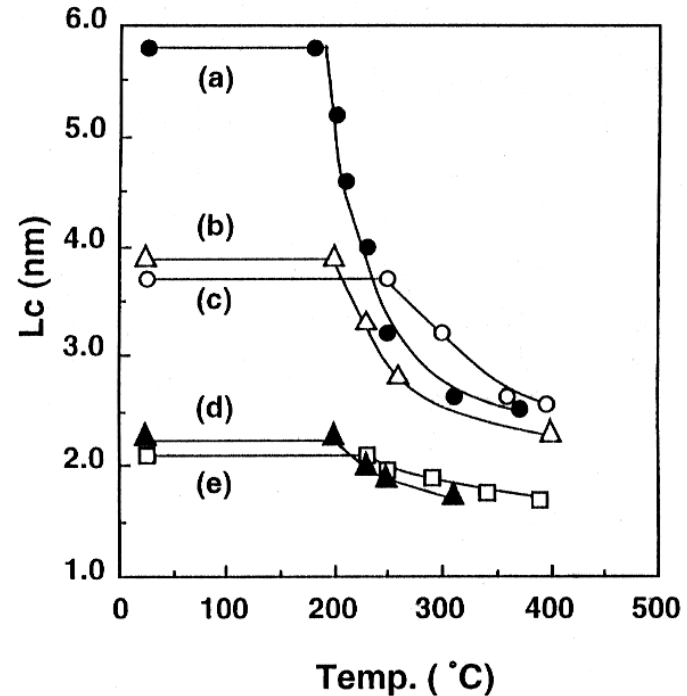
Fig. 1. Typical voltammetric behavior of graphite electrodes at a very slow scan rate,  $v = 4 \mu\text{V/s}$  in EC-DMC 1:3/LiAsF<sub>6</sub> solution. (Graphite flakes, KS-6, Timcal). Ultrathin, thin and thick electrodes correspond to submicronic (0.16 mg/cm<sup>2</sup>, see preparation procedure in Refs. [5,9]), 10 microns (3.3 mg/cm<sup>2</sup>), and 140 μm (12 mg/cm<sup>2</sup>), respectively. The current for the thin and ultrathin electrodes was normalized with that of the thick electrodes according to their mass ratios in order to have the 3 CV on the same scale. The electrode area was 2.4 cm<sup>2</sup> for the thin and the ultrathin, and  $\approx 0.2$  cm<sup>2</sup> for the thick electrode.

# Design and Its Thermal Change of Aromatic Stacking



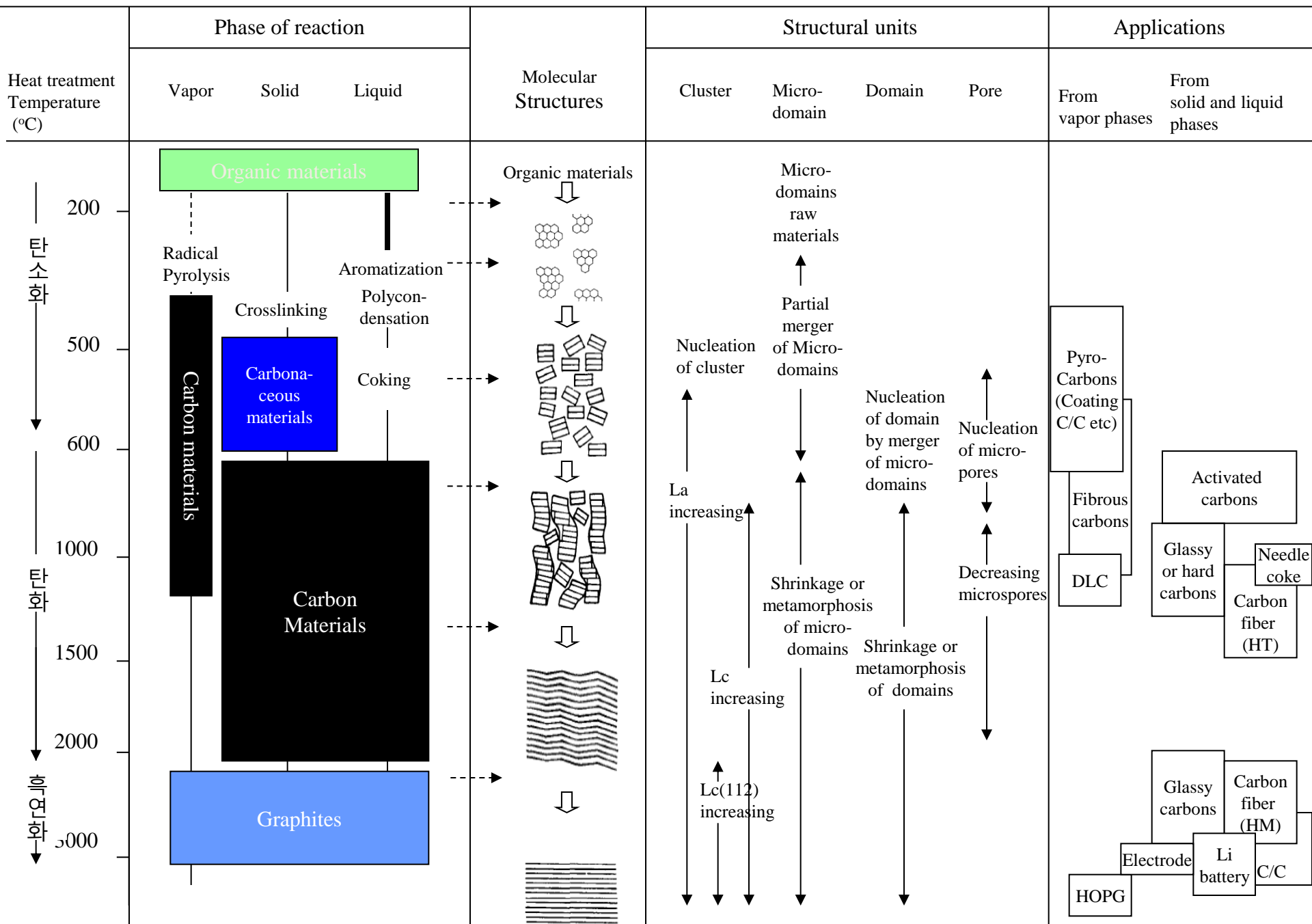
Molecular Models

Spider Wedge Stacking of mesophase pitch  
(Zimmer *et al. Advances in Liquid Crystal, New York, 1982, 5*)



Melt-XRD analysis

Change in Lc of mesophase pitch at higher temperature; (a) methylnaphthalene-derived pitch; (b) petroleum-derived mesophase pitch; (c); coal tar derived-mesophase pitch; (d) naphthalene-derived mesophase pitch; (e) anthracene-derived mesophase pitch  
(Korai *et al. Carbon, 1992, 30, 1019*)



Heat treatment  
Temperature (°C)

Phase of reaction  
Vapor    Solid    Liquid

Gas  
volatilization

Chemical and  
Physical changes

Molecular  
Structures

탄소화  
↓

탄화  
↓

흑연화  
↓

Organic materials

Radical  
Pyrolysis

Crosslinking

Carbon materials

Carbonaceous materials

Carbon Materials

Graphites

Aromatization

Polycondensation

Coking

H<sub>2</sub>O  
Low mol. Paraffin or Olefins

Low mol. Aromatic carbons

CH<sub>4</sub>, CO, NO<sub>2</sub>  
H<sub>2</sub>S, CO<sub>2</sub>  
H<sub>2</sub> etc.

H<sub>2</sub>  
CO, CO<sub>2</sub>  
H<sub>2</sub>S  
etc.

H<sub>2</sub>S  
HCN  
CS<sub>2</sub>  
N<sub>2</sub> etc.

H<sub>2</sub>  
H<sub>2</sub>S  
N<sub>2</sub> etc.

Main chain rearrangements  
Aromatization, Condensation  
Polymerization, Cross-linking  
Coking

Devolatilization  
Crack nucleation  
Stacking start  
Loss of viscosity (Inorganic Mat.)

Removal of heterogeneous atoms  
Dehydrogenation  
Micropore nucleation  
L<sub>a</sub> increasing

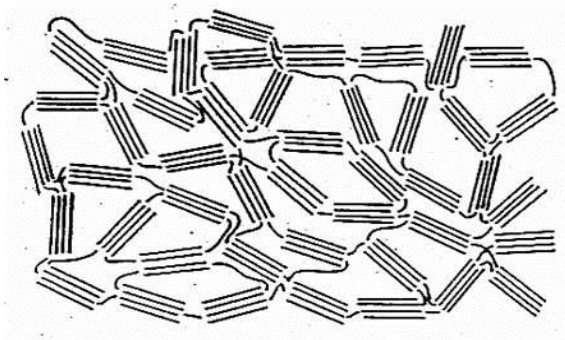
Removal of heterogeneous atoms  
L<sub>c</sub> increasing  
Reducing micro pores

Removal of inorganic materials  
Formation of 3 D graphitic structure

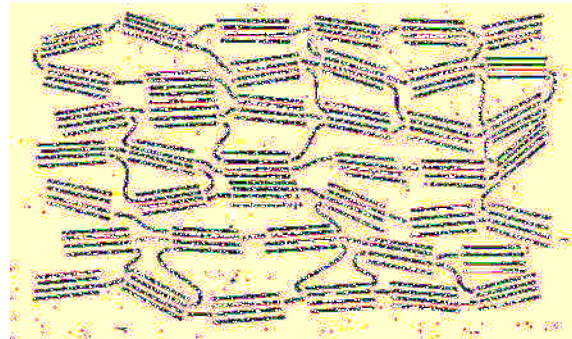
Organic materials



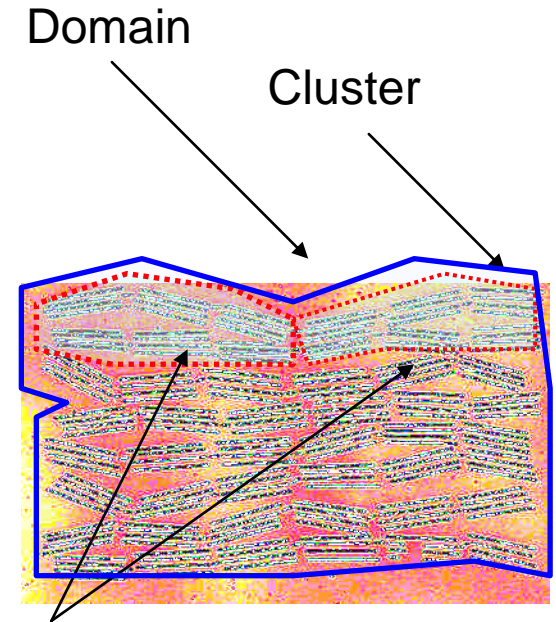
# Franklin's Models of Carbon Structures



(a) Non-Graphitizing (Isotopic)



(b) Partially Graphitizing

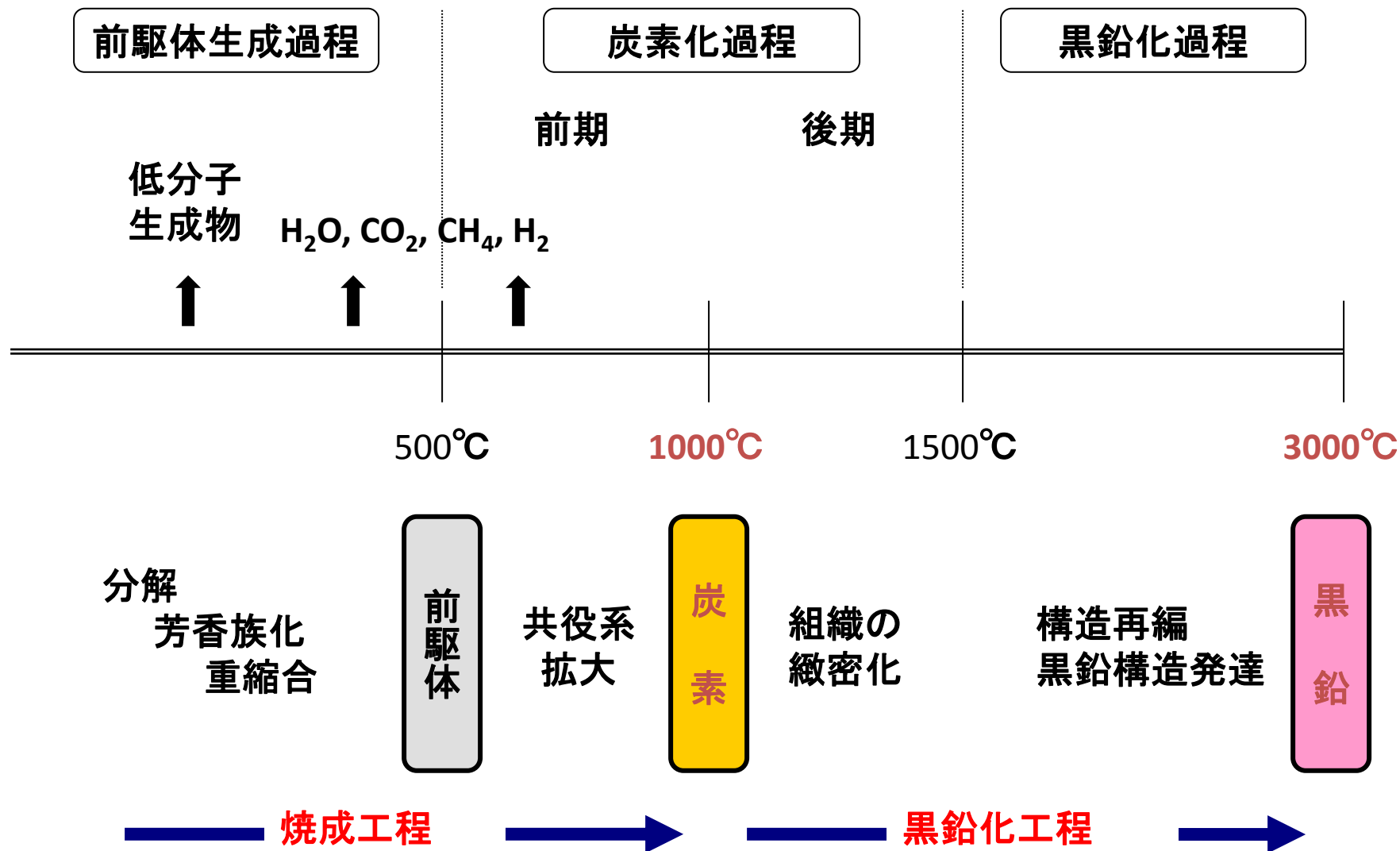


Microdomain

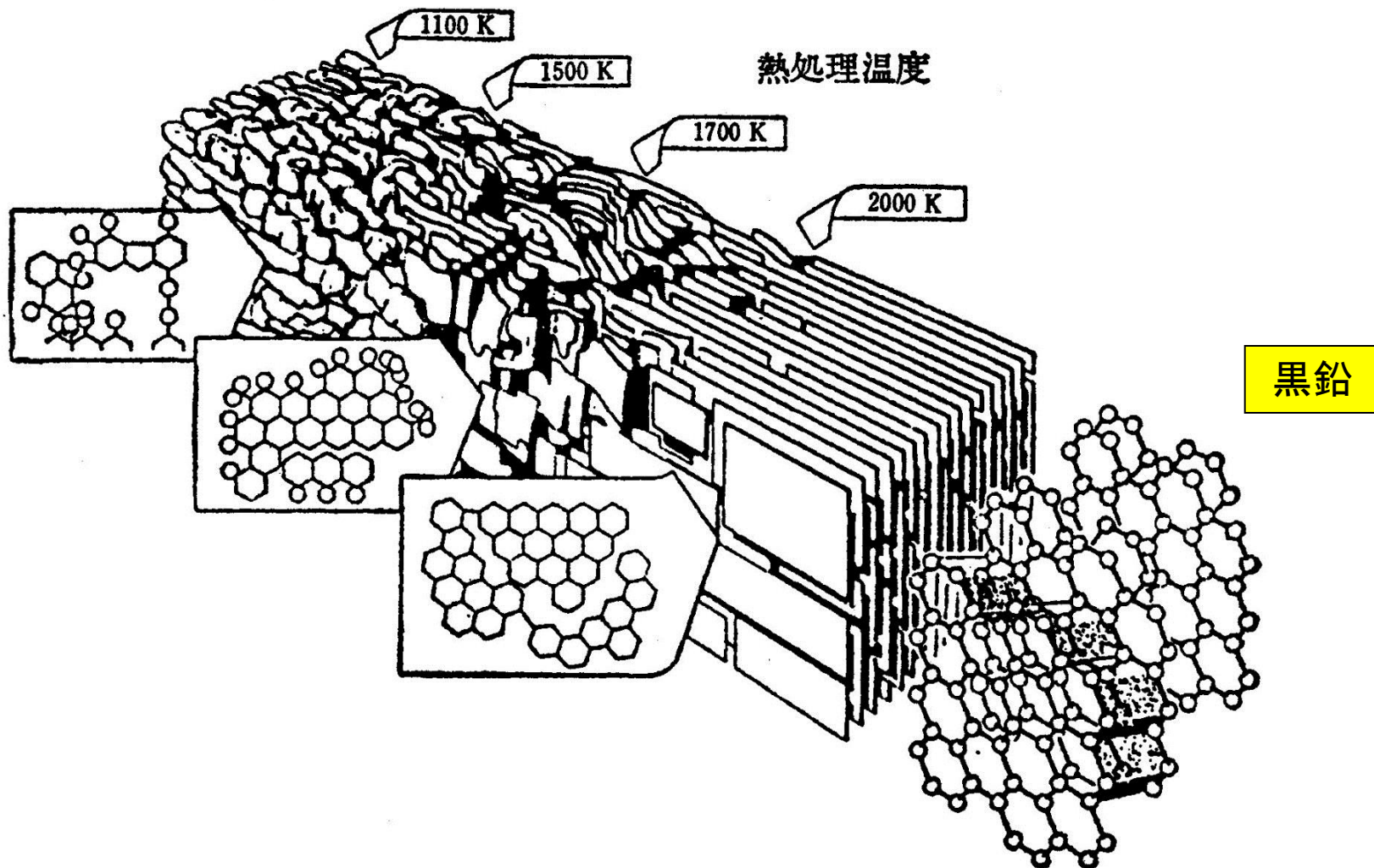
(c) Graphitizing



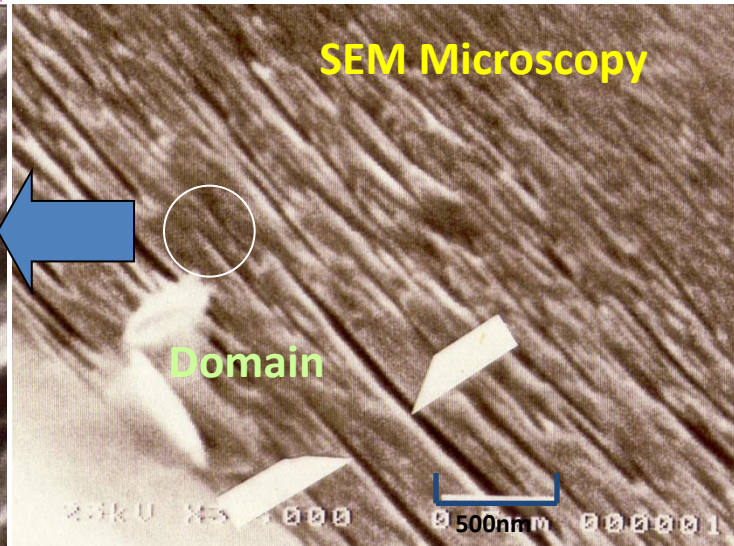
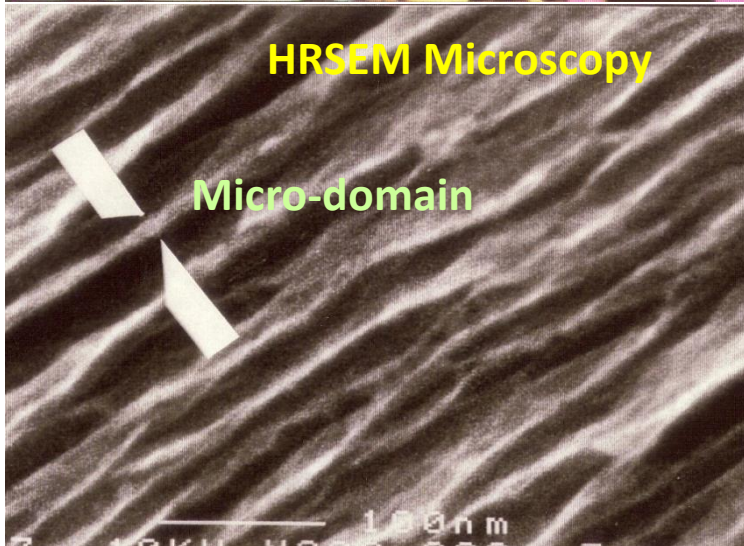
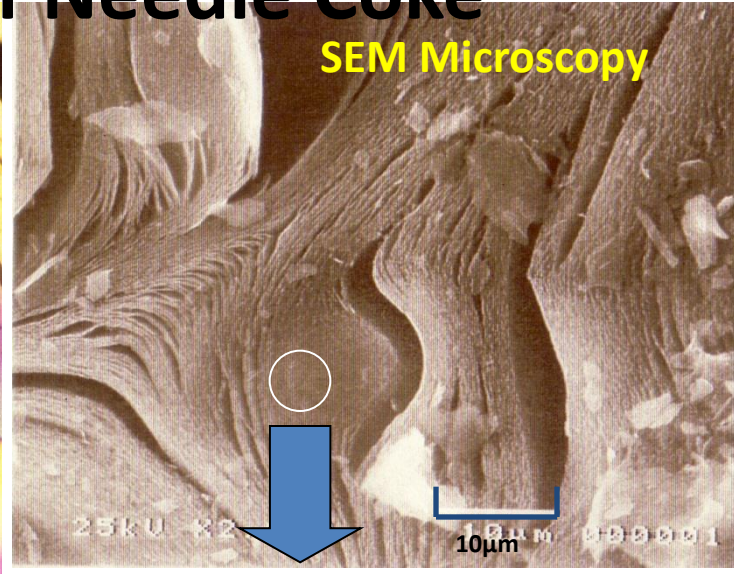
# 有機物の加熱による変化



# 熱処理温度による結晶構造変化



# Structure of Needle Coke

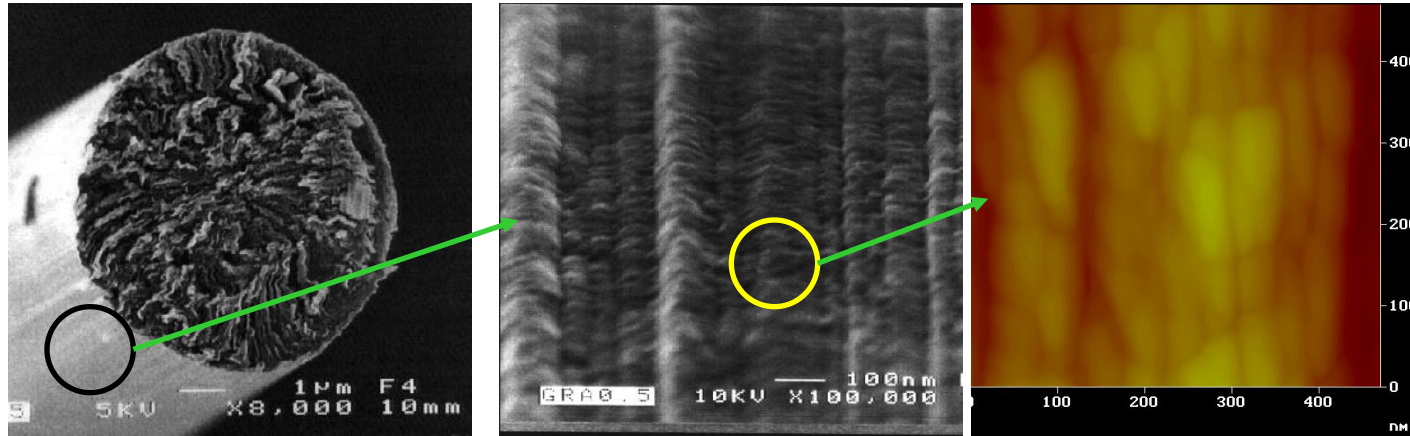




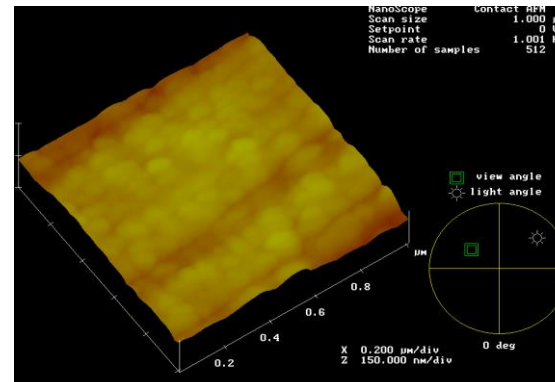
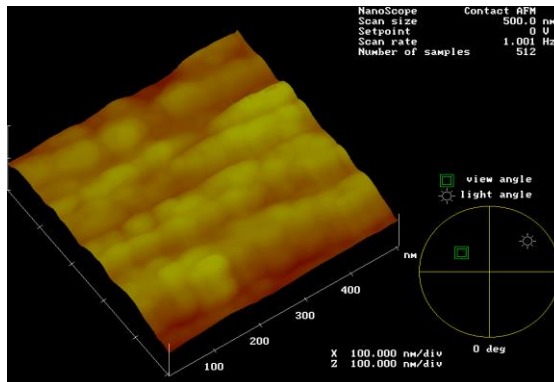
# Nanoscopic Structure of Mesophase Pitch Based Carbon Fiber

**Problem: Low Compressive Strength > Restriction of CFRP Application**

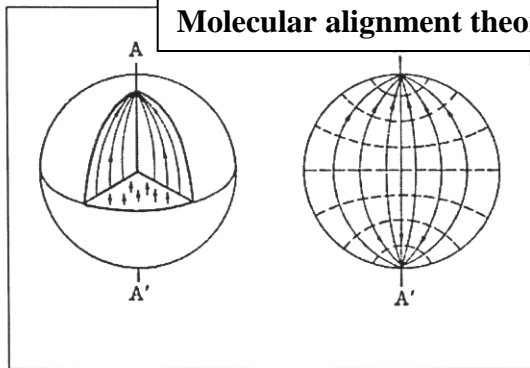
**Factor: Size and Distribution of Micro-domain**



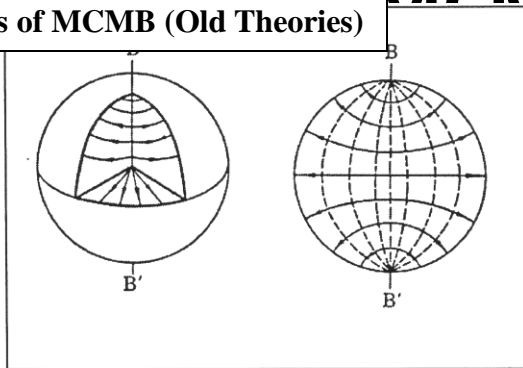
**Pleat Structure > Homogeneous / Small > Increasing Compressive Strength**



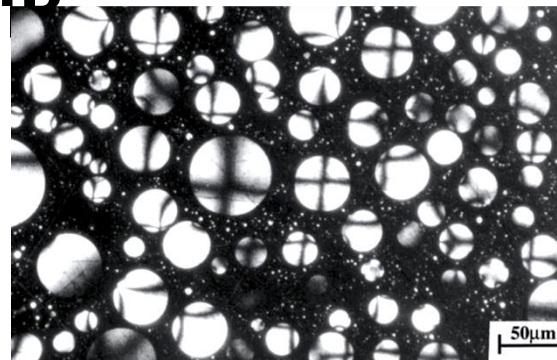
**Molecular alignment theories of MCMB (Old Theories)**



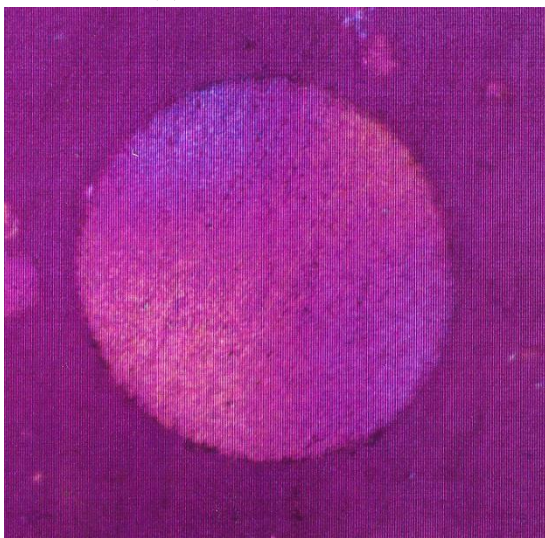
(a) Brooks-Taylor Type



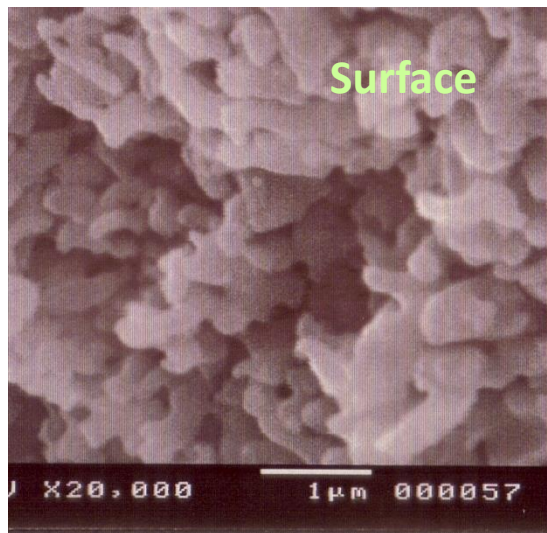
(b) Honda Type



**Optical Micrograph of MCMB in Isotropic Matrix**



Optical Micrograph of PI of AR pitch derived MCMB



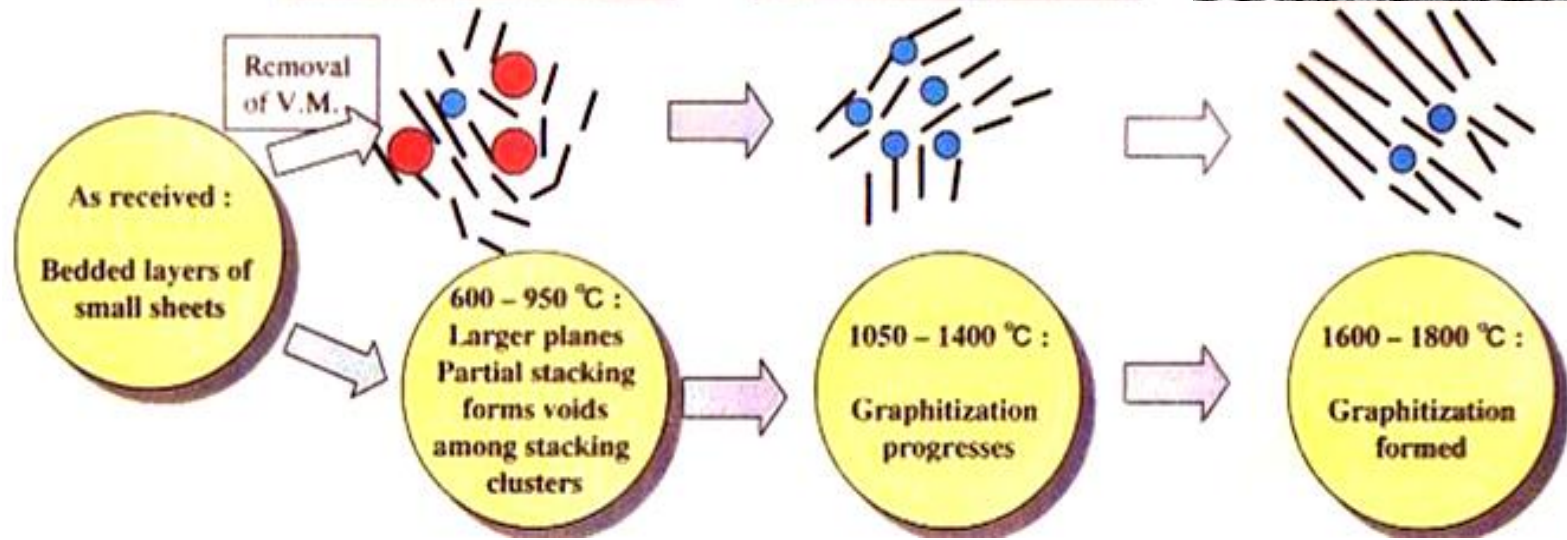
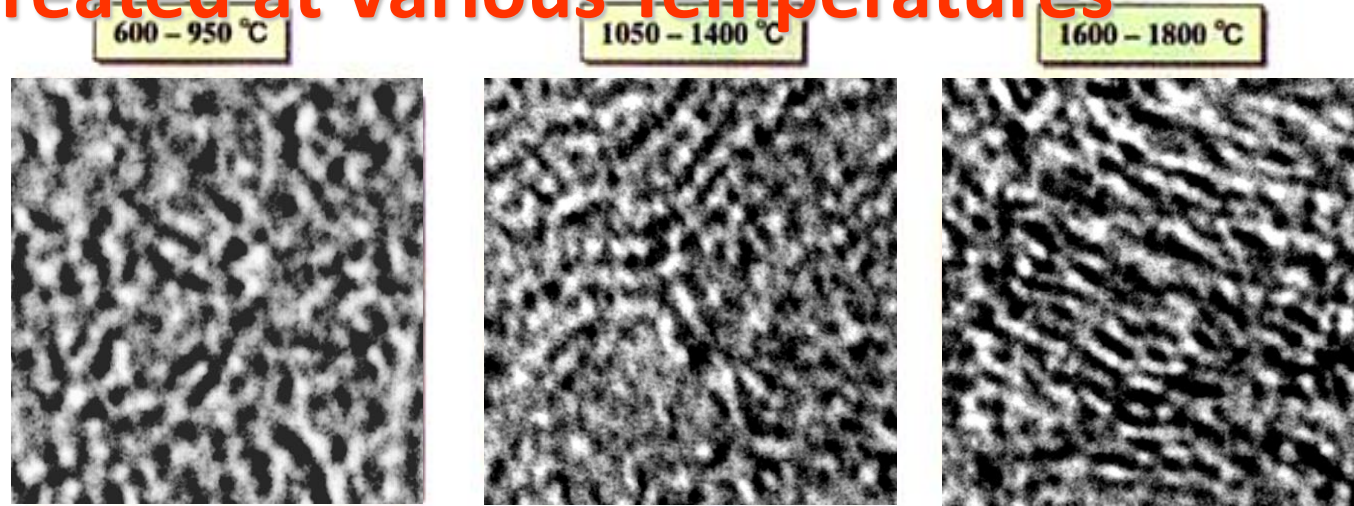
SEM Photograph of PI of AR pitch derived MCMB



SEM Photograph of PI of AR pitch derived MCMB



# TEM Images of Hongye Anthracites Heat Treated at Various Temperatures



# Structure of Activated Carbon

- Surface Area, Pore: Depth Volume

  - Surface Structure

  - Surface Chemistry

  - Based and Edge Plane, Substituents

  - Hetero atoms in Hexagon

- Carbon Structure of Wall

  - Nano, Micro, Macro Structure of Carbon Wall

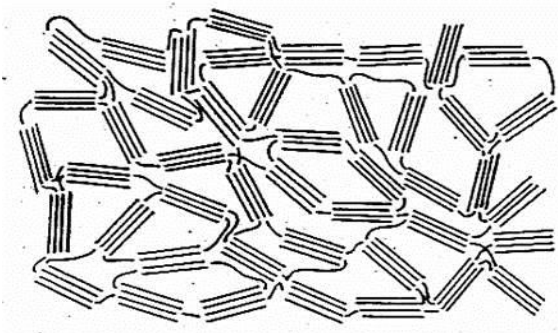
  - Graphitization Extent

  - Domain Structure

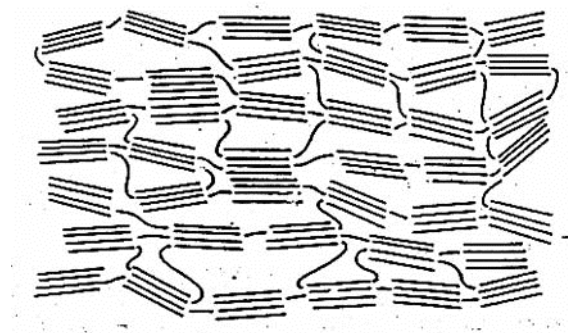
    - Density, Reactivity (Activated Surface)

    - Precursor : Structure and Reactivity

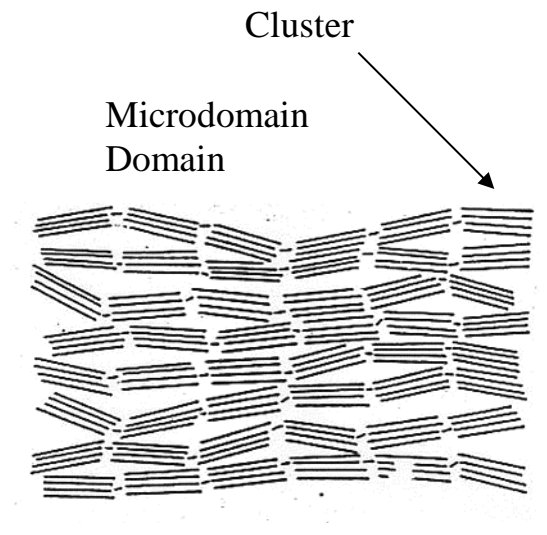
# Franklin's Models of Carbon Structures



(a) Non-Graphitizing (Isotopic)



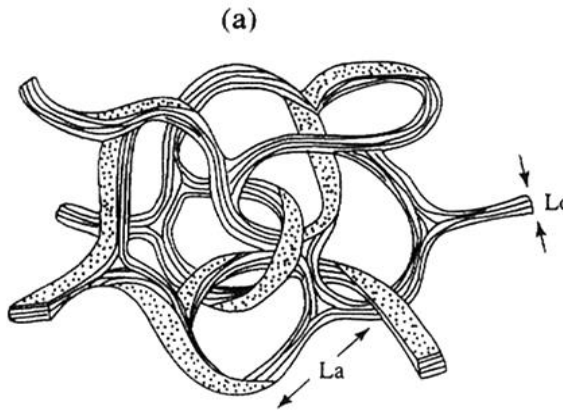
(b) Partially Graphitizing



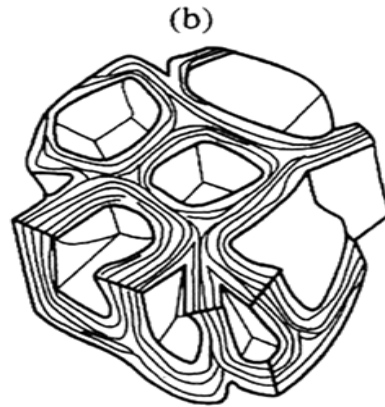
(c) Graphitizing



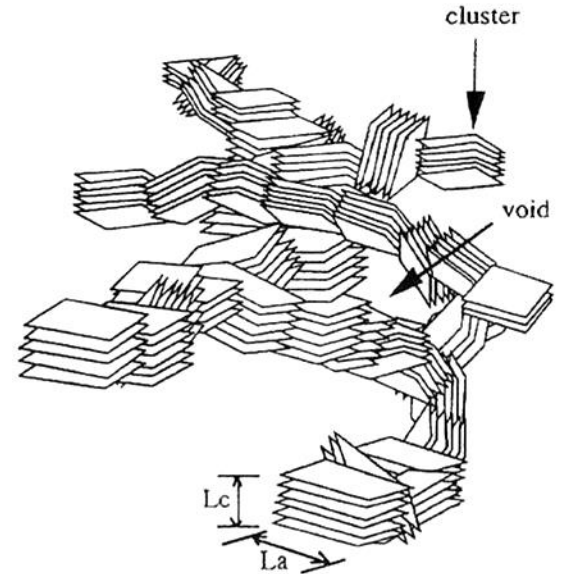
# Structural Models of Glassy Carbon Heated at High Temperature



La: Crystallite size  
Lc: Crystallite thickness  
**Jenkins-Kawamura model**

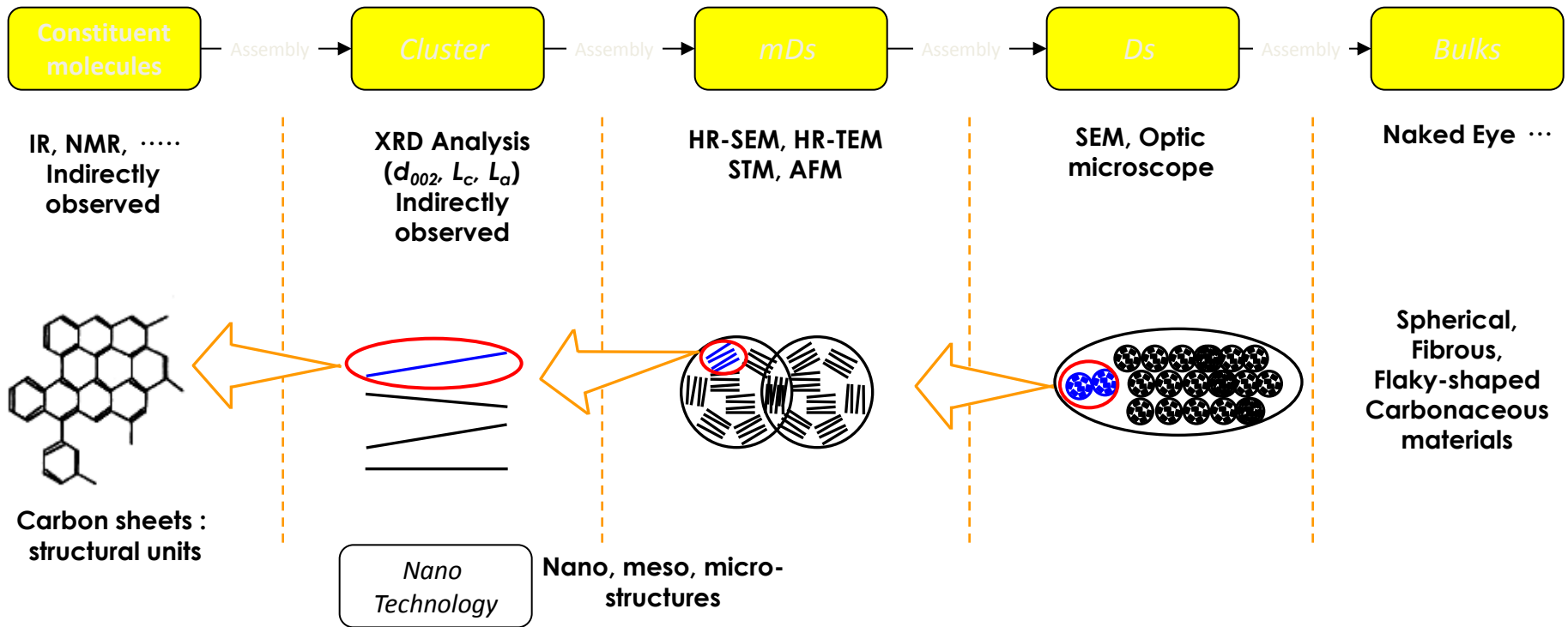


**Shiraishi model**



**Mochida model**

# Structural Hierarchy in Mesophase Pitch



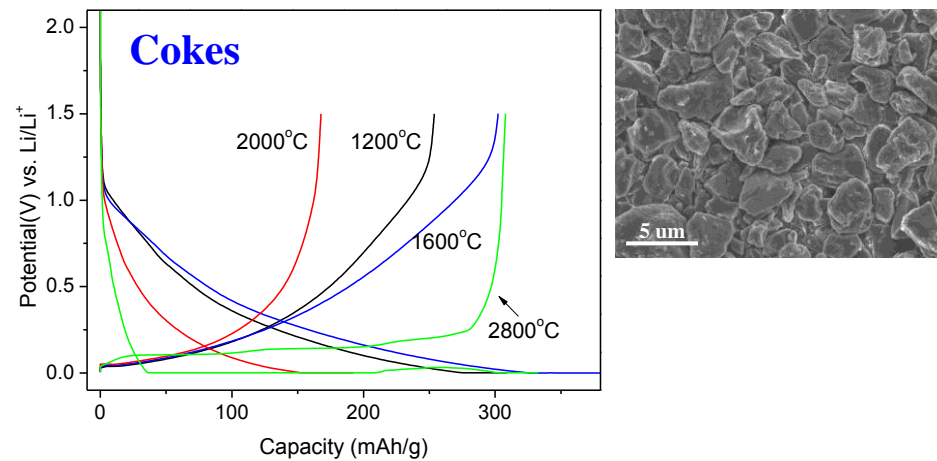
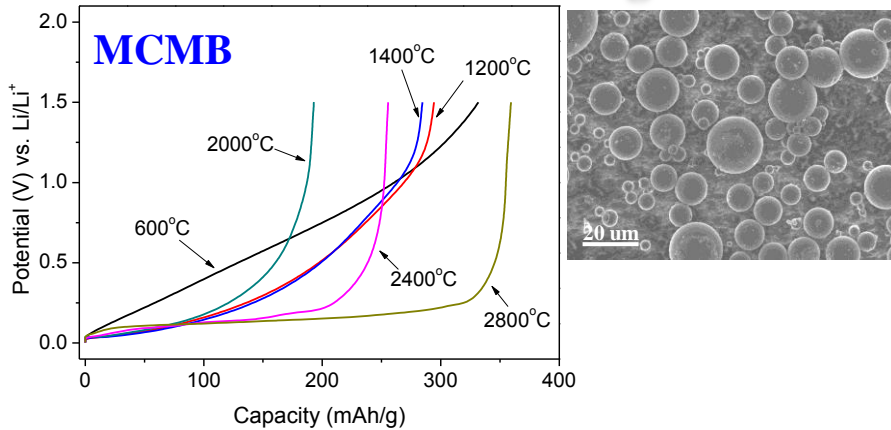
# MCMB

Lot		容量(mAh/g, 0~1.5V)			Dcap. (0~0.5V)	低電圧特性 (0.5V/1.5V)(%)
		1cy	2cy	3cy		
600°C 熱処理	ch	1497	440	368	124	38.5
	dis	396	342	321		
	効率(%)	26.5	77.6	87.2		
1200°C 熱処理	ch	393	308	303	197	67.1
	dis	303	299	294		
	効率(%)	77.2	96.9	97.1		
1400°C 熱処理	ch	359	295	289	198	69.7
	dis	291	288	284		
	効率(%)	81.1	97.6	98.3		
2000°C 熱処理	ch	227	198	194	159	83.0
	dis	196	193	191		
	効率(%)	86.1	97.6	98.4		
2400°C 熱処理	ch	298	262	259	238	93.2
	dis	258	258	256		
	効率(%)	86.6	98.5	98.8		
2800°C 熱処理	ch	426	364	360	344	96.5

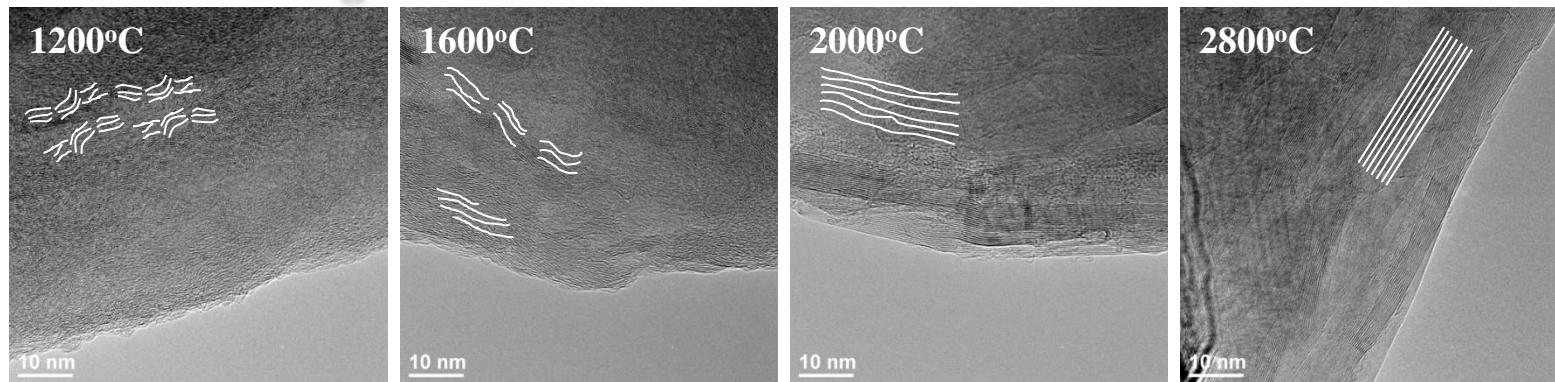
# Previous study of **Soft Carbon**

**Graphite has a limitation** at capacity and power density, such reason enforced to develop other carbon materials like soft carbon and hard carbon

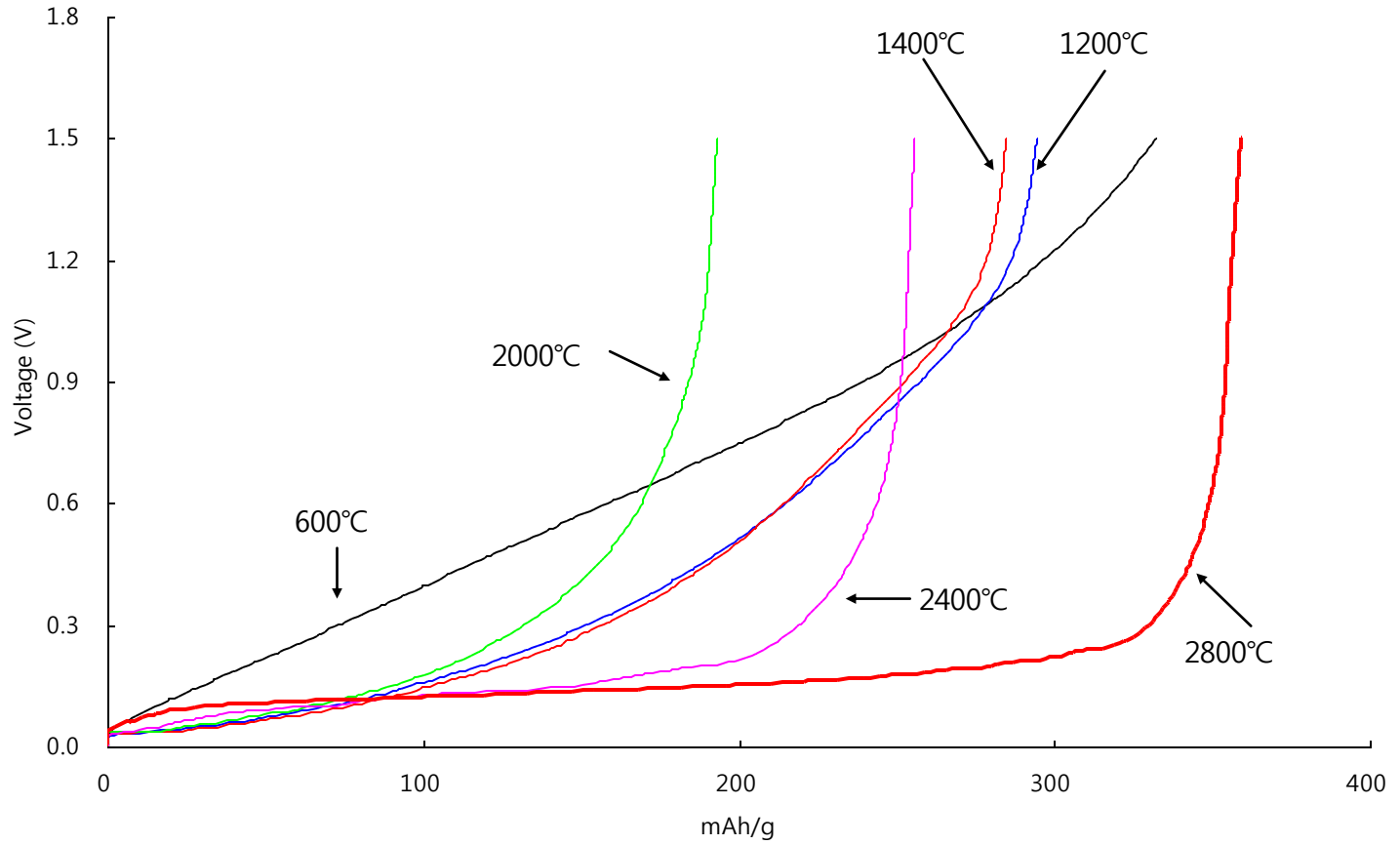
## Ch-Dis Profile & SEM image



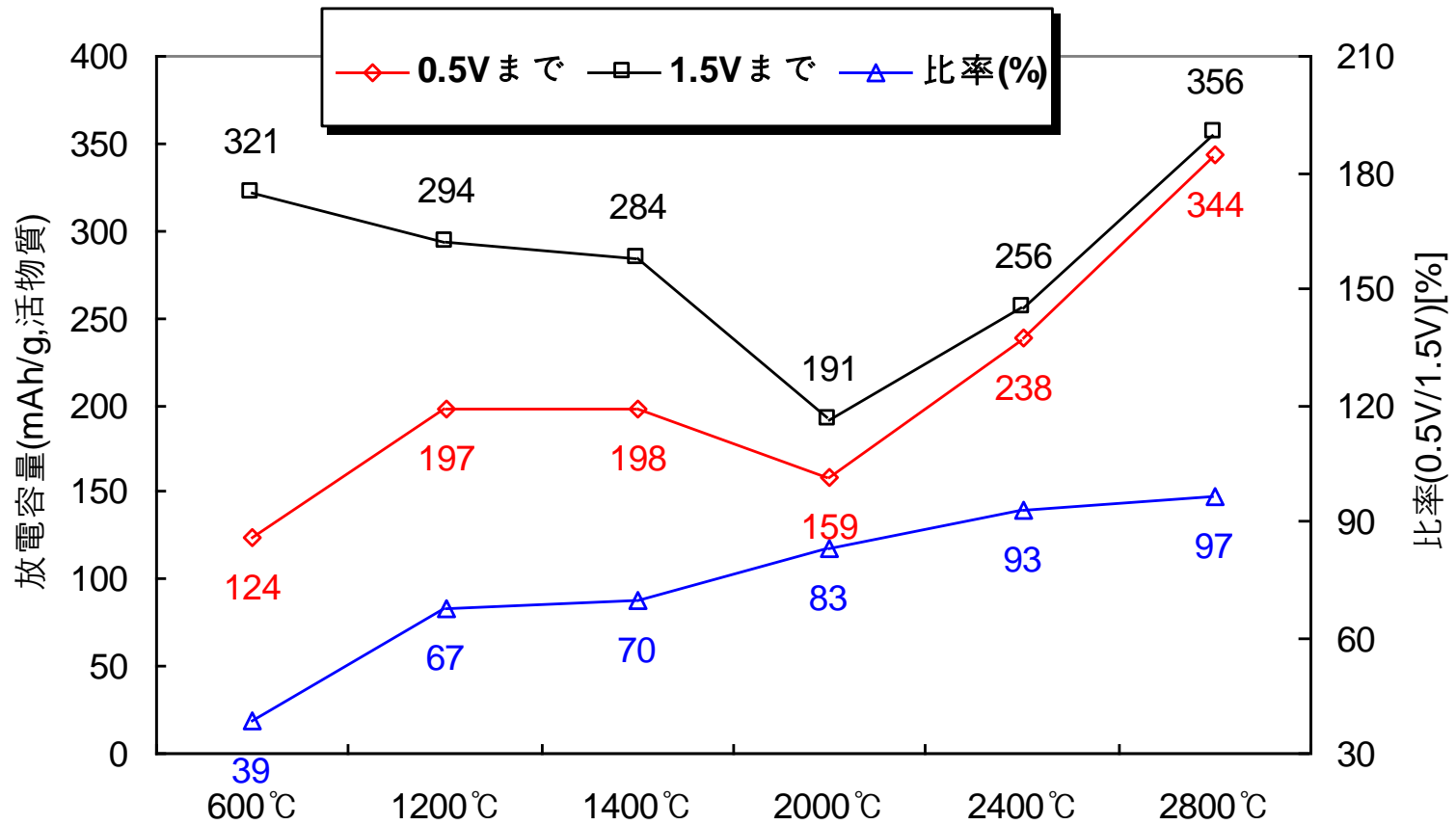
## Structural change of Cokes



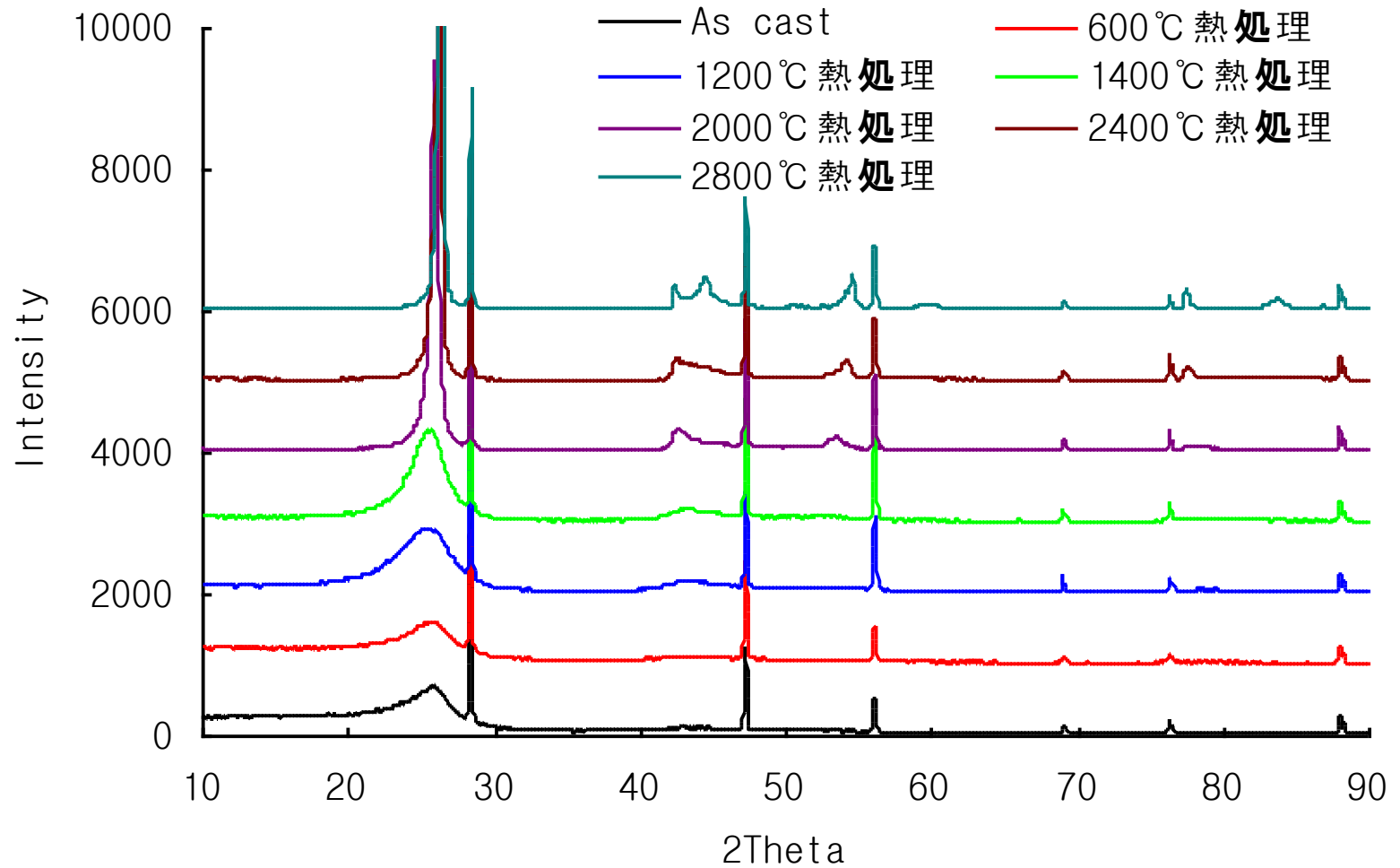
# MCMB



# MCMB



# MCMB

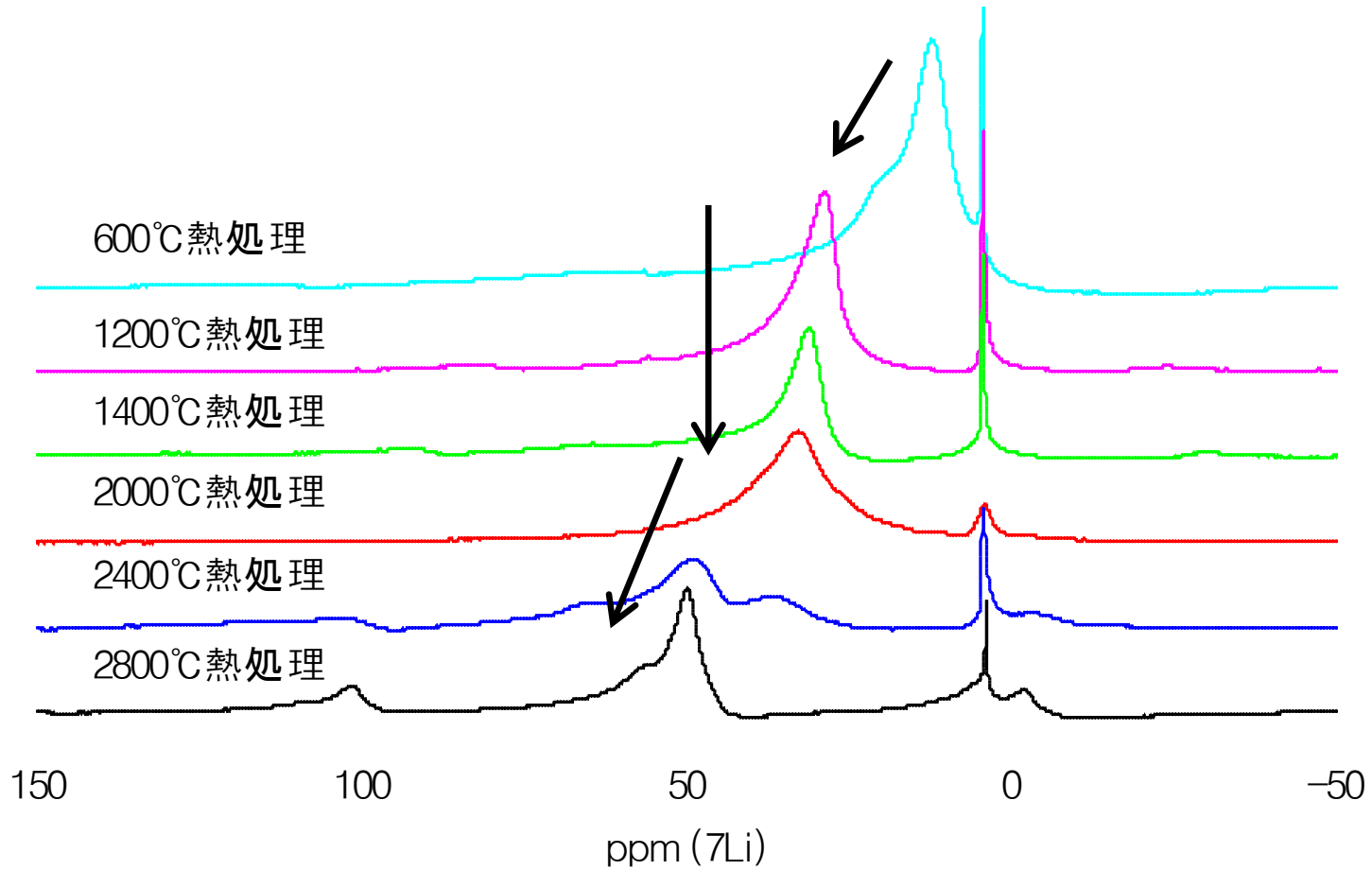


# MCMB

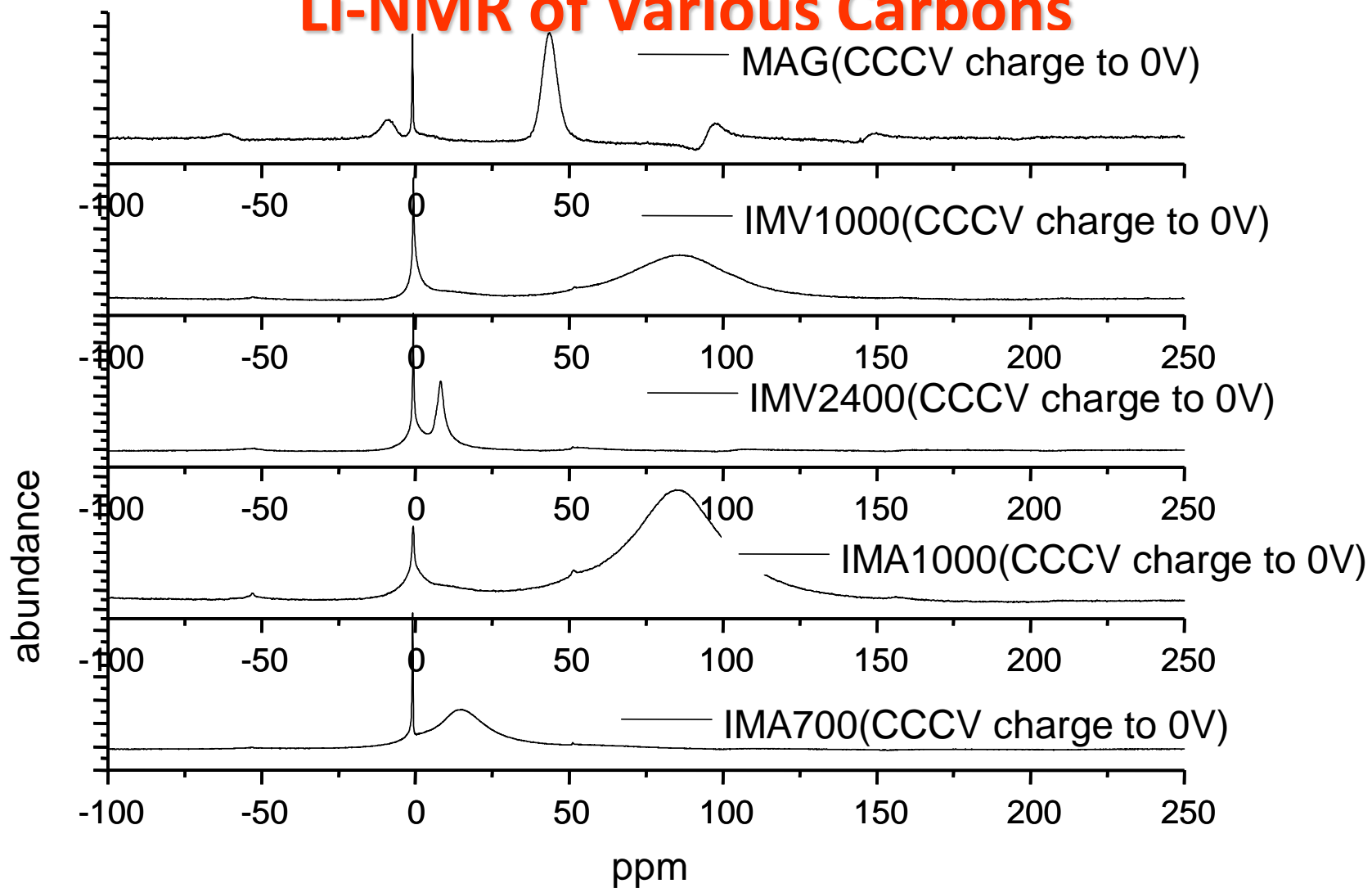
	$d_{002}$ (Å)	Lc002 (nm)
As cast	3.4945	3.1
600°C 熱處理	3.5138	3.1
1200°C 熱處理	3.5278	4.1
1400°C 熱處理	3.4876	6.8
2000°C 熱處理	3.4280	35.44
2400°C 熱處理	3.3887	53.70
2800°C 熱處理	3.3628	122.0



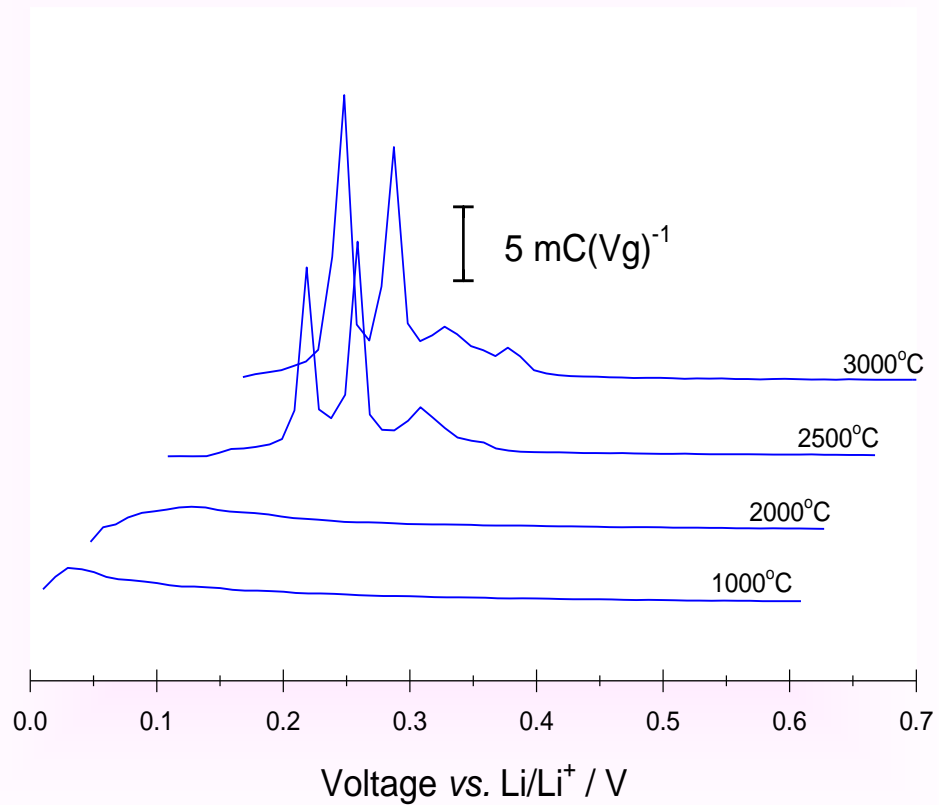
# MCMB



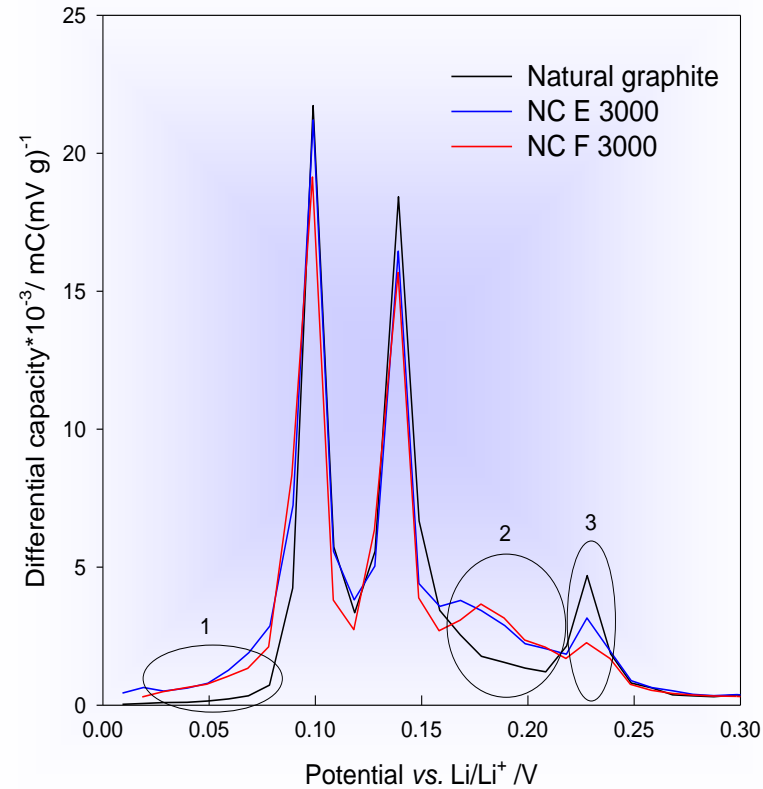
# Li-NMR of Various Carbons

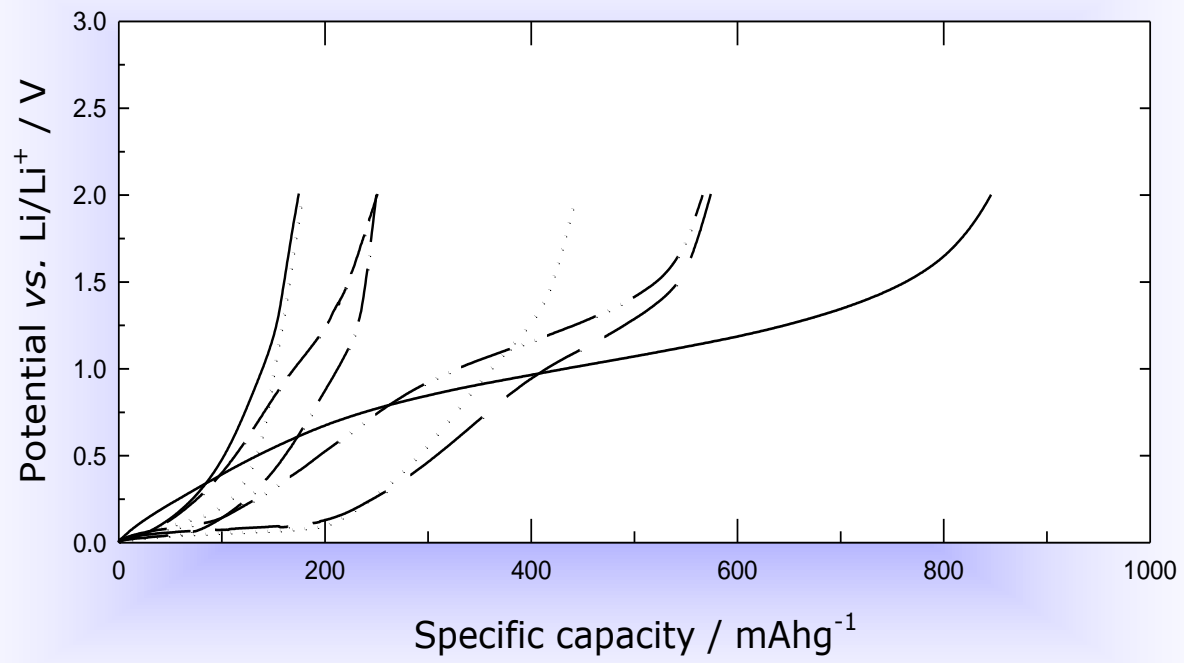


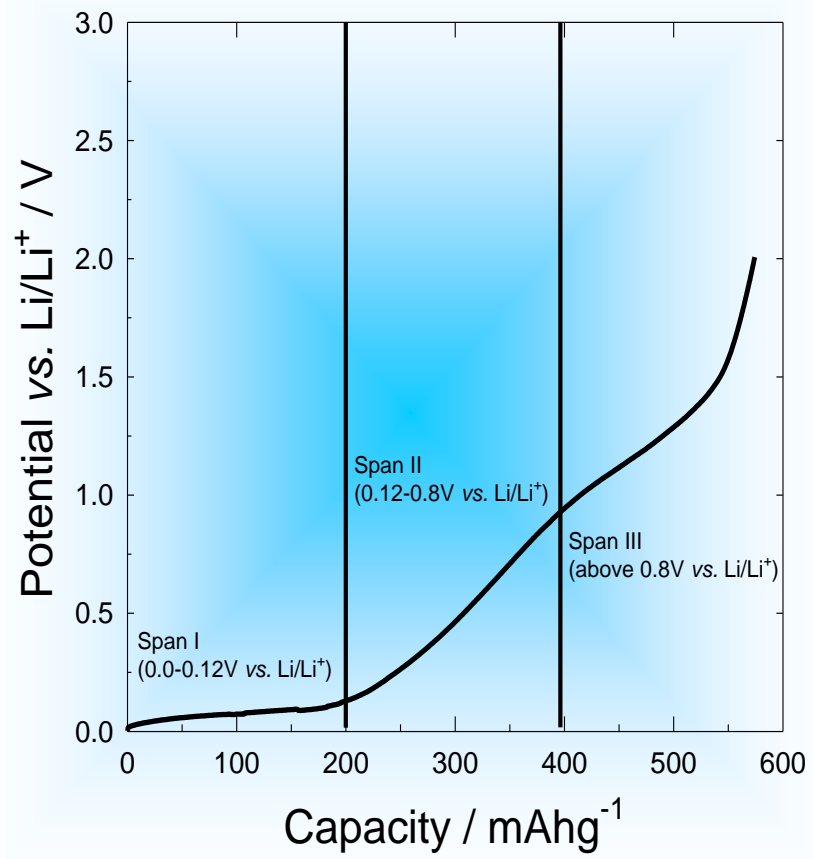
## Discharging EVS Profiles of NC E Series



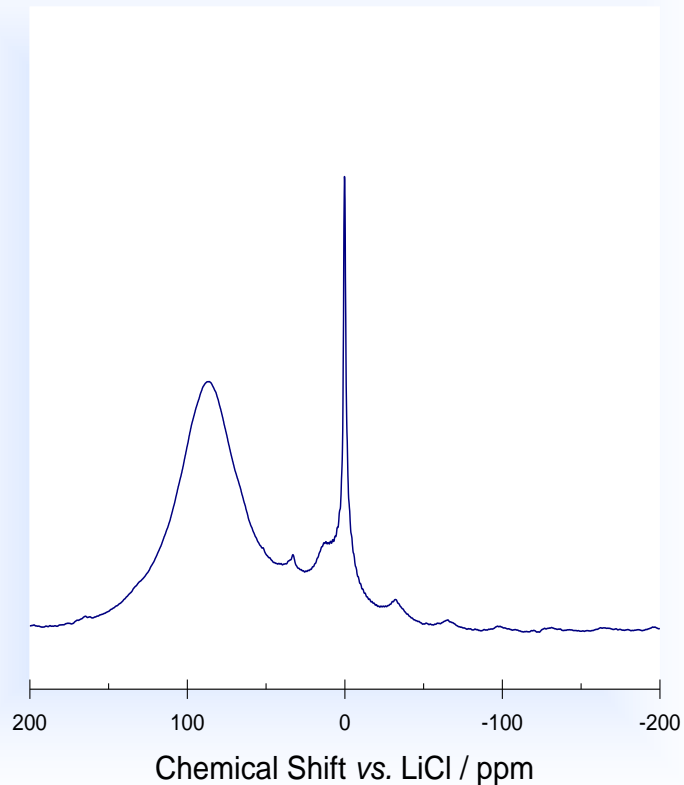
# Discharging EVS Profiles of natural and synthetic graphites





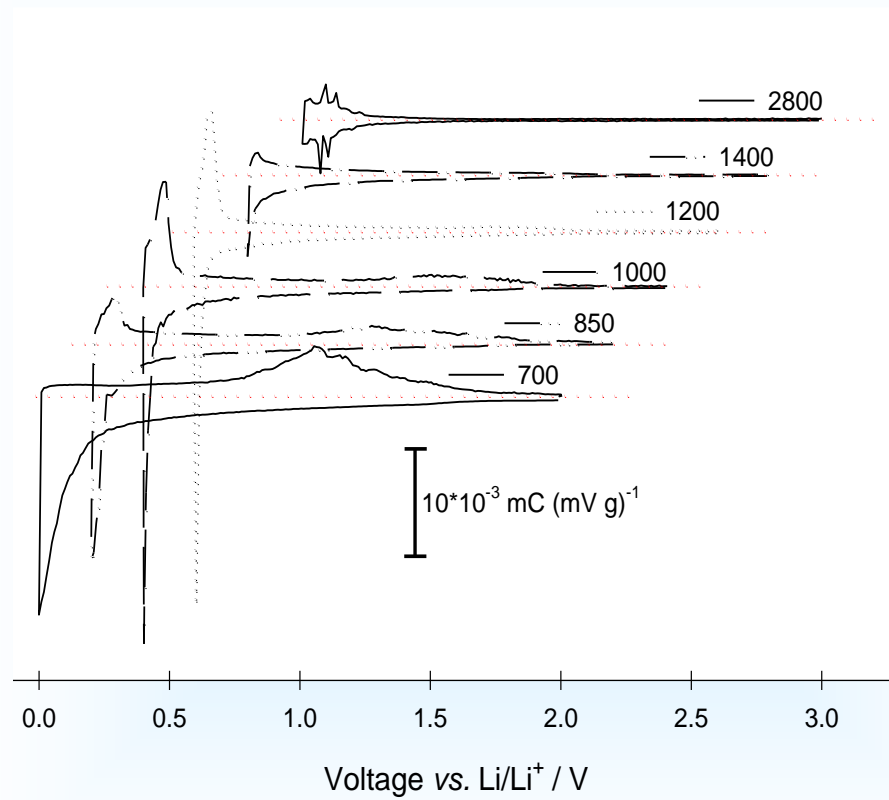


## $^7\text{Li}$ NMR MAS of MNIP 1000: fully-lithiated. RT

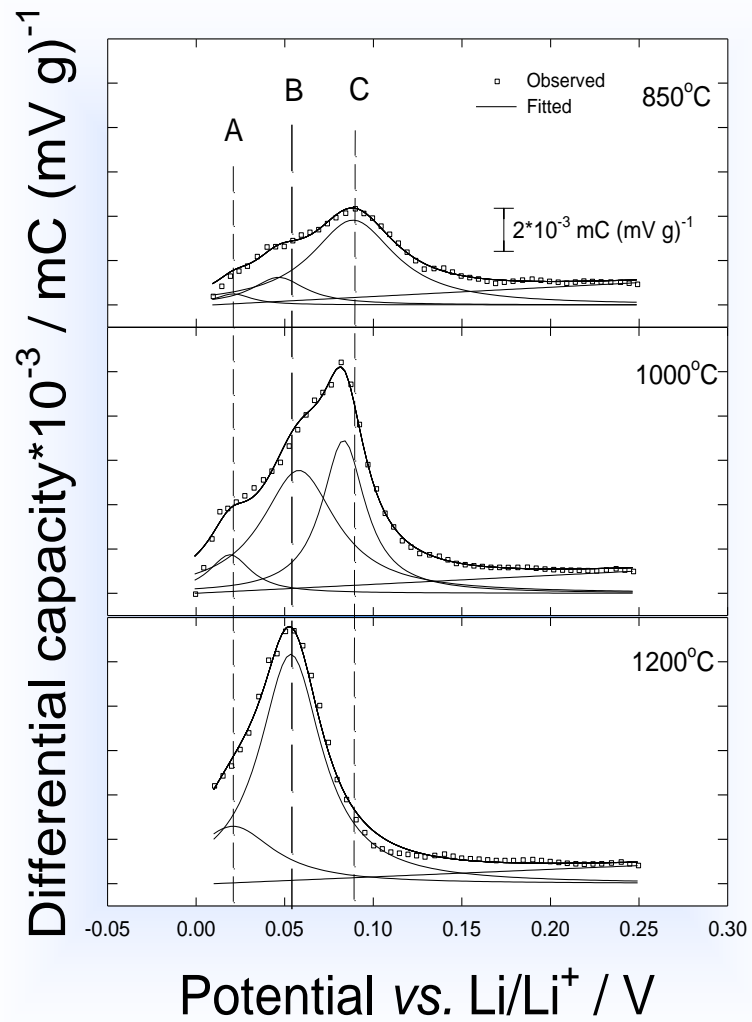


### **Four** Li's

- I. 0 ppm (Ionic): Electrolyte, decomposition product, etc.
- II. 6 ppm (Ionic): Site III (ultra-micropores)
- III. 12 ppm (Ionic):  
Site II (carbonaceous interlayers)
- IV. 82 ppm (Semi-metallic):  
Site I



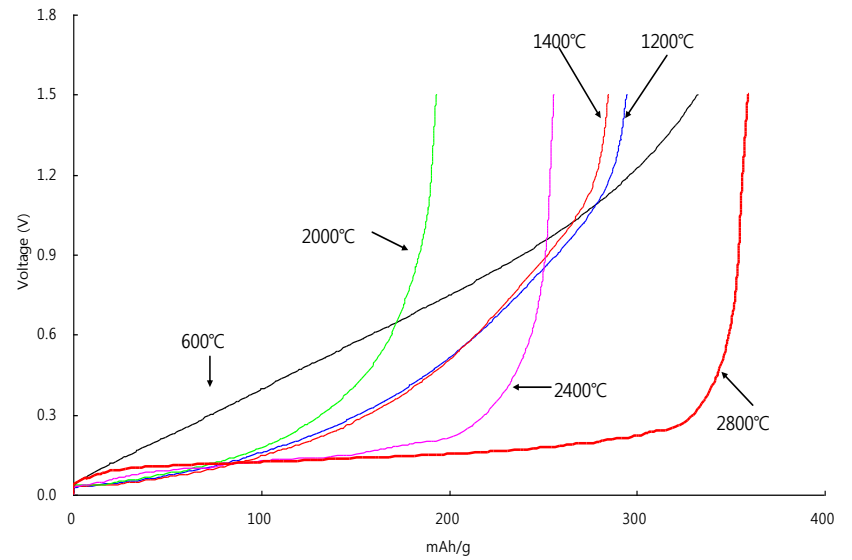
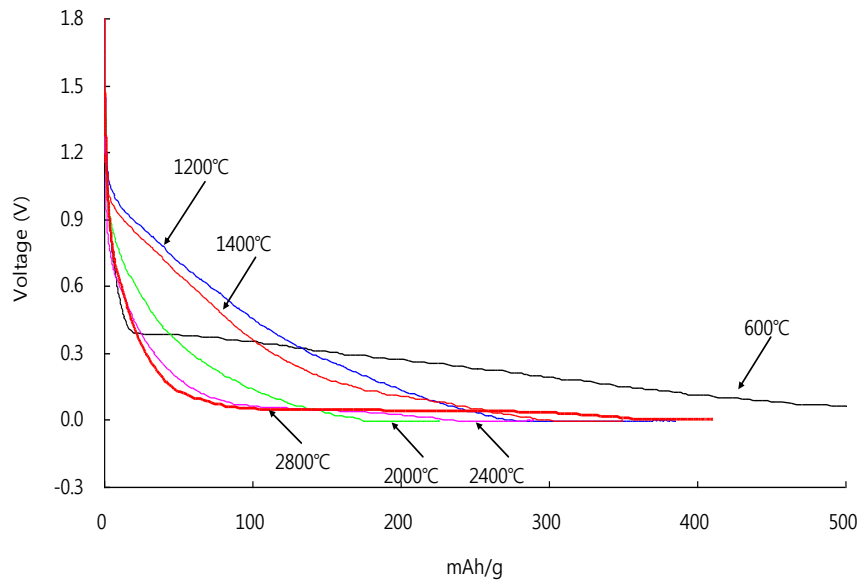




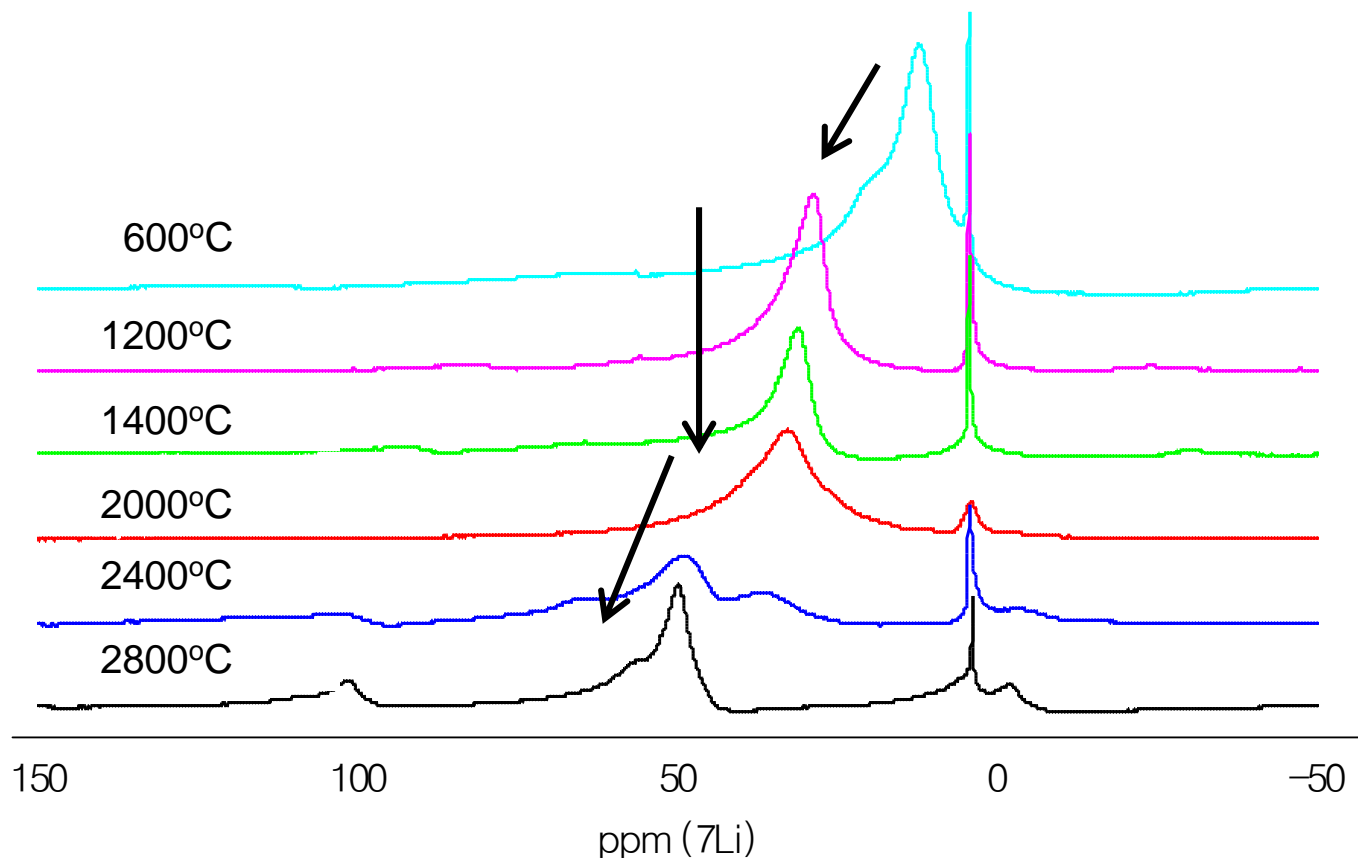
# Roles of Carbon for Anode of Li-ion Batteries

- Anodic Electrode to Hold Reduced Li-ion
  - Intercalation → Graphite
  - Surface Electron Transfer into Sealed Void  
→ Hard or Low Temperature  
Calcined Carbon
- Electron Conductive Material
  - Anodic Carbon and Cathode Material
- Expansion Moderator
  - Holding and Release of Ion Is Accompanied with  
Volumetric Change
  - Larger Capacity per Volume → Larger Expansion
- Moderation and Control of SEI
  - Irreversible Charge → Surface Coating, Composite  
Structure

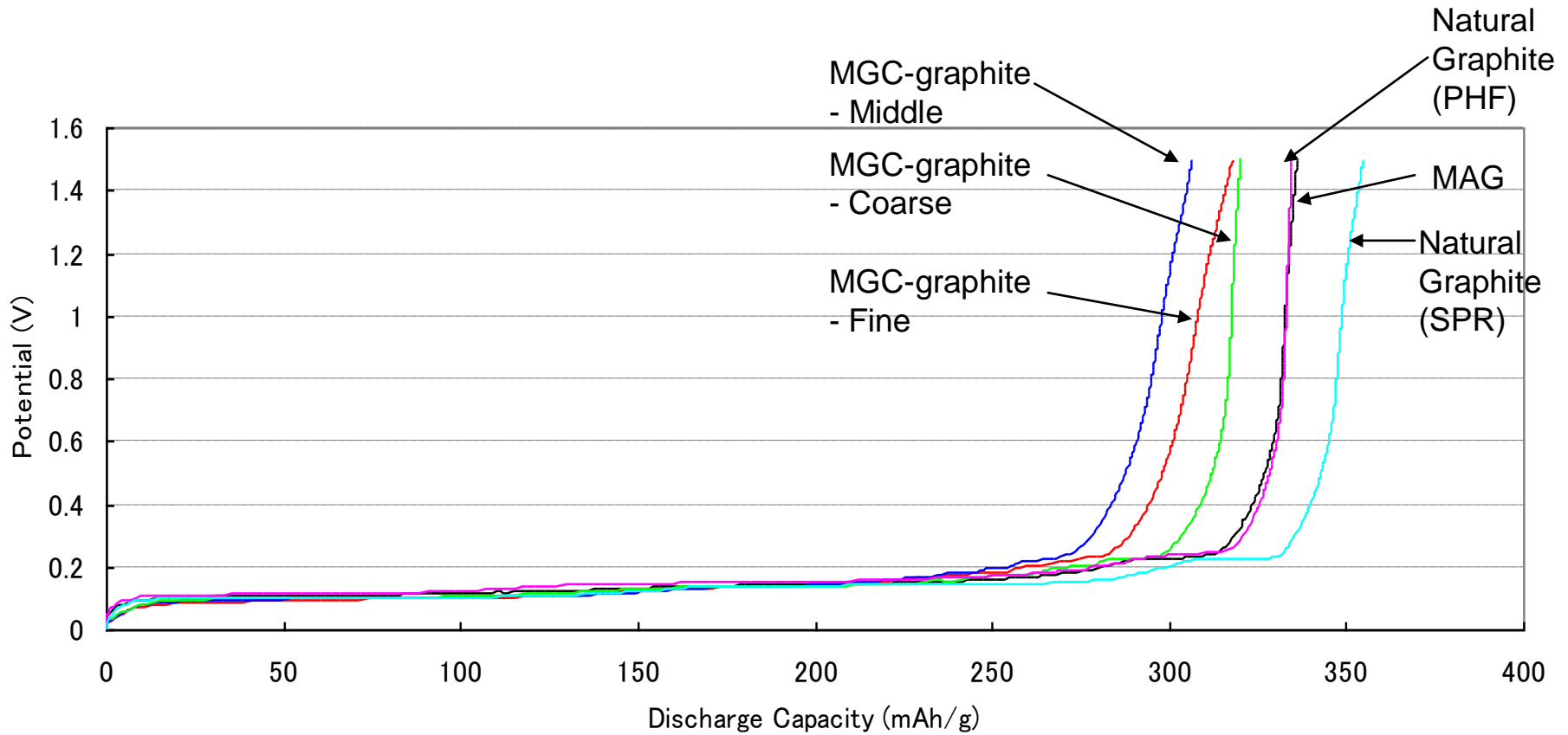
# Charge/Discharge Profiles of MCMB



# Li-NMR of Charged Li-ion in Heat Treated MCMB



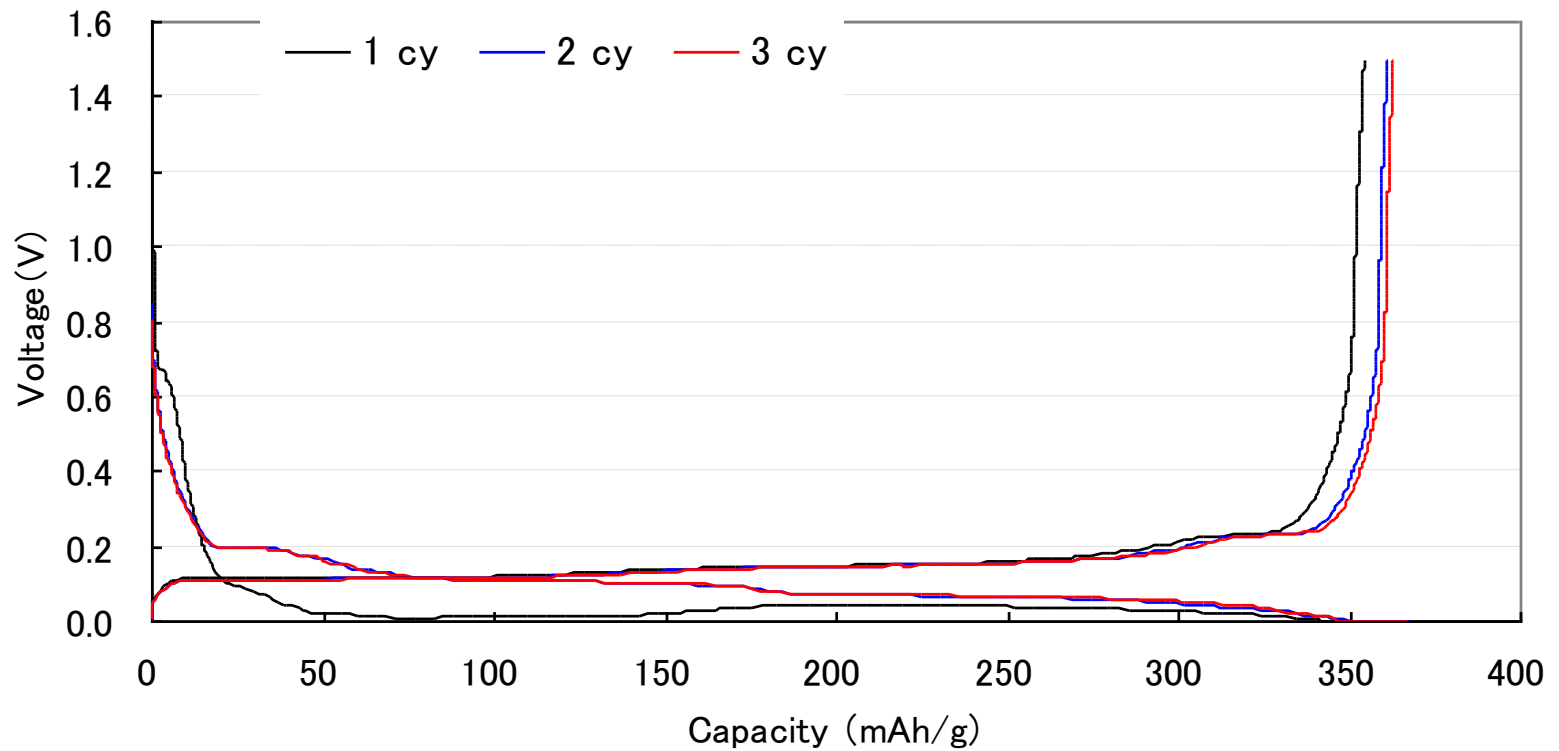
# Discharge Profiles of Typical Graphites



# Charge/Discharge Profiles of Synthetic Graphite

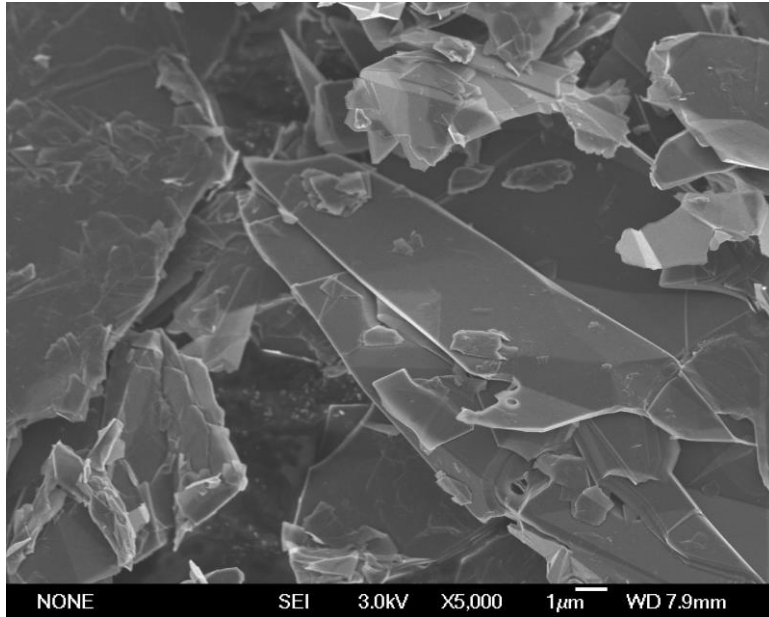
(MAG; Hitachi Chemical Co.)

0.1C, Half Cell Test, LiPF<sub>6</sub> 1M, EC+DEC



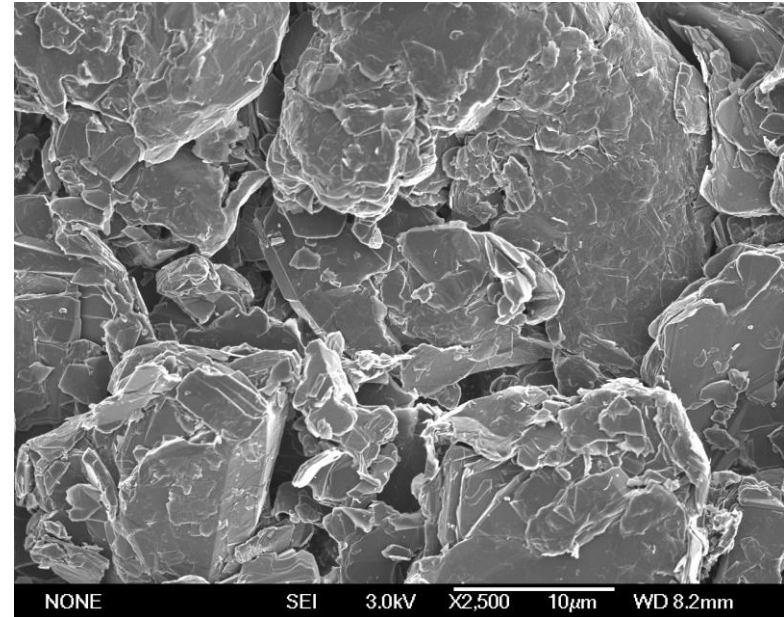
# Typical Graphites

## Natural Graphites



- Cheap
- **High graphitization degree**
- Large Irreversible Capacity
- Relatively poor Cycle Life
- **Poor Rate Capability**

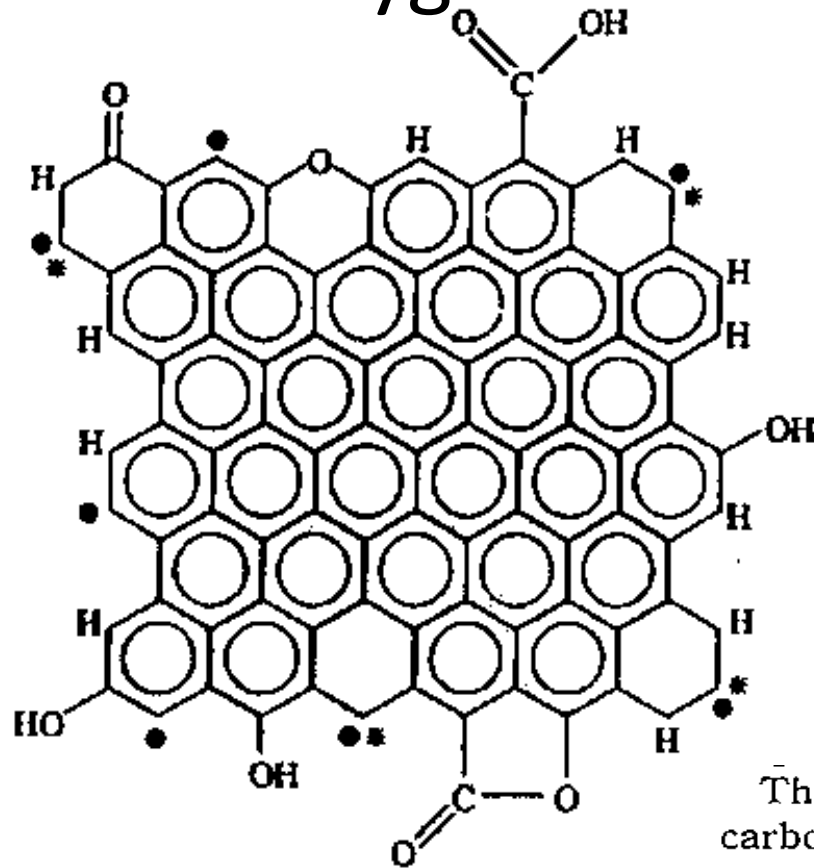
## Synthetic Graphites



- Good 1<sup>st</sup> cycle efficiency & Cycle Life
- **Relatively high graphitization degree**
- **Poor Rate Capability**

(MAG; Hitachi Chemical Co.)

# Surface Oxygen Functional Groups of AC

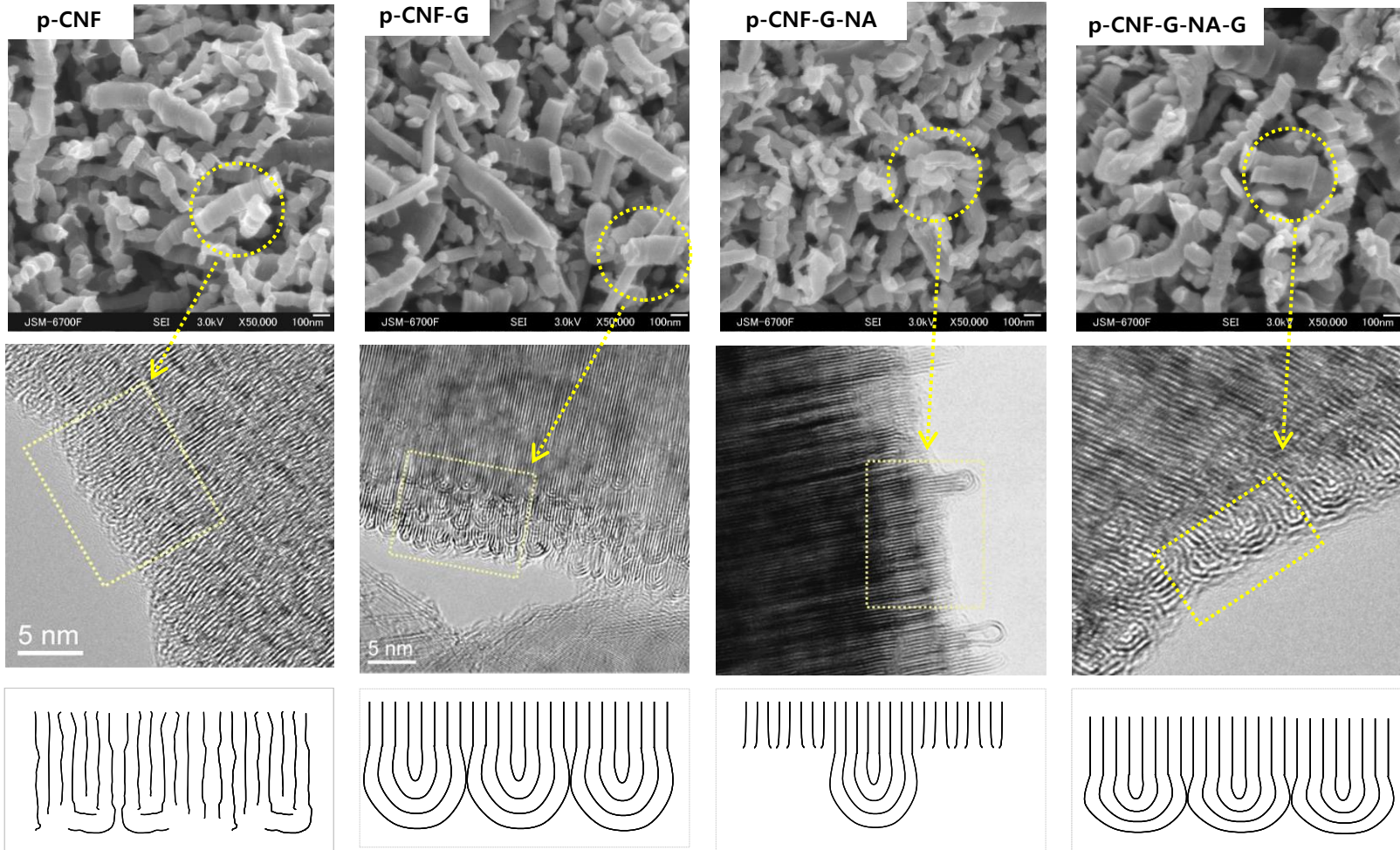


This structure is representative of an activated carbon with a crystallite width of 15 Å and an elemental analysis (by weight) of 87.5% C, 11.3% O, 1.2% H, ● represents an unpaired  $\sigma$  electron; ●\* represents an "in-plane  $\sigma$  pair" with \* being a localized  $\pi$  electron, (Radovic)



# SEM & TEM Images of PCNF Series

Ref.) S. Lim, et al.. *J. Phys. Chem. B* 108 (5), 1533 – 1536 (2004)



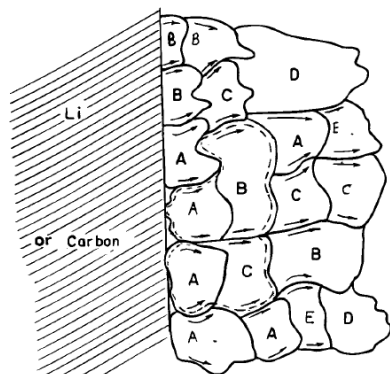
**According to the graphitization degree,  
we found some difference at edge plane by TEM analysis**

# Basic study of solid electrolyte interphase (SEI)

## Characteristics of SEI

- Reduction of electrolyte components on anodes on initial charge
  - Irreversible capacity loss
  - Decrease of first-cycle coulombic efficiency
- Passage of Li-ion migration, but high electronic resistivity
- Essential to determine the electrochemical properties and safety of Li-ion battery

< Schematic model of SEI formed on anodes >



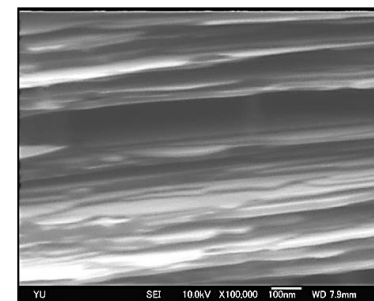
A =  $\text{Li}_2\text{O}$   
B =  $\text{LiF}$   
C =  $\text{Li}_2\text{CO}_3$   
D = Polyolefins  
E = Semicarbonates

E. Peled et al., J. Electrochem. Soc. 144 (1997) L208

## Previous researches on SEI

- Focused on SEI formation behavior of cross section of HOPG  
However, the cross section of HOPG was composed of edge planes and basal planes.

< Cross section of HOPG >



•T. Kim et al., Langmuir 22 (2006) 9086.

Necessary to prepare well-defined edge and basal surfaces

Ch.2

Study on SEI formation behavior on well-defined edge and basal surfaces prepared by carbon nanofibers as a model material.



# **Chap.2 Solid electrolyte interphase formation behavior of well-defined carbon surfaces for Li-ion battery systems**

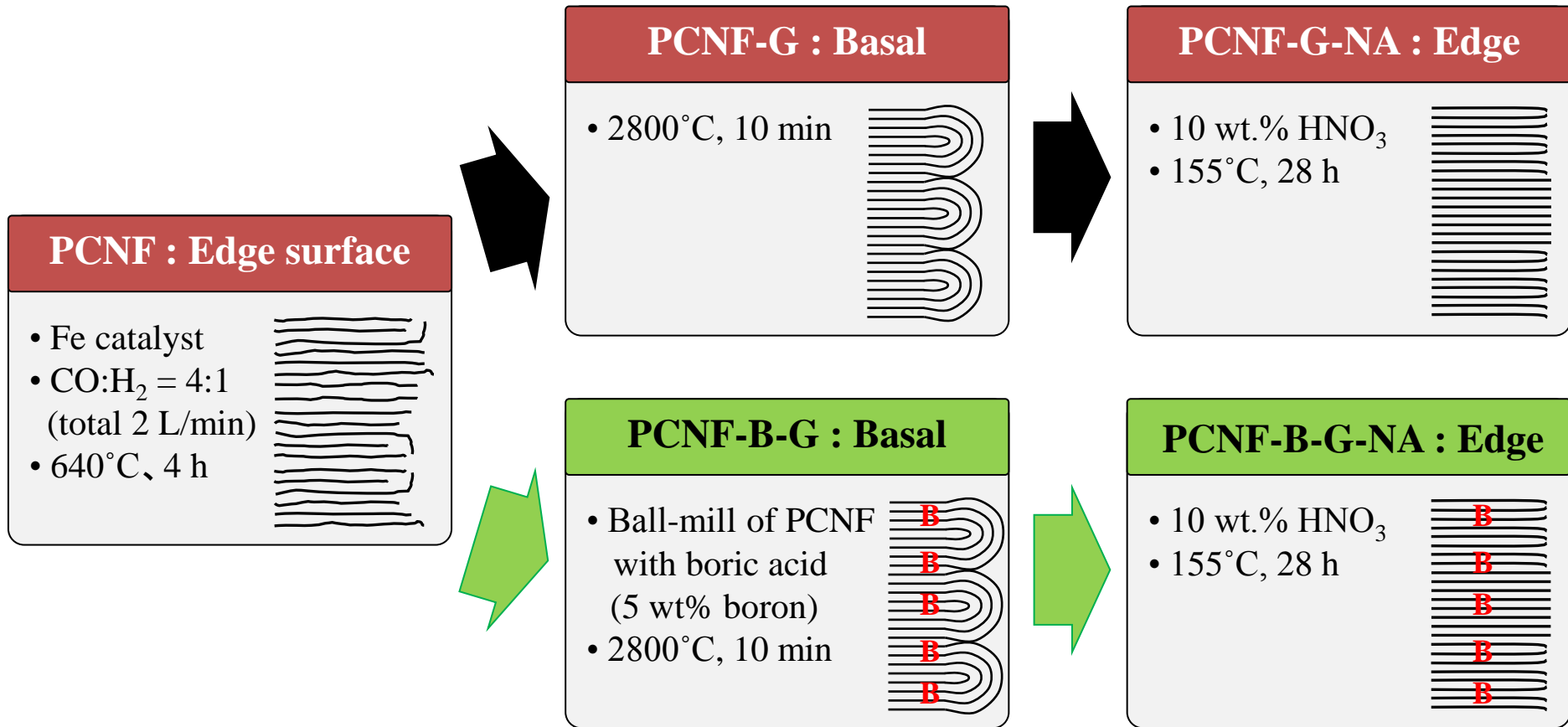
## ✓ Objectives

- To track the SEI formation behavior on well-defined edge and basal surfaces of platelet carbon nanofibers (PCNF) by TEM and to study the effects of its boron-doped surfaces on SEI formation

## ✓ Contents

- Preparation of PCNF with well-defined edge and basal surfaces as a model material
- Effect of edge and basal surfaces on the SEI formation
- Effect of boron doping on the SEI formation

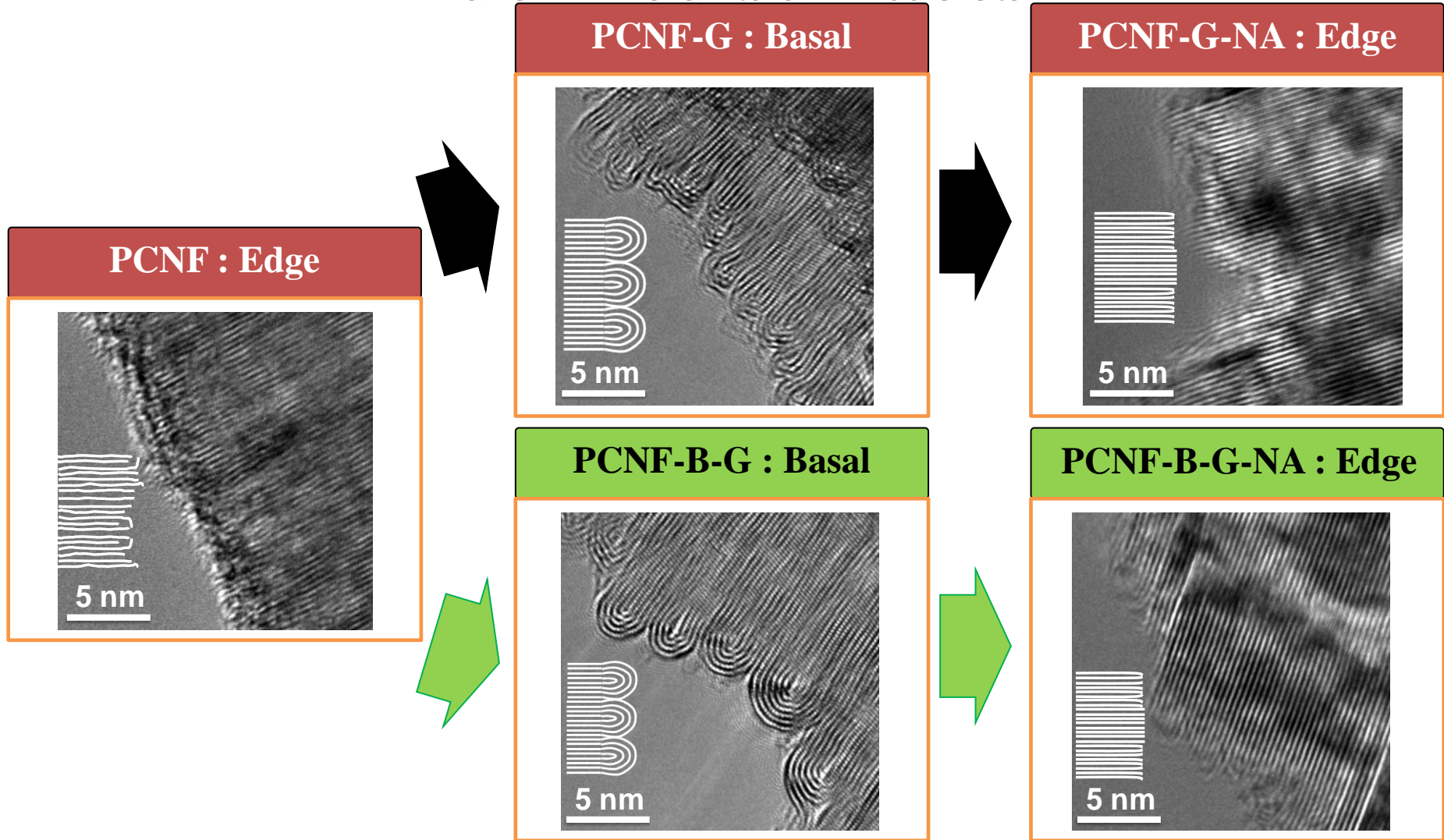
# Preparation of PCNFs with well-defined surfaces



**PCNF : Platelet carbon nanofibers**

\* S. Lim *et al.*, *J. Phys. Chem. B* 108 (2004) 1533.

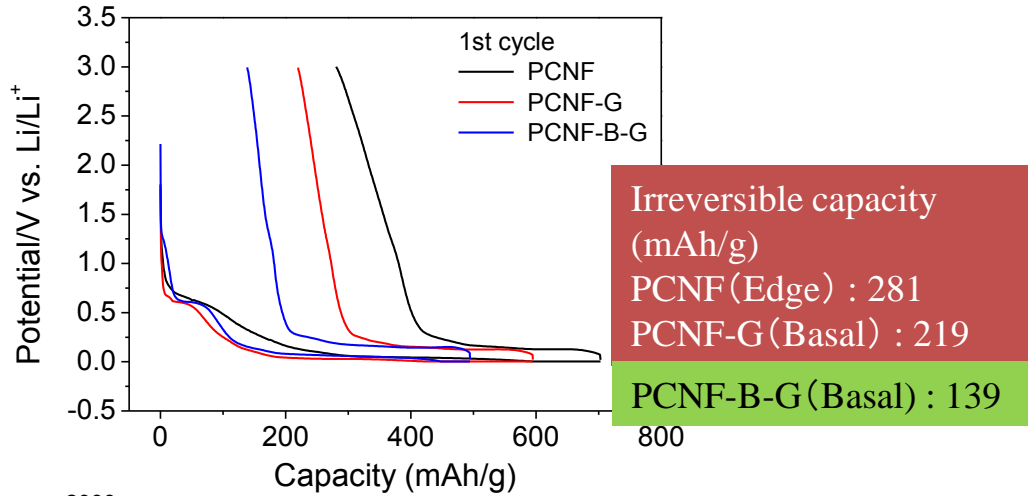
# TEM images of PCNFs with well-defined surfaces



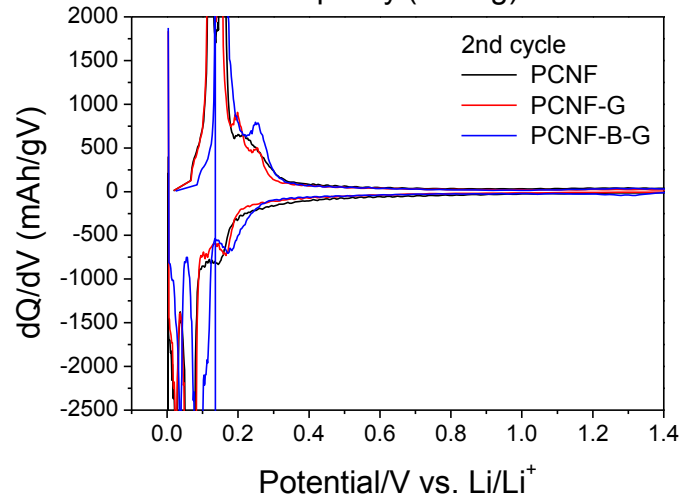
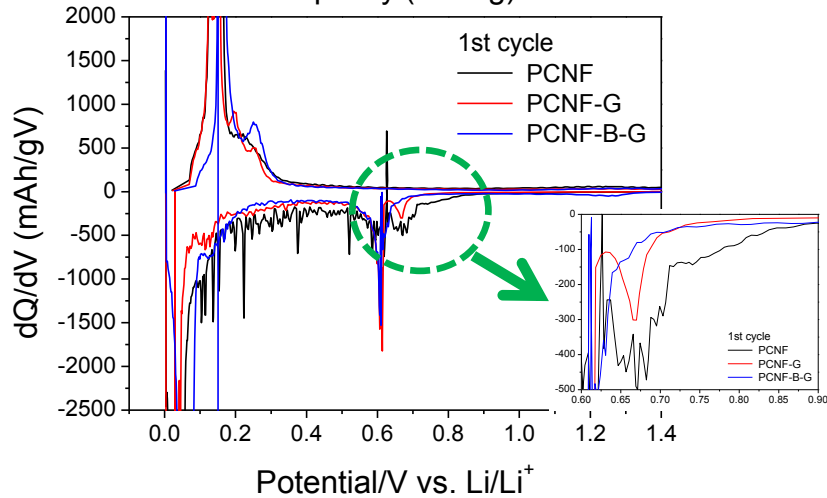
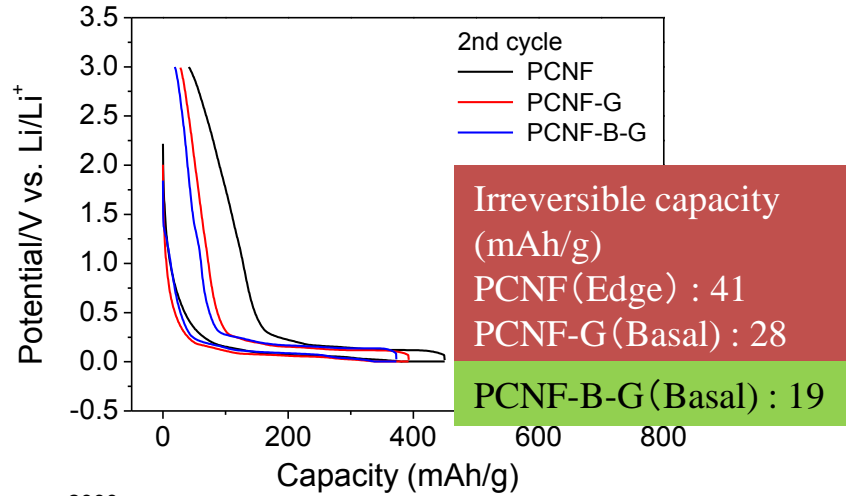


# Effect of edge and basal surfaces on the SEI formation

< First cycle >



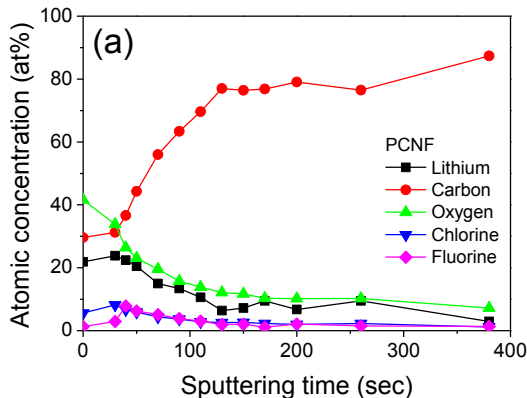
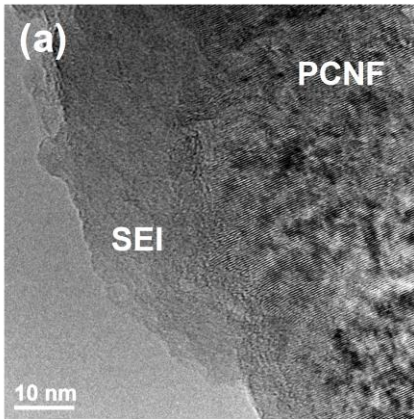
< Second cycle >



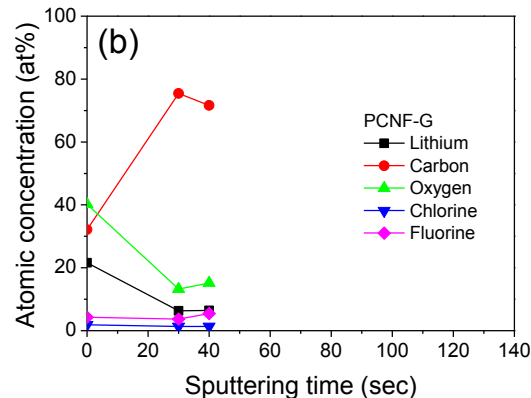
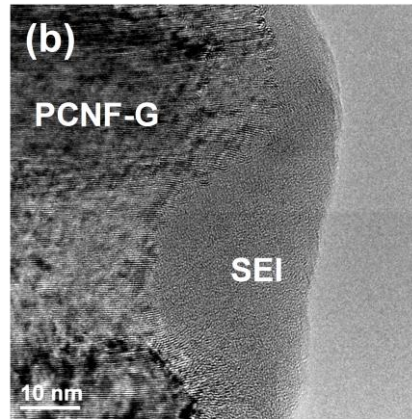
Electrolyte: 1 M LiClO<sub>4</sub> in EC/DEC (1:1 vol%), Binder : PVDF

# Effect of edge and basal surfaces on the SEI formation

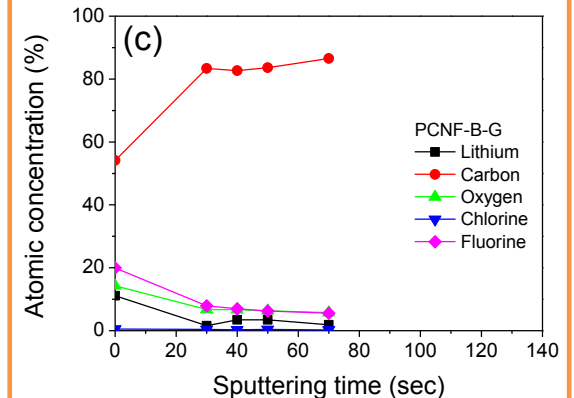
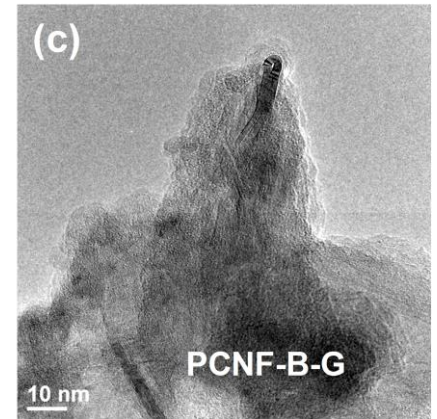
PCNF : Edge



PCNF-G : Basal



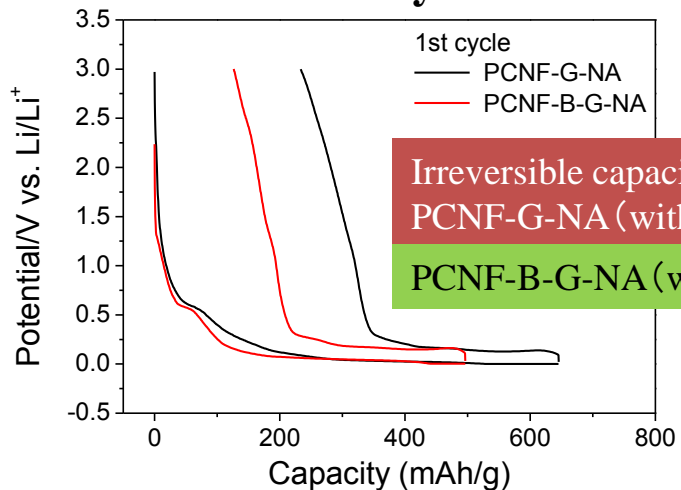
PCNF-B-G : Basal



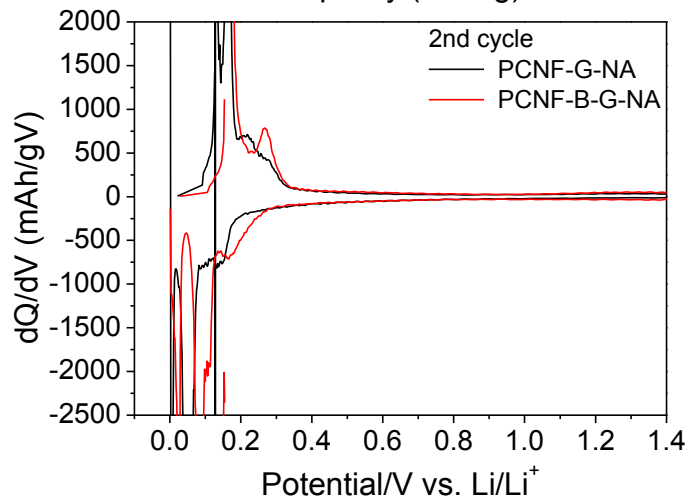
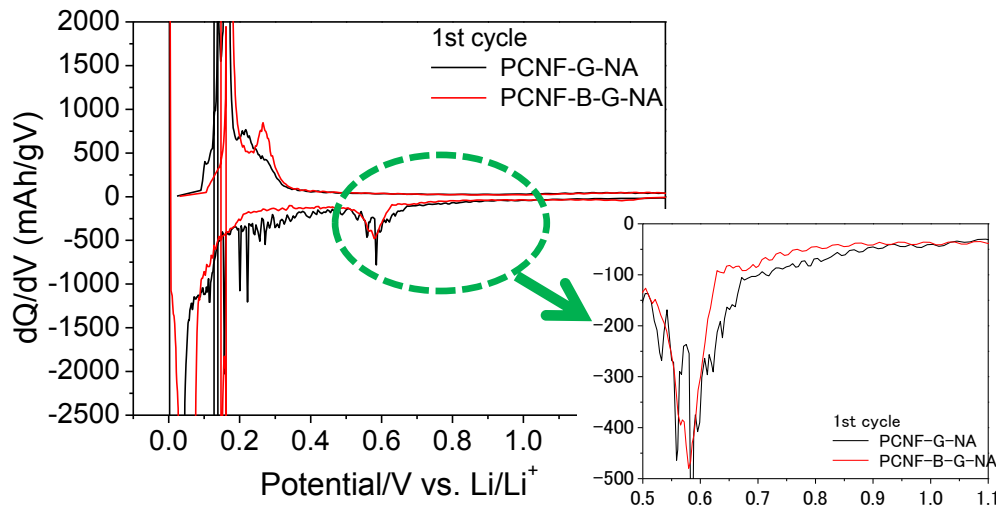
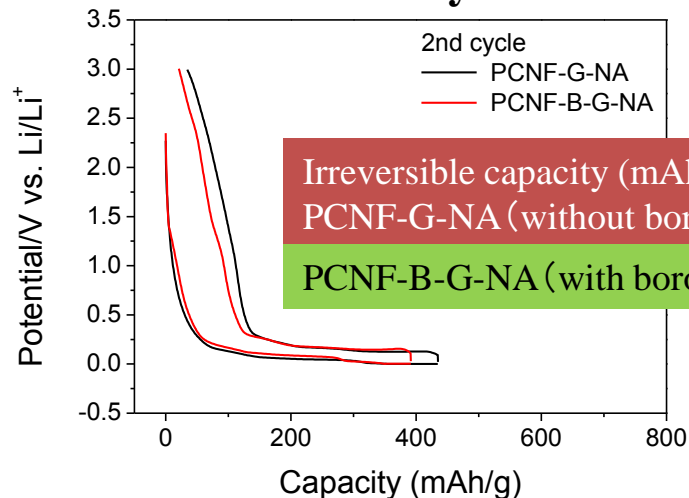
The XPS depth profiles indicated that the SEI of PCNF with edge surfaces was four times thicker than those of PCNF-G and PCNF-B-G with basal surfaces.

# Effect of boron doping on the SEI formation

< First cycle >



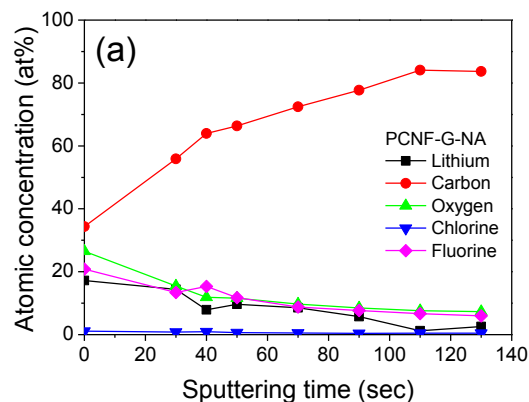
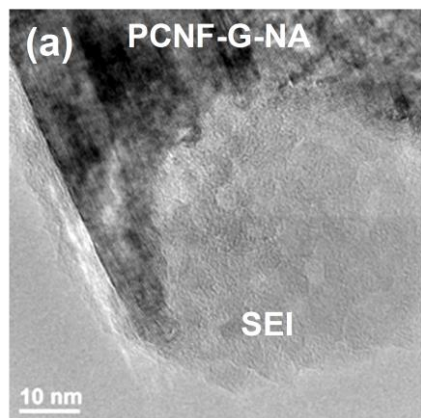
< Second cycle >



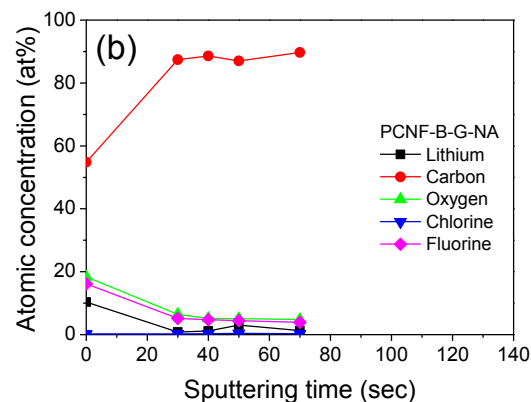
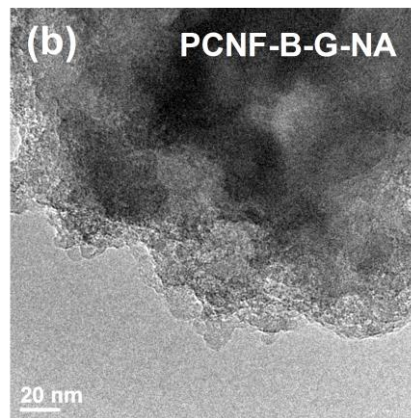


# Effect of boron doping on the SEI formation

PCNF-G-NA : Edge



PCNF-B-G-NA : Edge



**The SEI of PCNF-G-NA without boron doping was three times thicker than that of PCNF-B-G-NA with boron doping.**

# Explosion accident of Li-ion battery for EV (GM) 2012,04,12



**GM Worker Injured After Lithium-Ion Battery Explodes**

